

Formulation of a truss element for modelling the tensile response of FRCM strips

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Abstract

Modelling the tensile behaviour of Fabric Reinforced Cementitious Matrix (FRCM) is not a straightforward task due to the inner complexity of the mechanics of this kind of composite materials. In fact, after that the matrix is cracked, the compatibility between the fiber and the surrounding mortar is lost and the system behaves as two separate elements connected by a brittle interface. For this reason, several research studies proposed computational approaches for evaluating the tensile behaviour of FRCM composites, usually referring to brick-based 3D Finite Element Models (FEM) or to complex numerical procedures. This paper shows the formulation of a simplified coupled truss element for modelling the tensile behaviour of FRCM composites. The proposed element includes the interface slip between fiber and matrix and the brittle failure of the fabric and allows to describe the response of the system between the cracks in closed form. The proposed element is also adopted for assessing the tensile constitutive response of FRCMs through a simplified assembly procedure, proving to be reliable and computationally efficient.

Keywords: FRCM, tensile behaviour, cracking, interface

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1. Introduction

The use of Fabric Reinforced Cementitious Matrix (FRCM) composites has become increasingly popular for applications in masonry structures in the last ten years [1]. Its popularity depends especially from the advantages arising from their applications when applied on masonry surfaces of historical buildings, especially due to the good compatibility between the inorganic matrix and the masonry substrate. For this reason, a huge amount of research works was recently carried out for characterizing the constitutive behaviour of these materials, with particular reference to the tensile behaviour and the bond with masonry [2][3].

The tensile behaviour of FRCM composites proved to be complex and affected from the interaction between matrix and fabric. In fact, after that matrix is cracked, the compatibility between fiber and mortar is lost and the fabric tends to slip with respect to the surrounding mortar. Failure can be reached due to the tensile breakage of the fiber yarns or due to the de-bonding at the the fabric-matrix interface.

The experimental studies available in the recent literature and concerning the mechanical characterization of FRCM composites highlighted a wide variability of results due to the great number of variables involved, including the nature of the fiber, matrix grade, test set-up, the treatment of the fabric surface. As a consequence, a huge amount of research work has been carried out in the last years, investigating on the effect of the different key parameters. Numerous studies were performed on the kind of fibre, such as basalt [4] [5] [6] [7] [8], carbon [9] [10] [11], glass [9] [12] and Polybenzoxazole (PBO) [13] [14]. Other studies investigated the role of the test set-up, analyzing the role of the boundary loading conditions, such as the clevis or clamping grip [15] [16] or the capabilities of the Digital Image Correlation (DIC) for providing more detailed data on the field of strains or displacements [2] [8].

26 In this background, despite several experimental studies were performed on this topic,
27 fewer studies addressed analytical or numerical approaches for modelling the tensile be-
28 haviour of FRCC strips. Simplified analytical models were proposed in the past [4] [17],
29 aiming to model the behaviour of the composite with a multilinear shape, following the
30 well-known rules adopted for reinforced concrete members. Other works followed a nu-
31 merical approach, often based on 3D Finite Element (FE) models. In this context a FE
32 multiscale approach was proposed by Bertolesi et al. [13] for modelling the bond be-
33 haviour with the substrate, while Monaco et al. [18] [19] modelled the tensile behaviour of
34 FRCC by adopting a cohesive interface between matrix and fabric. Both works adopted
35 the Concrete Damage Plasticity model for simulating the tensile behaviour of the matrix
36 and assumed a bilinear law with damage for the interface. More recently, a simplified
37 uniaxial model was proposed by Grande and Milani [20], who also proposed recently [21]
38 numerical strategies to solve the governing equations of uniaxial models of FRCC strips
39 under different loading conditions.

40 Despite these numerical approaches provide a detailed mechanical response of the
41 FRCC systems, they require a complex calibration of the model for assessing the input
42 variables, which are often unknown and should be empirically or tentatively estimated,
43 with the risk of including unavoidable uncertainties. Additionally, the 3D brick-based
44 FE simulations require a strong computational cost, including long calculation time and
45 numerical convergence issues, such as mesh sensitivity, hourglassing or locking.

