



Sol–gel synthesis and photocatalytic activity of ZnO–SnO₂ nanocomposites



Abdessalem Hamrouni^{a,b}, Noomen Moussa^a, Francesco Parrino^b, Agatino Di Paola^{b,*},
Ammar Houas^{a,c}, Leonardo Palmisano^b

^a Unité de Recherche Catalyse et Matériaux pour l'Environnement et les Procédés URCMEP (UR11ES85), Faculté des Sciences de Gabès/Université de Gabès, Campus Universitaire, Cité Erriadh, 6072 Gabès, Tunisie

^b Dipartimento di Energia, Ingegneria dell'Informazione e Modelli Matematici (DEIM), University of Palermo, Viale delle Scienze Ed. 6, Palermo 90128, Italy

^c Al Imam Mohammad Ibn Saud Islamic University (IMSIU), College of Sciences, Department of Chemistry, Riyadh 11623, Saudi Arabia

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ABSTRACT

ZnO–SnO₂ nanocomposites were synthesized by a facile sol–gel synthesis route and characterized through X-ray diffraction, BET specific surface area analysis, UV–vis diffuse reflectance spectroscopy, photoluminescence spectroscopy, X-ray photoelectron spectroscopy, scanning electron microscopy, transmission electron microscopy. The photocatalytic activity of the samples was tested using the degradation of 4-nitrophenol under UV light as model reaction. The ZnO/SnO₂ molar ratio was varied in order to study its influence on the photoefficiency of the samples. The ZnO–SnO₂ nanocomposites showed higher photoactivity than the pure oxides and in particular the sample with ZnO/SnO₂ molar ratio equal to 1/0.05 resulted the best one among the tested powders. The high activity of the mixed samples was attributed to the presence of heterojunctions between the two oxides, which allows an improved charge separation of the photogenerated electron–hole pairs, due to the differences between the energy levels of the conduction and valence bands of ZnO and SnO₂. Photovoltage measurements were performed to determine the energy band structure of the ZnO–SnO₂ heterojunction.

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1. Introduction

Since Fujishima and Honda in 1972 [1] described the photoelectrolysis of water using supported TiO₂ as photoanode, heterogeneous photocatalysis has been object of considerable attention in the field of green chemistry and of “Advanced oxidation processes” (AOPs) [2–4] mainly for air and water remediation [5].

When a photocatalyst is irradiated with light of energy equal or higher than its band gap energy, electron/hole pairs (e[−]/h⁺) are produced. In an aqueous or gaseous medium, reactants can adsorb on the photocatalyst surface and react directly or indirectly with the photogenerated electrons and holes.

ZnO is a semiconductor material with band gap energy of 3.2 eV, thus absorbing in the UV region of the electromagnetic spectrum [6]. ZnO has been broadly studied as a photocatalyst due to its low cost, and its good optoelectronic and catalytic properties [5]. In

several reports, ZnO was found efficient for the photodegradation of many non-biodegradable organic compounds [7–14] due to the position of its valence band which gives a strong oxidizing power to the photogenerated holes [7]. However, the photocatalytic activity of ZnO is affected by the fast recombination of the photogenerated charge carriers which reduces the efficiency of the photocatalytic processes. In order to overcome this problem, simultaneous doping with two kinds of atoms [15,16] or coupling ZnO with other semiconductors of suitable electronic properties [17–47] have been found to be a viable strategy to increase the charge separation of the photogenerated electron/hole pairs. For this reason, many coupled semiconductor systems have been proposed as photocatalysts such as ZnO–TiO₂ [17–19], ZnO–CdS [20,21], ZnO–AgBr [22], ZnO–Ag₂S [23]. In particular, ZnO–SnO₂ coupled oxides [24–47] have been largely studied to ameliorate the photocatalytic activity of ZnO. The performance of each system depends essentially on the position of the conduction and valence bands of the two semiconductors which directly influences the separation and the recombination of the photogenerated e[−]/h⁺ pairs [48].

Tennakone and Bandara were the first to investigate ZnO–SnO₂ composites as potential photocatalysts for the sensitized degradation of dyes [24,25]. Their catalytic activity was superior to that

* Corresponding author. Tel.: +39 91 238 63729; fax: +39 702 5020.
E-mail addresses: hamrouni-28@hotmail.fr (A. Hamrouni),
agatino.dipaola@unipa.it (A. Di Paola).

of individual ZnO or SnO₂ and was attributed to the wider charge separation ability of the composite system. Other researchers [24–30,34,42] related the enhanced photocatalytic properties of the ZnO–SnO₂ systems to the vectorial transfer of electrons and holes between ZnO and SnO₂ but only recently this hypothesis has been confirmed by determination of the band alignment of the ZnO–SnO₂ heterostructure by X-ray photoelectron spectroscopy [45].

Coupled ZnO/SnO₂ photocatalysts have been obtained by a variety of different processes. The most common method was the heterogeneous co-precipitation from aqueous solutions of ZnSO₄ and SnCl₄ using NaOH [26–28] or NH₃·H₂O [29,47]. Nanosized ZnO/SnO₂ samples were synthesized by homogeneous co-precipitation using ethyl acetate as precipitating agent [31] or by fractional homogeneous co-precipitation using urea as the latent precipitant under boiling reflux conditions [43]. Yang et al. [41] prepared ZnO/SnO₂ composites via cetyltrimethylammonium bromide assisted co-precipitation starting from Zn(NO₃)₂ and SnCl₄ solutions. Recently, a novel method employed the high-gravity reactive precipitation using a rotating packed bed [44].

Other methods for preparing ZnO–SnO₂ photocatalysts included the hydrothermal method using an aqueous solution containing ZnO rods, SnCl₄ and NaOH [32] or a mechanochemical processing of anhydrous chloride precursors with Na₂CO₃ [30]. Uddin et al. [45] prepared nanoporous ZnO–SnO₂ photocatalysts by a two step procedure involving the homogeneous precipitation of SnO₂ particles combined with a hydrothermal treatment and the reaction of the as-prepared SnO₂ particles with zinc acetate. Network structured ZnO/SnO₂ samples were synthesized by a two-step solvothermal method [35].

Only few works have concerned the employment of the sol–gel method to obtain ZnO/SnO₂ composites [36–40]. In particular, Zhenya et al. [36] used Zn(CH₃COO)₂·2H₂O, oxalic acid and SnCl₂·2H₂O as raw materials and the samples were treated under high pressure at 6 GPa. One-dimensional ZnO–SnO₂ nanofibers were synthesized by combination of sol–gel process and electrospinning technique [37]. Prevalently, the sol–gel dip coating technique was used to prepare ZnO/SnO₂ thin films using zinc acetate and SnCl₄ dissolved in CH₃OH [39] or CH₃COCH₂COCH₃ and CH₃OC₂H₄OH [41]. Recently, Torres Martínez et al. [40] studied the antibacterial performance of thin films prepared from mixtures of Zn(CH₃COO)₂·2H₂O, and SnCl₂·2H₂O dissolved in ethylene glycol and ethanol solutions, respectively.

In a previous study Hamrouni et al. [47] demonstrated that ZnO–SnO₂ nanocomposites, prepared with the co-precipitation method, were more efficient than the corresponding pure oxides. The aim of this study was to further improve the photocatalytic activity of the ZnO–SnO₂ coupled catalysts. ZnO–SnO₂ powders were prepared by a novel facile sol–gel synthesis route. The samples were tested for the photodegradation of 4-nitrophenol (4-NP) under UV-light irradiation in solid-liquid phase. 4-NP was chosen as a probe molecule for the photocatalytic runs because substituted phenols are ubiquitous contaminants in water streams [49]. The influence of the calcination temperature and the ZnO/SnO₂ molar ratio was investigated. A morphological and photoelectrochemical investigation was carried out in order to establish a correlation between ZnO–SnO₂ heterojunction effect and enhanced photocatalytic activity. For the first time, the proposed charge transfer mechanism between ZnO and SnO₂ was based on the experimental determination of the position of the conduction and valence bands of the two semiconductors by means of photovoltage measurements carried out in slurry systems. These data, strictly related to the real reaction conditions, are more reliable than those obtained with other methods reported in literature [39,50].

2. Experimental

2.1. Synthesis

Zinc acetate dihydrate (Sigma–Aldrich) and tin chloride pentahydrate (Sigma–Aldrich) were used as starting materials. Methanol (Sigma–Aldrich) and 28% ammonium hydroxide solution (Fluka) were used as solvent and additive, respectively. All the chemical products were used as received without any further purification. The synthesis of the ZnO–SnO₂ mixed photocatalysts started by slowly adding a 0.1 M methanol solution of Zn(CH₃COO)₂·2H₂O, previously prepared at 70 °C under vigorous stirring, to a 0.1 M methanol solution of SnCl₄·5H₂O prepared in the same conditions. Typically, 8.12 g of Zn(CH₃COO)₂·2H₂O were dissolved in 370 ml of methanol and the amount of SnCl₄·5H₂O solution was opportunely determined in order to obtain powders with molar ratio ZnO/SnO₂ equal to 1/0.02, 1/0.05, 1/0.25 labeled as ZnSn_{0.02}, ZnSn_{0.05} and ZnSn_{0.25}, respectively. The obtained solutions were continuously stirred for 2 h, keeping the temperature at 70 °C. Then, the NH₄OH solution was added dropwise until pH reached the value of 8. The solutions became gels which were dried for 20 h at 110 °C to produce xerogels. Finally, the photocatalysts were obtained by calcining the xerogels for 2 h at different temperatures (500, 600 and 700 °C). The pure oxides were synthesized in the same way by using only the corresponding precursors.

2.2. Characterization

X-ray diffraction patterns of the powders were recorded at room temperature by an Itai Structures APD 2000 powder diffractometer using the Cu K_α radiation and a 2θ scan rate of 2°/min. The mean grain size of the photocatalysts was estimated using the Debye–Scherrer equation:

$$D = \frac{0.9\lambda}{\beta \cos \theta} \quad (1)$$

where D is the mean grain size (nm), λ is the wavelength of the X-ray radiation (0.154 nm), β is the corrected full width at half maximum height and θ is the diffraction angle. The specific surface area (SSA) of the samples was determined with a Micro-metrics Flow Sorb 2300 instrument by using the single-point BET method. UV–vis diffuse reflectance spectra (DRS) of the samples were recorded by using a Shimadzu UV-2401PC spectrophotometer. BaSO₄ was used as a reference and the spectra were recorded in the range 200–800 nm.

The morphology of the catalyst was examined using a Philips XL30 ESEM scanning electron microscope (SEM), operating at 30 kV on specimens upon which a thin layer of gold was deposited. Transmission electron microscopy (TEM) measurements were obtained using a JEM-2100 (JEOL, Japan) operating at 200 kV accelerating voltage, equipped with an energy dispersive X-ray spectrometer (EDS, Oxford, UK) suitable for element identification. The powders were dispersed in isopropanol and a small drop was deposited on a 200 mesh carbon-coated copper grid, which was introduced into the TEM chamber analysis after complete solvent evaporation.

X-ray photoelectron spectroscopy (XPS) analyses were performed with a VG Microtech ESCA 3000 Multilab equipped with a dual Mg/Al anode. The spectra were excited by the non-monochromatised Al K_α source (1486.6 eV) run at 14 kV and 15 mA.

2.3. Photovoltage and photoluminescence measurements

Photovoltage measurements were performed in a Pyrex reactor with a total volume of 150 mL. 100 mg of photocatalyst were added to 100 mL of a 0.1 M NaNO₃ solution. A platinum electrode was used as working electrode and the reference was an Ag/AgCl

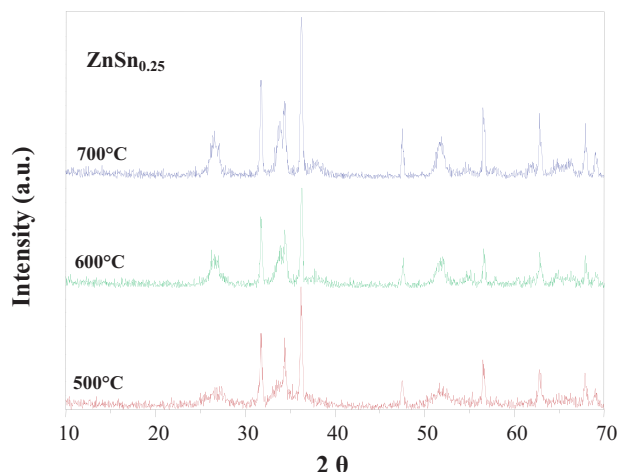


Fig. 1. XRD patterns of $\text{ZnSn}_{0.25}$ calcined for 2 h at different temperatures.

electrode. After degassing for 30 min with N_2 , 20 mg of methyl viologen (1,1'-dimethyl-4,4'-bipyridinium dichloride) were added to the suspension. Then, the mixture was once more degassed for 30 min. The pH was adjusted using HNO_3 and NaOH solutions. During the experiments, the suspension was irradiated by two UV lamps (Philips HPK 125 W). The values of pH and potential (V) were measured using a pH meter Thermo Orion 720A and a multimeter Metex 3800, respectively.