46 In this context, this paper presents a novel formulation of a coupled truss element for
47 calculating the tensile response of FRCC strips. The governing equation of the element
48 is obtained in strong form by analyzing the force transmission mechanism between fabric
49 and matrix within the crack spacing. The two phases are modeled as two truss elements
50 connected by a pure shear interface, under the assumption of brittle linear behaviour of
51 materials and interface. Previous studies [22] [23] highlighted that the load transfer mech-

52 anism in FRCM composites is more complex than only shear stress transfer at the fabric to
53 matrix interface. The textile of the FRCM systems is multidirectional, generally bidirec-
54 tional, fabric and the mortar-textile interlock at the voids of fabric known to be one of the
55 stress-transfer mechanisms. However, the proposed paper aims to provide a simple model
56 for achieving the local and global response, respecting a phenomenological/empirical for-
57 mulation of the interface, which respects the experimental observation. The formulation
58 allows to achieve the local response within the crack spacing in terms of elongations, in-
59 ternal forces and shear stress at the interface. The proposed element is therefore adopted
60 for assembling a procedure for calculating the tensile response of FRCM strips. Results
61 are finally compared with experimental results available in the literature, showing good
62 agreement. The proposed procedure keeps the accuracy of a formulation capable of ob-
63 taining the local and global response of the composite material but with the advantage of
64 reducing the computational cost with respect to brick-based 3D FEM analyses. Addition-
65 ally, it requires the calibration of only one mechanical parameter i.e. the shear stiffness of
66 the interface k , which is here proposed as a function of the mechanical properties of the
67 fabric and of the number of textile layers adopted.

68 **2. Element formulation**

69 The proposed formulation is based on the study of a part of FRCM strip between two
70 successive cracks with length equal to s (Figure 1). It is assumed that the behaviour of
71 the composite between the cracks can be represented by two sub-elements, represented
72 by the matrix and the fabric, each one with truss behaviour and with axial stiffness equal
73 to $E_m A_m$ and $E_f A_f$ respectively, as shown in Figure 1. The two elements are connected
74 by a continuous shear interface, with stiffness k , for sake of simplicity the interactions
75 with transverse bundles of bidirectional textile are neglected, according to experimental

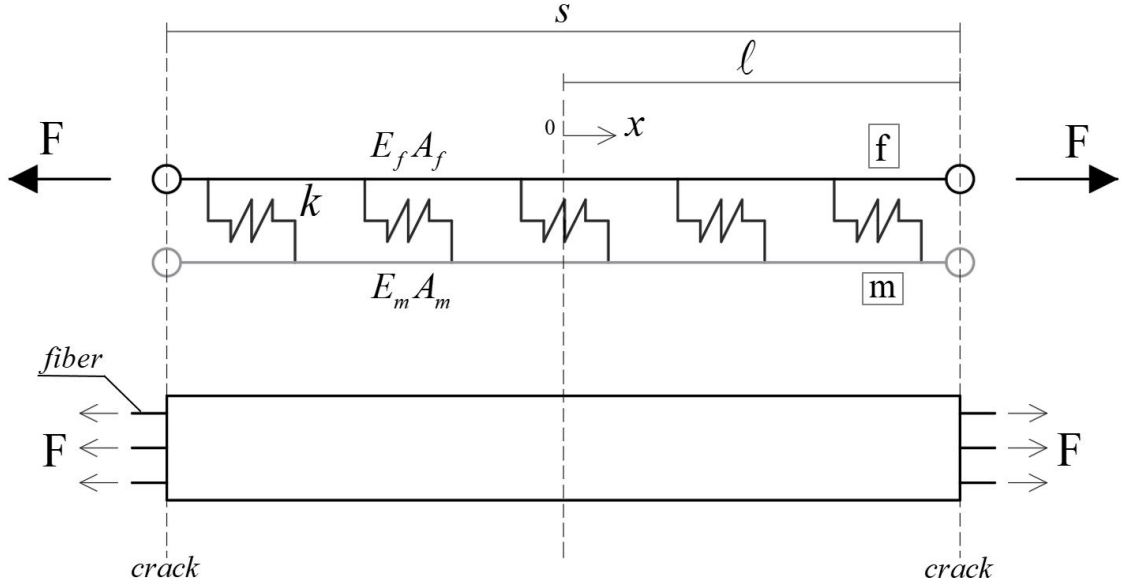


Figure 1: Coupled truss model

76 evidence [9]. All the materials are assumed to be elastic with a pure brittle constitutive
 77 law. The axial elongations of the fiber and matrix are denoted as $u_f(x)$ and $u_m(x)$.