Photoluminescence spectra were recorded at room temperature with a Jobin Yvon Spex Fluoromax II spectrofluorimeter equipped with a Hamamatsu R3896 photomultiplier.

2.4. Photocatalytic experiments

A cylindrical Pyrex batch photoreactor containing 0.15 L of the 4-NP solution (20 mg/L) was used to carry out the photodegradation experiments. For each photocatalytic run, the amount of photocatalyst was chosen in order to ensure that almost all the photons emitted by the lamp were absorbed from the suspension. A 125 W medium pressure Hg lamp (Helios Italquartz, Italy) with a maximum emission at about 365 nm was axially positioned within the photoreactor. The photon flux emitted by the lamp was $\Phi_i = 14 \text{ mW cm}^{-2}$. The temperature of the solution was controlled by circulation of water through a Pyrex thimble surrounding the lamp. A magnetic stirrer was used to guarantee the homogeneity of the reacting mixture. Before the light was turned on, the suspension was kept in the dark for 2 h to reach the adsorption–desorption equilibrium. Samples were taken at different time intervals to evaluate the progress of the degradation of 4-NP. Each sample was filtered by a 0.2 μm filter and the residual 4-NP concentration was measured by the UV–vis spectrophotometer. Total organic carbon (TOC) analyses were performed using a Shimadzu 5000A TOC analyser; each sample was analyzed three times and the average values were reported.

3. Results

3.1. Structural and morphological characterization

Preliminary tests were carried out to find the calcination temperature that ensured the best compromise between crystallinity and surface area of the catalysts. Fig. 1 shows X-ray diffraction patterns of the $\text{ZnSn}_{0.25}$ sample calcined at 500, 600 and 700 °C. The peaks were found to be relative to ZnO with hexagonal wurtzite structure (JCPDS card number: 36-1451) and SnO_2 with tetragonal structure (JCPDS card number: 41-1445) and no other phases were

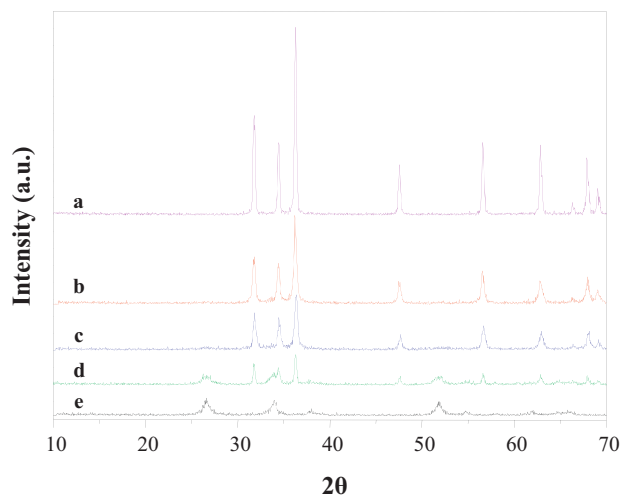


Fig. 2. XRD patterns of (a) pure ZnO, (b) $\text{ZnSn}_{0.02}$, (c) $\text{ZnSn}_{0.05}$, (d) $\text{ZnSn}_{0.25}$ and (e) pure SnO_2 , calcined at 600 °C for 2 h.

detected. ZnO was well crystallized already at 600 °C whereas SnO_2 was almost amorphous at 600 °C and its crystallinity increased with the calcination temperature. The average grain sizes of the various samples were determined by using the diffraction peaks (1 0 1) for ZnO and (1 1 0) for SnO_2 . As shown in Table 1, the main grain sizes of ZnO and SnO_2 increased with the calcination temperature and, correspondingly, the specific surface area of $\text{ZnSn}_{0.25}$ decreased.

Fig. 2 shows the diffractograms of pure ZnO, pure SnO_2 and various ZnO– SnO_2 samples calcined at 600 °C. The crystallinity degree of ZnO decreased with increasing the amount of SnO_2 in agreement with previous results obtained for coupled ZnO/ SnO_2 photocatalysts prepared by co-precipitation [26,27,44]. The peaks related to SnO_2 were detected in $\text{ZnSn}_{0.25}$ but were very small for $\text{ZnSn}_{0.05}$ and not appreciable for $\text{ZnSn}_{0.02}$. This can be due both to the small amount of SnO_2 present which remained below the detection limit of the instrument and/or to the very low crystallinity of SnO_2 which appeared almost amorphous after calcination at 600 °C. It is worth noting that all the samples obtained were more pure and crystalline than those synthesized by the co-precipitation method [47].

Mixing two different oxides generally limits the growth of both. As shown in Table 1, at the same calcination temperature the mean sizes of ZnO and SnO_2 were larger in the single oxides than in the mixed ZnO– SnO_2 samples. The addition of SnO_2 caused a reduction of the main crystallite size of ZnO. This finding is consistent with literature results reporting that the inhibition of the growth of ZnO increased with increasing the SnO_2 content [26,27,47]. The specific surface areas of the mixed samples were higher than that of ZnO, lower than that of SnO_2 and increased with the Sn content (see Table 1).

SEM observations revealed that all the samples consisted of aggregates of particles whose average sizes appeared quite close to those calculated from the XRD patterns. As shown in Fig. 3a, the particles of pure ZnO had a hexagonal shape whereas the particles of SnO_2 (Fig. 3b) showed a sphere-like morphology. The micrographs of the mixed ZnO– SnO_2 samples revealed the presence of both morphologies. Fig. 3c and d show the representative SEM images of $\text{ZnSn}_{0.05}$ and $\text{ZnSn}_{0.25}$, respectively. The small crystallites of the different phases were interwoven with each other forming tightly bound nanoclusters.

The surface structure of the mixed ZnO– SnO_2 samples was investigated by XPS. The survey spectrum revealed the presence of O, Sn and Zn. Fig. 4 shows the high resolution spectra of the Zn and Sn species, respectively. The peak centered at 1021.6 eV (Fig. 4a) was attributed to $\text{Zn } 2p_{3/2}$ of Zn(II) [51]. The peaks located at 487.1

Table 1
Physical characteristics of pure ZnO, pure SnO₂ and ZnO–SnO₂ nanocomposites.

	ZnO (600 °C/2 h)	SnO ₂ (600 °C/2 h)	ZnSn _{0.02} (600 °C/2 h)	ZnSn _{0.05} (600 °C/2 h)	ZnSn _{0.25} (500 °C/2 h)	ZnSn _{0.25} (600 °C/2 h)	ZnSn _{0.25} (700 °C/2 h)
Mean size of ZnO (nm)	34.8	–	26.1	17.5	13.5	15.0	29.0
Mean size of SnO ₂ (nm)	–	10.2	–	–	6.8	7.5	34.0
SSA (m ² /g)	4.2	26.9	10.0	22.4	24.4	22.8	19.8
Band gap (eV)	3.20	3.55	3.15	3.10	3.19	3.18	3.19

and 495.5 eV (Fig. 4b) were assigned to Sn 3d_{5/2} and Sn 3d_{3/2} of Sn(IV) [52]. These results confirmed that the oxidation states of Zn and Sn were +2 and +4, respectively, in agreement with the XRD results.

Fig. 5 shows the results of TEM observations of the ZnSn_{0.05} sample. Energy dispersive spectroscopy (EDS) analysis confirmed that the sample consisted of Zn, Sn and O (copper came from the grid of the sample holder) which is consistent with the XRD and XPS results (Fig. 5a). According to the TEM image of Fig. 5b, the sample was composed of nanoparticles with sizes in the range of about 30–80 nm and each nanoparticle was attached to several other nanoparticles. The high-resolution image of Fig. 5c evidenced the presence of intergrown lattices of ZnO and SnO₂.

3.2. DRS spectroscopy

Fig. 6A shows the UV–vis absorption spectra of the various samples calcined at 600 °C. ZnO presented the highest absorbance values in the UV region whilst SnO₂ revealed a low intensity absorbance due probably to its poor crystallinity. The absorbance of the mixed samples progressively decreased with increasing the content of SnO₂.

The band gap values (E_g) of the photocatalysts were estimated by extrapolation of the linear part of the plots of $(\alpha h\nu)^2$ vs the

energy of the exciting light (see Fig. 6B) assuming that all the samples were direct crystalline semiconductors [53]. As shown in Table 1, the E_g values of ZnO and SnO₂ were 3.20 and 3.55 eV, respectively. The band gap values of the samples containing ZnO and SnO₂ ranged between 3.10 and 3.19 eV and were lower than those of both pure oxides independently on the Zn/Sn molar ratio investigated. In any case the band gap energy was very close to that of bare ZnO being the amount of SnO₂ relatively low.

The influence of the temperature on the band gap energy was negligible being the values obtained for the sample ZnSn_{0.25} at 500, 600 and 700 °C almost the same.

3.3. Photovoltage and photoluminescence measurements

The values of the flat-band potentials of the ZnO and SnO₂ samples were determined by the slurry method proposed by Roy et al. [54], measuring the variation of the photovoltage with the pH of suspensions of the powders in the presence of an electron acceptor. Fig. 7 shows the photovoltage vs pH curves obtained by irradiation of ZnO and SnO₂ suspensions in the presence of methyl viologen dichloride. It was not possible to adjust the pH below 5 (to measure the photovoltage for pH lower than 5) because ZnO dissolves under these conditions.

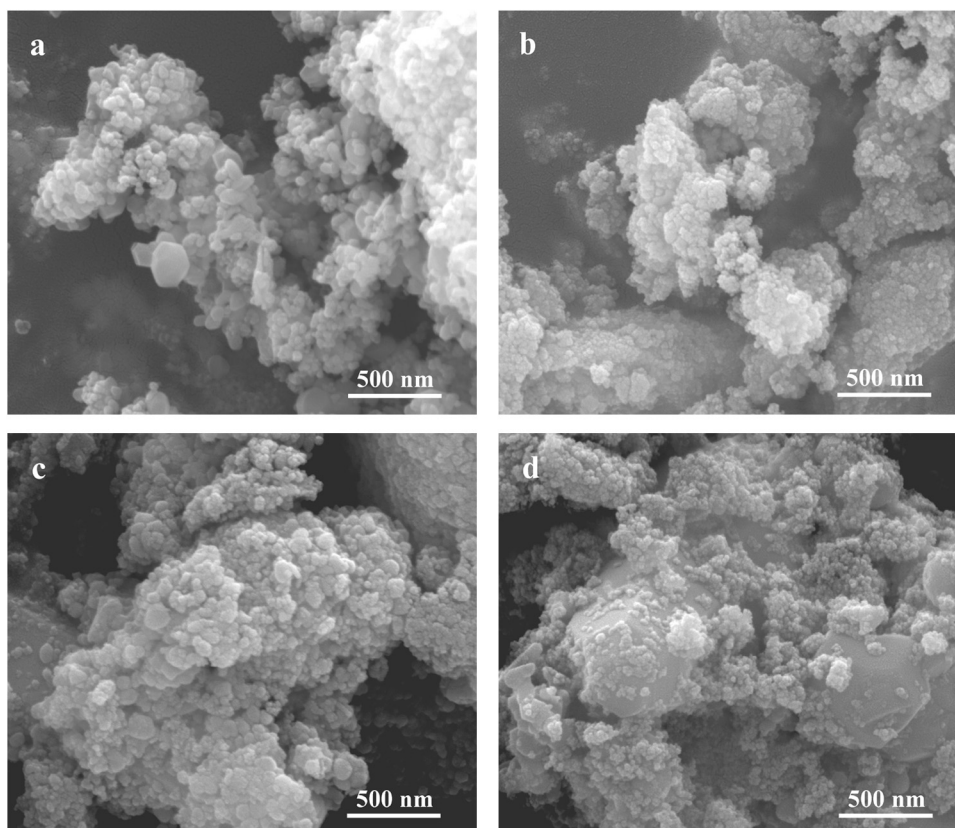


Fig. 3. SEM micrographs of (a) ZnO, (b) SnO₂, (c) ZnSn_{0.05}, (d) ZnSn_{0.25} calcined at 600 °C for 2 h.