78 Under the assumptions of Figure 2, the equilibrium equation of the mortar layer can
 79 be written as

$$\frac{d^2 u_m(x)}{dx^2} = \frac{p_x(x)}{E_m A_m} \quad (1)$$

80 being $p_x(x)$ the shear force per unit length transferred from the fiber through the interface.

81 The constitutive law of the interface is assumed to be elastic with pure brittle behaviour,
 82 which means that a linear relation can be established between the shear stress and the
 83 relative axial elongation of the elements. In this way, the shear stress per unit length is
 84 expressed as

$$p_x(x) = \tau(x)t_i = k(u_f(x) - u_m(x))t_i \quad (2)$$

85 where t_i is the depth of the interface. If Eq.2 is assumed, the equilibrium equation of the

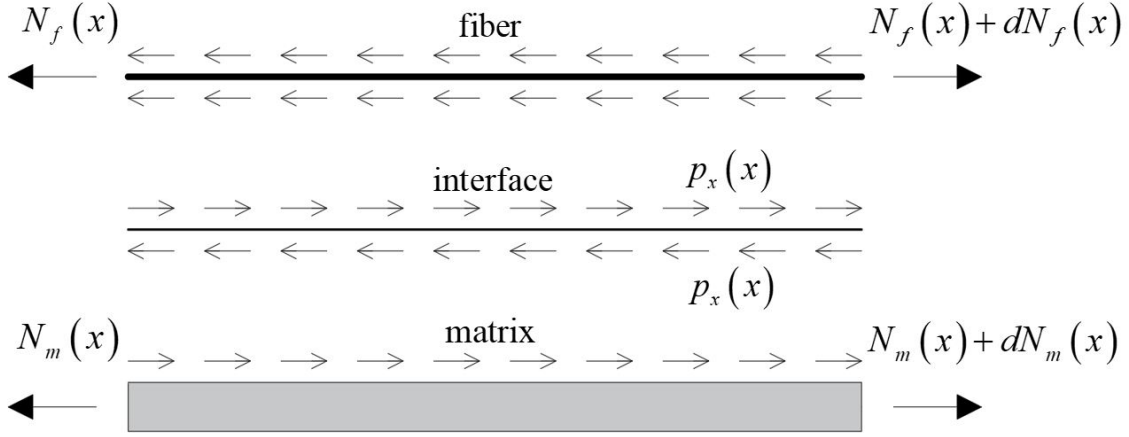


Figure 2: Internal forces and sign assumptions

86 matrix is written as

$$\frac{d^2 u_m(x)}{dx^2} = -\frac{k}{E_m A_m} (u_f(x) - u_m(x)) t_i \quad (3)$$

87 The equilibrium of the fabric can be expressed with similar considerations, leading to
88 the following differential form of the equilibrium

$$\frac{d^2 u_f(x)}{dx^2} = \frac{k}{E_f A_f} (u_f(x) - u_m(x)) t_i \quad (4)$$

Eqs.3 and 4 represent a system of two ordinary differential equations for calculating the axial response of the system in terms of $u_f(x)$ and $u_m(x)$. A convenient expression of this system can be written in the following form

$$\frac{d^2 u_m(x)}{dx^2} + \beta_m^2 (u_f(x) - u_m(x)) = 0 \quad (5a)$$

$$\frac{d^2 u_f(x)}{dx^2} - \beta_f^2 (u_f(x) - u_m(x)) = 0 \quad (5b)$$

where β_m and β_f are the two relative stiffness parameters between the mortar or the fabric

and the interface

$$\beta_m = \sqrt{\frac{kt_i}{E_m A_m}} \quad (6a)$$

$$\beta_f = \sqrt{\frac{kt_i}{E_f A_f}} \quad (6b)$$

89 and observing that from a dimensional point of view both parameters are the inverse of a
90 length.