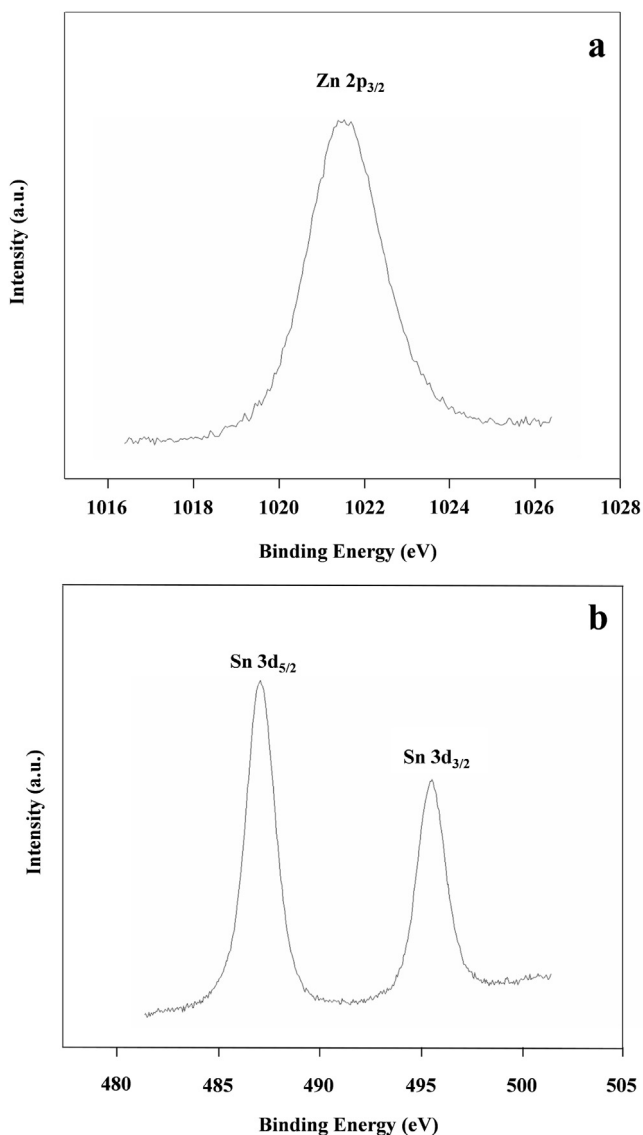


Fig. 4. XPS spectra of the ZnSn_{0.25} sample: (a) Zn 2p region; (b) Sn 3d region.

The pH value of the inflection point (pH_0) of the obtained sigmoidal titration curves allows to calculate the flat band potential at pH 7 by the equation:

$$E_{FB}(pH = 7) = E_{MV^{2+}/MV^{+\bullet}}^0 + 0.059(pH_0 - 7) \quad (2)$$

where $E_{MV^{2+}/MV^{+\bullet}}^0$ is the standard potential of the redox couple $MV^{2+}/MV^{+\bullet}$ equal to -0.445 V vs (NHE) [55]. The values obtained were -0.36 and -0.11 V for ZnO and SnO₂, respectively. These values are in good accordance with the flat band potentials of -0.48 V for ZnO and -0.21 V for SnO₂, reported by Butler et al. [50].

Fig. 8 shows the photoluminescence (PL) spectra of bare ZnO and ZnSn_{0.05}. The PL spectrum of ZnO was characterized by a narrow ultraviolet peak at 380 nm. Coupling SnO₂ with ZnO led to a decrease of PL intensity. Similar results were obtained by Asokan et al. [56] who synthesized nanocomposite ZnO–SnO₂ nanofibers by an electrospinning method.

Photoluminescence originates from the radiative recombination of photogenerated electron–hole pairs [57]. The different position of the respective conduction bands of ZnO and SnO₂ allows the photogenerated electrons migrate from the conduction band of ZnO

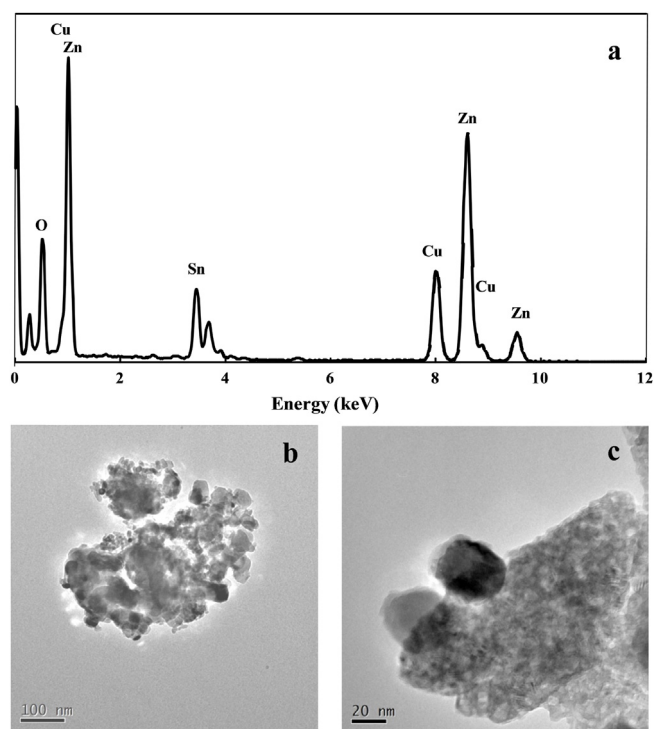


Fig. 5. (a) EDS spectrum and (b, c) TEM images of ZnSn_{0.05}.

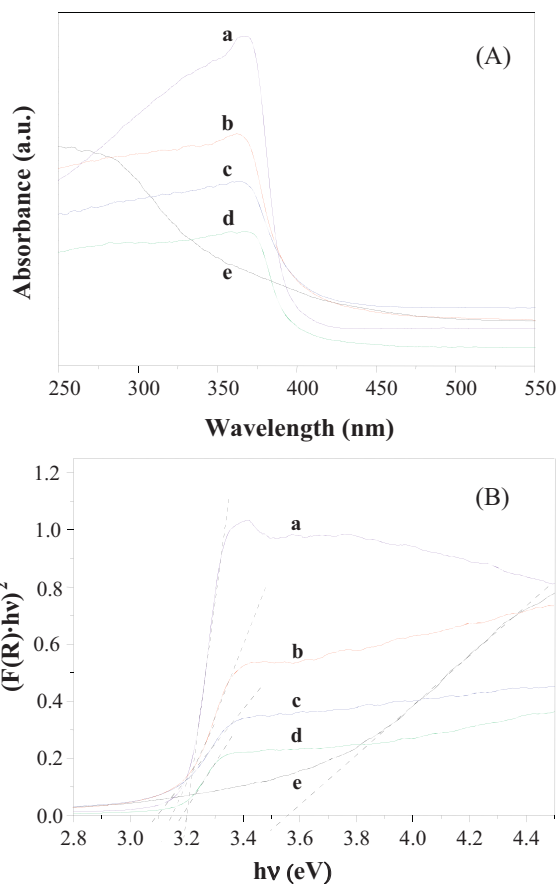


Fig. 6. UV–vis absorption spectra (A) and plots of $(F(R)hv)^2$ vs hv (B) of: (a) pure ZnO, (b) ZnSn_{0.02}, (c) ZnSn_{0.05}, (d) ZnSn_{0.25} and (e) pure SnO₂.

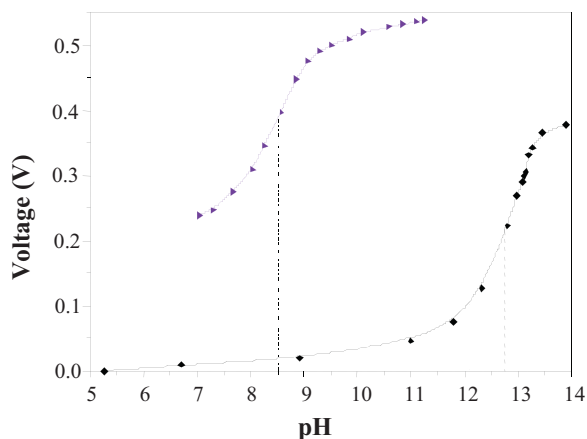


Fig. 7. Effect of pH on the photovoltage developed by irradiation of (▲) pure ZnO, and (◆) pure SnO₂ calcined at 600 °C for 2 h.

to that of SnO₂, reducing the recombination of the photoinduced electrons on the ZnO surface and weakening the PL signal.

3.4. Photocatalytic activity

The photocatalytic activity of the powders was investigated by following the degradation of 4-NP under UV light. Fig. 9 shows the kinetics of photodegradation of 4-NP in the presence of the various samples calcined at 600 °C for 2 h. The reaction rate constant, k , was calculated from the initial slope of the concentration vs time profiles. The k values are reported in Table 2.

The photocatalytic activity of SnO₂ was very low compared to that of ZnO but all the ZnO–SnO₂ samples were more active than the two pure oxides showing that the contemporaneous presence of two phases resulted in a synergistic effect. In particular, ZnSn_{0.05} revealed the highest photocatalytic efficiency.

Fig. 10 shows the effect of the calcination temperature on the photocatalytic activity of the ZnSn_{0.25} sample. The best efficiency was obtained when the sample was calcined at 600 °C whereas the powders calcined at 500 and 700 °C were less active. This result is mainly due to two opposite effects: growth of the size of the

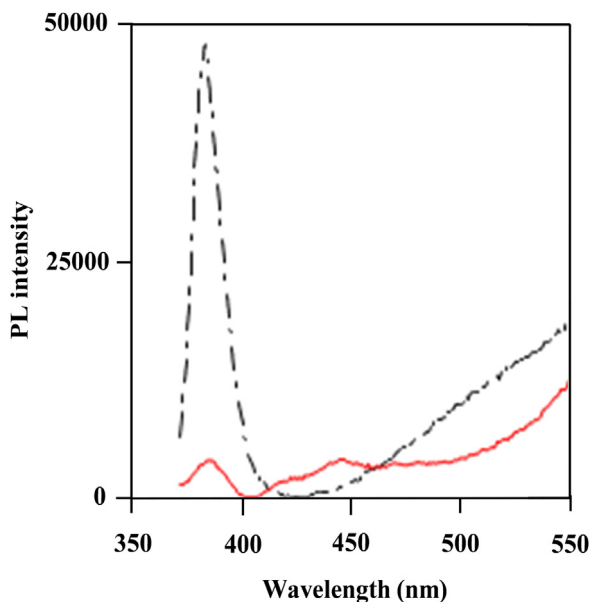


Fig. 8. Photoluminescence spectra of pure ZnO (dash-dotted line) and ZnSn_{0.05} (solid line).

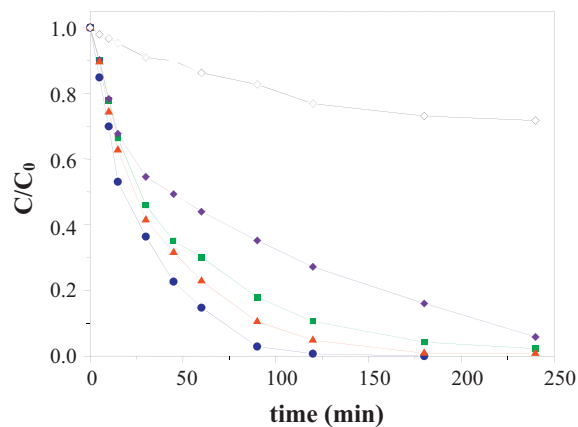


Fig. 9. Degradation kinetics of 4-NP over: (◇) SnO₂, (◆) ZnO, (▲) ZnSn_{0.02}, (●) ZnSn_{0.05} and (■) ZnSn_{0.25} calcined at 600 °C for 2 h.