91 The two equations of the system can be coupled together by considering the relation
92 between the functions $u_m(x)$ and $u_f(x)$ from Eq.5a

$$u_f(x) = u_m(x) - \frac{1}{\beta_m^2} \frac{d^2 u_m(x)}{dx^2} \quad (7)$$

93 and its second order derivative

$$u_f''(x) = -\frac{d^2 u_m(x)}{dx^2} - \frac{1}{\beta_m^2} \frac{d^4 u_m(x)}{dx^4} \quad (8)$$

94 If Eq.8 is introduced in Eq.5b, the system in Eqs. 5 is replaced by a single fourth order
95 differential equation

$$\frac{d^4 u_m(x)}{dx^4} + \frac{d^2 u_m(x)}{dx^2} \left(1 - \frac{\beta_f^2}{\beta_m^2}\right) = 0 \quad (9)$$

96 which represents the strong form of the equilibrium, ruling the axial behaviour of the the
97 two trusses coupled by a continuous shear interface. It is observed that Eq.9 can be solved
98 in $u_m(x)$, obtaining the field of axial elongations in the fabric from Eq.7. The solution of
99 the homogeneous equation Eq.9 belongs to the following general form

$$u_m(x) = c_1 \frac{e^{\eta x}}{\eta^2} + c_2 \frac{e^{-\eta x}}{\eta^2} + c_3 x + c_4 \quad (10)$$

100 where

$$\eta = \sqrt{\beta_f^2 + \beta_m^2} \quad (11)$$

101 It should be noted that when the interface is infinitely deformable (i.e. $k = 0$), Eqs.5
 102 represent the classic form of the equilibrium of two unloaded truss elements and Eq.10
 103 describes the linear trend of the elongations.

Boundary conditions (BC) are enforced for particularizing Eq.10 and obtaining the solution of the scheme of Figure 1. In particular, compatibility conditions (essential BC) along the symmetry axis ($x = 0$) are enforced by considering that elongations are equal to zero

$$u_m(0) = 0 \quad (12a)$$

$$u_f(0) = \frac{d^2 u_m(x)}{dx^2} \Big|_{x=0} = 0 \quad (12b)$$

Additionally, equilibrium conditions need to be imposed at the ends of the two trusses. Reminding that the axial force inside the crack ($x = l$) is carried only by the fabric, the equilibrium conditions (natural BC) are written as follow

$$N_m(l) = \frac{du_m(x)}{dx} \Big|_{x=l} = 0 \quad (13a)$$

$$N_f(l) = \frac{du_f(x)}{dx} \Big|_{x=l} \frac{kt_i}{\beta_f^2} = F \quad (13b)$$

104 being F the overall axial force inside the crack sustained by the fabric.

105 The solution achieved in this way is expressed by the following function

$$u_m(x) = F \frac{e^{-\eta x} \beta_f^2 \beta_m^2 (e^{\eta l} - e^{\eta(l+2x)}) + \eta x (e^{\eta x} + e^{\eta(2l+x)})}{(1 + 2e^{2\eta l}) kt_i \eta^3} \quad (14)$$

106 Eqs.14 and 7 represent the solution of the system. It should be noted that as expected
 107 the response of each component is linearly dependent from the value of applied force in
 108 the fabric and consequently, the functions $u_m(x)$ and $u_f(x)$ can be normalised with respect
 109 to F and adopted as general solutions for any value of force.

Finally, the trend of the axial force in the two trusses is found through the first order derivative of the axial elongations. In particular,

$$N_m(x) = \frac{kt_i}{\beta_m^2} \frac{du_m(x)}{dx} \quad (15a)$$

$$N_f(x) = \frac{kt_i}{\beta_f^2} \frac{du_f(x)}{dx} \quad (15b)$$

110 If Eq.14 and 7 are introduced in Eq.15, the following expressions hold

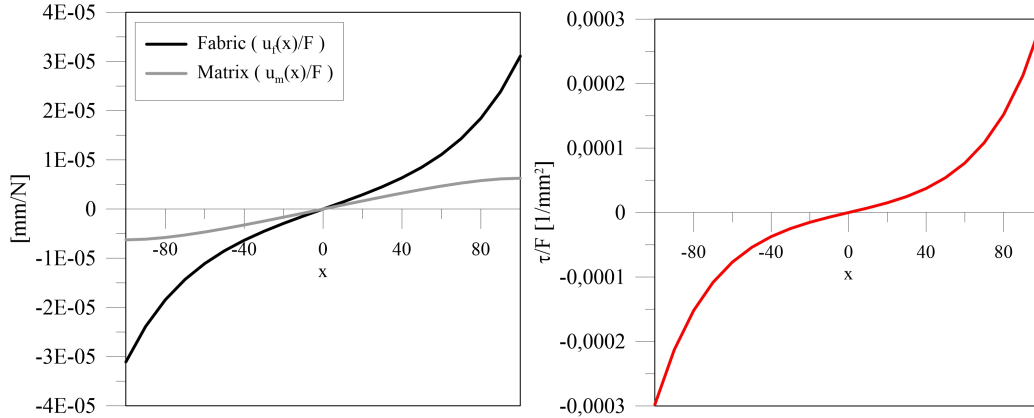
$$N_m(x) = F\beta_f^2 \frac{e^{-\eta x}(e^{\eta x} - e^{\eta l} + e^{\eta(2l+x)} - e^{\eta(2l+x)})}{\eta^2(1 + e^{2\eta l})} \quad (16a)$$

$$N_f(x) = F \frac{e^{-\eta x}(\beta_f^2 e^{\eta l} + \beta_f^2 e^{\eta(2x+l)} + \beta_m^2 e^{\eta x} + \beta_m^2 e^{\eta(2l+x)})}{\eta^2(1 + e^{2\eta l})} \quad (16b)$$

111 It is observed that the fields of the axial force $N_m(x)$ and $N_f(x)$ are only a function
 112 of the relative stiffness parameters β_f , β_m and of the half crack spacing l , being linearly
 113 dependent from the F .

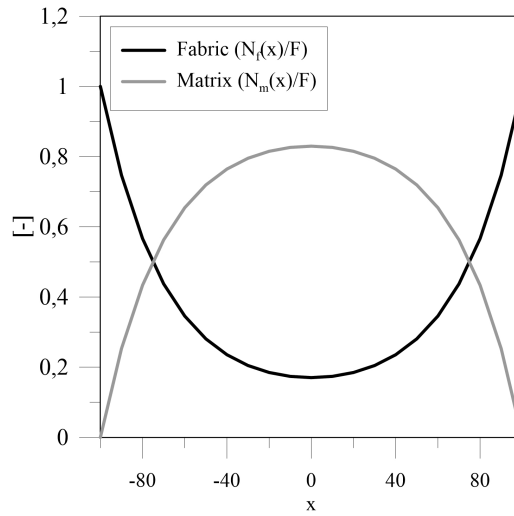
114 2.1. Numerical applications

115 Fig.3 shows the results of a numerical example, referring to an FRCM strip with $\beta_m =$
 116 0.01 , $\beta_f = 0.032$ and $s = 2l = 200mm$. Results are reported in normalised form with
 117 respect to F . The theoretical prediction of local effects in mortar and fabric layers seems
 118 to be in good agreement with experimental evidences, as can be observed in the study
 119 proposed by Saidi et Gabor [24]. It is observed as the trend of the axial elongations is
 120 more marked in the fabric with respect to the mortar. The trend of axial elongation in
 121 the fiber is almost linear in the central part of the specimen and tends to an exponential
 122 amplification near the loaded ends. Conversely, the slope of the axial elongation in the
 123 matrix assumes a constant value near the extremities of the strip. Fig.3(b) shows the trend
 124 of the normalised shear stress at the interface. It is observed that this trend reflects that



(a) Normalised axial elongations

(b) Normalised shear stress



(c) Axial force

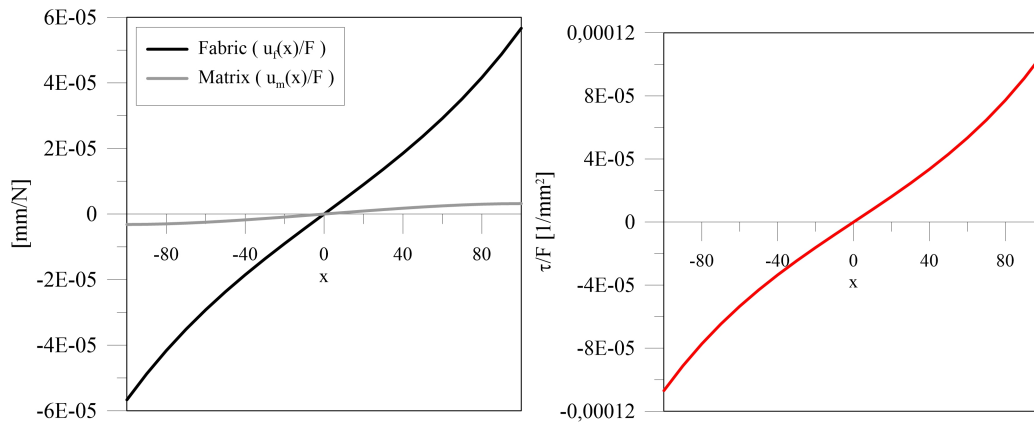
Figure 3: Numerical example for $\beta_m = 0.01$, $\beta_f = 0.032$, $s = 2l = 200\text{mm}$.