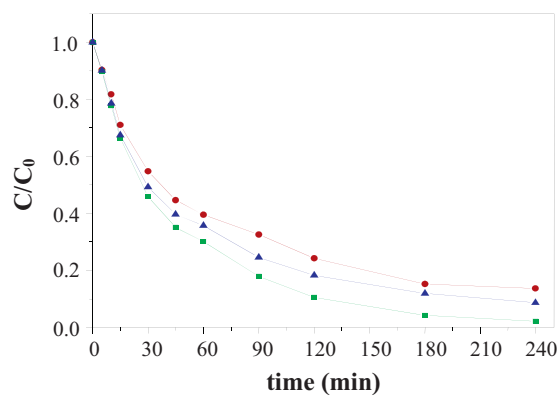


Fig. 10. Degradation kinetics of 4-NP over ZnSn_{0.25} calcined for 2 h at: (●) 500, (■) 600 and (▲) 700 °C.

particles leading to a low surface area and increase of the crystallinity with the temperature. The sample calcined at 500 °C has the highest SSA (24.4 m²/g) but it is the least crystalline. Instead, the sample calcined at 700 °C is the most crystalline but it has the lowest specific surface area (19.8 m²/g).

Fig. 11 shows the total organic carbon elimination kinetics in the presence of ZnSn_{0.05}. It is worth noting that about 40% of TOC remained when 4-NP totally disappeared. After 240 min, more than 20% of organic carbon was still present in the system due to the formation of long-lived by-products [58,59]. Anyway, the complete mineralization of the substrate was reached after a longer time.

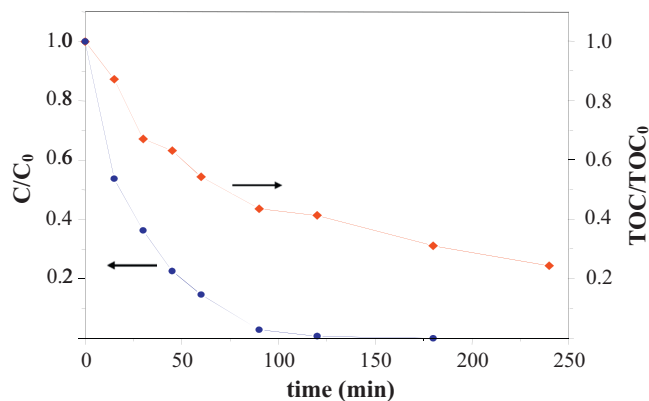


Fig. 11. Degradation (●) and total organic carbon elimination (◆) kinetics of 4-NP over ZnSn_{0.05} calcined for 2 h at 600 °C.

Table 2
Photodegradation rate constants of 4-NP on pure ZnO, pure SnO₂ and ZnO–SnO₂ photocatalysts.

	ZnO (600 °C/2 h)	SnO ₂ (600 °C/2 h)	ZnSn _{0.02} (600 °C/2 h)	ZnSn _{0.05} (600 °C/2 h)	ZnSn _{0.25} (500 °C/2 h)	ZnSn _{0.25} (600 °C/2 h)	ZnSn _{0.25} (700 °C/2 h)
$k \times 10^3$ (min ⁻¹)	25.2	3.3	30.2	40.0	22.0	26.4	25.2

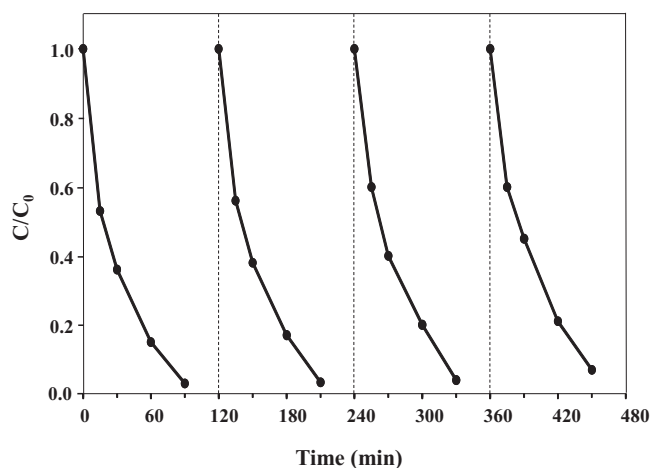


Fig. 12. Reusability tests of ZnSn_{0.05}.

Fig. 12 shows the results obtained after reuse of ZnSn_{0.05}. After each photocatalytic run, the powder was recuperated, washed with distilled water and a fresh solution of 4-NP was added. Although the degradation efficiency slightly decreased after each run, the ZnO–SnO₂ catalyst remained effective and reusable under UV light irradiation.

Fig. 13 shows a comparison between the kinetics of degradation of 4-NP in the presence of ZnSn_{0.05} samples synthesized by two different methods. The ZnO–SnO₂ photocatalyst prepared by the sol-gel method revealed a photocatalytic activity higher than that of the sample prepared by the conventional co-precipitation method [47].

4. Discussion

The high photocatalytic activity of the ZnO–SnO₂ nanocomposites with respect to the pure oxides can be explained by the contemporaneous presence of two semiconductors possessing different energy levels for their corresponding conduction and valence

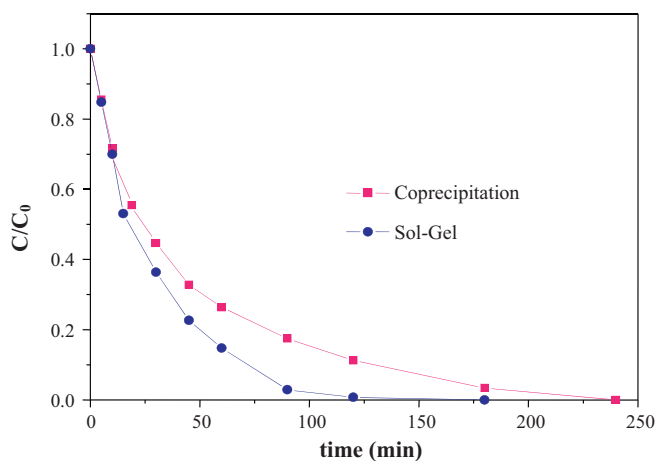


Fig. 13. Degradation kinetics of 4-NP over ZnSn_{0.05} samples calcined for 2 h at 600 °C, synthesized by different methods.

bands. Depending on the potentials of the photogenerated holes and electrons, it is possible a vectorial transfer of charge carriers from a semiconductor to another, leading to more efficient electron–hole separation and higher photocatalytic activity [48].