125 of the relative slip, calculated as $u_f(x) - u_m(x)$, multiplied for a magnification factor
 126 (i.e. the shear stiffness of the interface k). The shear stress tends to intensify at the ends
 127 of the element, near the cracks, due to the high value of the axial force sustained by the
 128 fiber in that zones. Finally, Fig.3(c) shows the trend of the normalised axial force in the
 129 two elements. It is observed as the axial load F entirely carried by the fabric for $x = l$

130 steeply decreases along the fiber down to its minimum value in correspondence of $x = 0$.
131 Conversely, the axial force in the matrix $N_m(x)$ tends to increase from the crack $x = l$ to
132 the middle section $x = 0$. It is observed that in this section the matrix carries a value of
133 axial load equal to about the 40% of F . This trend of the axial force shows that the two
134 elements interact through the interface and the transition of the axial stress from the fiber
135 to the matrix is related to the shear stiffness of the interface k .

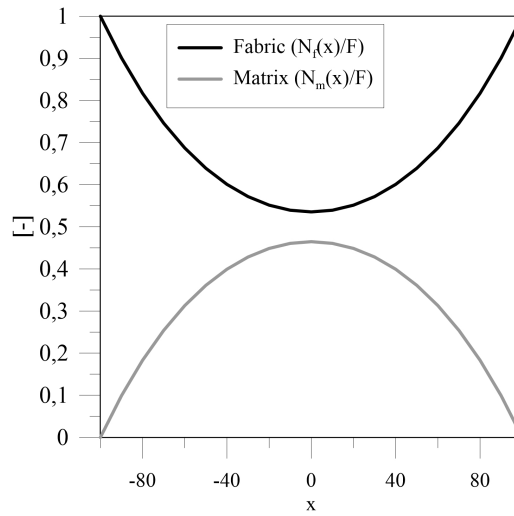
136 This mechanism is clear by comparing the results of Fig.3 with those reported in Fig.4.
137 This last refers to an element with $\beta_m = 0.0045$, $\beta_f = 0.013$ and the same crack spacing of
138 the previous example. It is observed that these lower values of β_m and β_f are obtained by
139 adopting lower values of interface stiffness k . Results confirm that the contribution of the
140 mortar becomes less evident for a more deformable interface (Fig.4(a)). The elongations in
141 the matrix assume low values with respect to those in the fabric, and the trend of these last
142 tends to be linear for lower value of k . As a consequence, the shear stresses are reduced
143 with respect to the previous example, due to the increased deformability of the interface
144 (Fig.4(b)). The trend of the shear stress almost reproduces that of the displacements in the
145 fabric, and the interface is more uniformly stressed along its length with lower value of
146 stresses. This fact is reflected in terms of axial force, as shown in Fig.4(c). As expected,
147 the contribution of the matrix is less evident in this case. The values of axial force in the
148 mortar are lower with respect to the previous case, due to the lower ability of the interface
149 in transferring the stress from the loaded element to the matrix.

150 The effect of the interface stiffness on the force transmission of the element can be
151 observed in Fig.5, which shows the normalised interface shear stress and the axial force in
152 the fabric for different values of k , from 2 to 12 N/mm^3 . It is worth noting that the shear
153 stress keeps a similar trend in the central zone of the strip for the different values of k .
154 When k increases, the slope of the function tends to rise suddenly and the response proves
155 to be less sensitive to the value of the interface stiffness, as shown from the curves which



(a) Normalised axial elongations

(b) Normalised shear stress



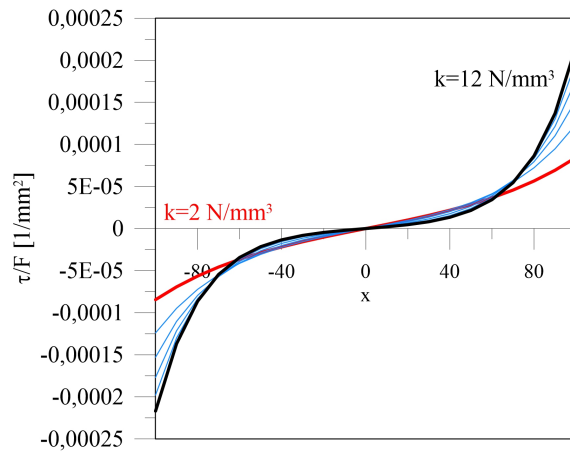
(c) Axial force

Figure 4: Numerical example for $\beta_m = 0.0045$, $\beta_f = 0.013$, $s = 2l = 200mm$.