Both ZnO and SnO₂ are n-type semiconductors so that, assuming that the difference between flat band potential and conduction band edge is negligible, it is possible to locate the valence band edge of ZnO and SnO₂ by adding the band gap energy to the flat-band potential value. The results have shown that the valence band of ZnO ($E_{VB} = 2.84$ eV) is lower than that of SnO₂ ($E_{VB} = 3.44$ eV). Similarly, the conduction band of ZnO ($E_{CB} = -0.36$ eV) is lower than that of SnO₂ ($E_{CB} = -0.11$ eV). Fig. 14 shows a schematic diagram of the energy band structures of ZnO and SnO₂ at pH = 7. Although these data may not be the exact absolute values of the conduction and valence band potentials of ZnO and SnO₂, they should offer a correct estimation of the relative band edge positions of the two semiconductors.

When the ZnO–SnO₂ nanocomposites are irradiated by UV light, electrons can transfer from the more cathodic conduction band of ZnO to the more anodic conduction band of SnO₂. Analogously, holes transfer can occur from the valence band of SnO₂ to the valence band of ZnO. The efficient charge separation can increase the lifetime of the charge carriers and enhance the efficiency of the interfacial charge transfer to adsorbed substrates. The inhibition of the electron–hole recombination is confirmed by the PL spectra of the mixed ZnO–SnO₂ samples that revealed a weakening of the PL intensity with respect to that of pure ZnO. A mechanistic scheme of the charge separation in the ZnO–SnO₂ system is shown in Fig. 14.

The enhanced photoactivity of the ZnO–SnO₂ nanocomposites is so attributable to the formation of local heterojunctions between

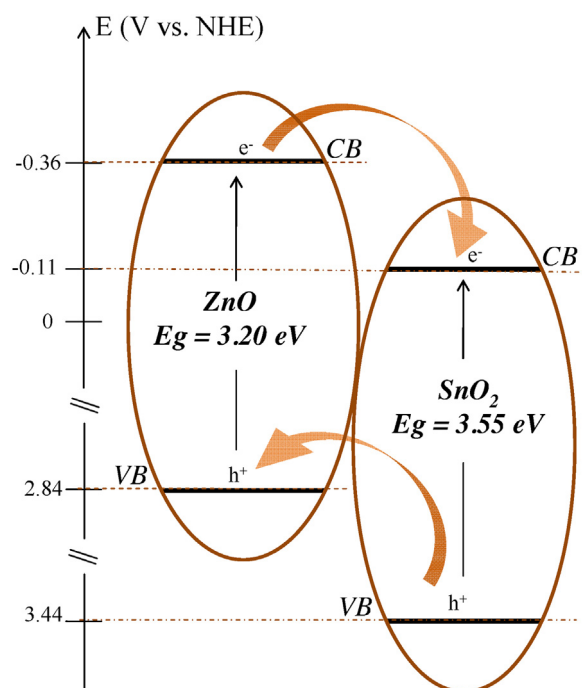


Fig. 14. Schematic diagram representing the charge-transfer processes between coupled ZnO–SnO₂ particles.

ZnO and SnO₂ particles which facilitate the separation of the photo-generated e⁻/h⁺ pairs. The photocatalytic efficiency is related to the number of different particles attached to each other that depends on the ZnO/SnO₂ molar ratio. As shown by the XRD and TEM results, the SnO₂ particles present in the ZnSn_{0.02} and ZnSn_{0.05} samples are probably very small and finely dispersed on the surface of the ZnO particles so that a large number of heterojunctions may exist: this justifies the enhanced photoactivity of the coupled ZnO–SnO₂ samples with respect to pure ZnO. With increasing the content of SnO₂ the photoactivity decreases due to the presence of large crystals of SnO₂ that reduces the number of heterojunctions and/or to the increase of free SnO₂ particles that have poor photocatalytic activity. Therefore, there is an optimum SnO₂ content that leads to a maximum photocatalytic efficiency.

A comparison of the data reported in Tables 1 and 2 shows that there is a correspondence between the E_g values of the ZnO–SnO₂ nanocomposites and the photocatalytic results. In particular, the photodegradation rate constants of 4-NP increased with decreasing the band gap values of the mixed samples. In agreement with previous results obtained with various coupled semiconductors [40,60–63], ZnSn_{0.05}, which presented the lowest band gap value, was the most active sample.

The experimental results have shown that the coupled ZnO–SnO₂ samples synthesized with the sol–gel method were more crystalline and with a mean particle size smaller than that of the samples obtained by co-precipitation at the same calcination temperature [47]. Besides, the particles synthesized with the sol–gel method revealed a sphere-like morphology whereas the co-precipitated particles had a plate-like form [47]. The highest photocatalytic activity of the sol–gel prepared samples can be so ascribed not only to the enhanced crystallinity but also to the smaller size and the more regular form of the single particles that allow the formation of a larger amount of heterojunctions. A similar effect of the differences of particle size was reported by Zhang et al. [29] who compared the photoefficiency of coupled ZnO–SnO₂ samples prepared by co-precipitation using NH₃·H₂O as the precipitant with samples obtained using NaOH.

5. Conclusion

ZnO–SnO₂ nanocomposite photocatalysts have been synthesized with a facile sol–gel route. The photoefficiency of the ZnO–SnO₂ samples was superior to that of pure ZnO or SnO₂. The photocatalytic activity was found to be dependent on the calcination temperature and on the ZnO/SnO₂ molar ratio. ZnSn_{0.05} revealed the highest efficiency for the degradation of 4-NP. The enhanced photoactivity was attributed to the charge separation of the photoproduced electron/hole pairs arising from the coupling of ZnO with SnO₂. The small size of the particles and the intimate contact with each other are crucial factors to increase the efficiency of the samples.

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