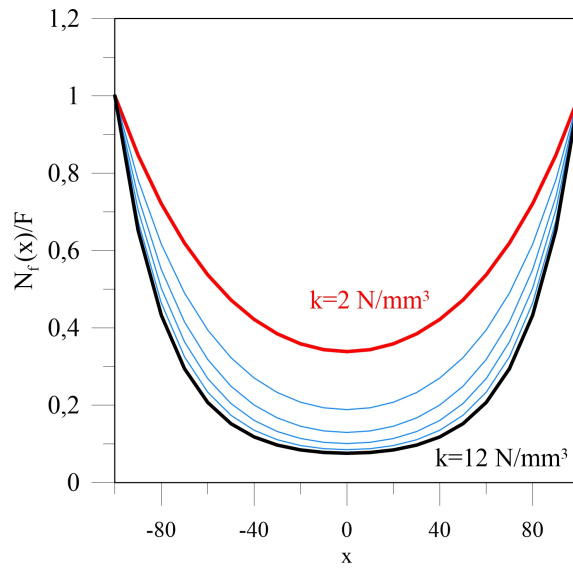
156 tend to be closer.

157 3. Model implementation

158 The proposed formulation of the coupled truss element can be exploited for calculating
 159 the constitutive tensile response of FRCM strips.



(a) Normalised shear stress



(b) Normalised axial force in the fabric

Figure 5: Parametric example for different values of k .

160 On the basis of the results presented in the previous section, it is possible to calculate
 161 the tensile load F_{cr} corresponding to the formation of a crack in correspondence of $x = 0$.
 162 In particular, the maximum axial stress in the matrix $N_m(0)/A_m$ can be imposed to be

163 equal to the tensile strength of the mortar f_{mt} . The following relation holds

$$N_m(0) = f_{mt} t_i \frac{k}{E_m \beta_m^2} = \epsilon_{mt} \frac{k t_i}{\beta_m^2} = F_{mt} \quad (17)$$

164 being ϵ_{mt} the ultimate tensile strain of the matrix, and labelling F_{mt} the axial tensile ca-
165 pacity of the matrix within the strip.

166 If $N_m(0)$ is evaluated through Eq.16a for $x = 0$, Eq.17 can be considered an equa-
167 tion with a single unknown variable F_{cr} , which solution is represented by the following
168 expression:

$$\frac{F_{cr}}{F_{mt}} = \frac{(e^{2\eta l} + 1)\eta^2}{(e^{\eta l} - 1)^2 \beta_f^2} \quad (18)$$

169 Finally, the stiffness of the system is obtained by dividing the value of F_{cr} for the value
170 of $u_f(x)$ calculated for $x = l$ and for $F = F_{cr}$.

$$\frac{K_{cr}}{E_f A_f} = \frac{\eta^3 \text{Cosh}(\eta l)}{\eta l \beta_m^2 \text{Cosh}(\eta l) + \beta_f^4 \text{Sinh}(\eta l)} \quad (19)$$

171 Eq.19 represents the normalised axial stiffness of the coupled truss element considering
172 the interaction between matrix and fabric. It is observed that both normalised expressions
173 18 and 19 depend only by the relative stiffness parameters β_f and β_m and from the half
174 crack spacing l .

175 On the basis of the previous expressions, an easy procedure can be followed to calcu-
176 late the axial force vs. displacement response of a FRCM strip. The procedure is based on
177 the progressive assumption of the crack spacing s and number of cracks n_c , hypothesizing
178 that the strip can be modeled by n_e series trusses (Fig.6).

179 The procedure can be resumed with the following points:

- 180 • the crack spacing at the first step s^I is assumed to be equal to the length of the strip
181 L -i.e. two cracks are formed at the ends ($n_c^I = 2$)-;

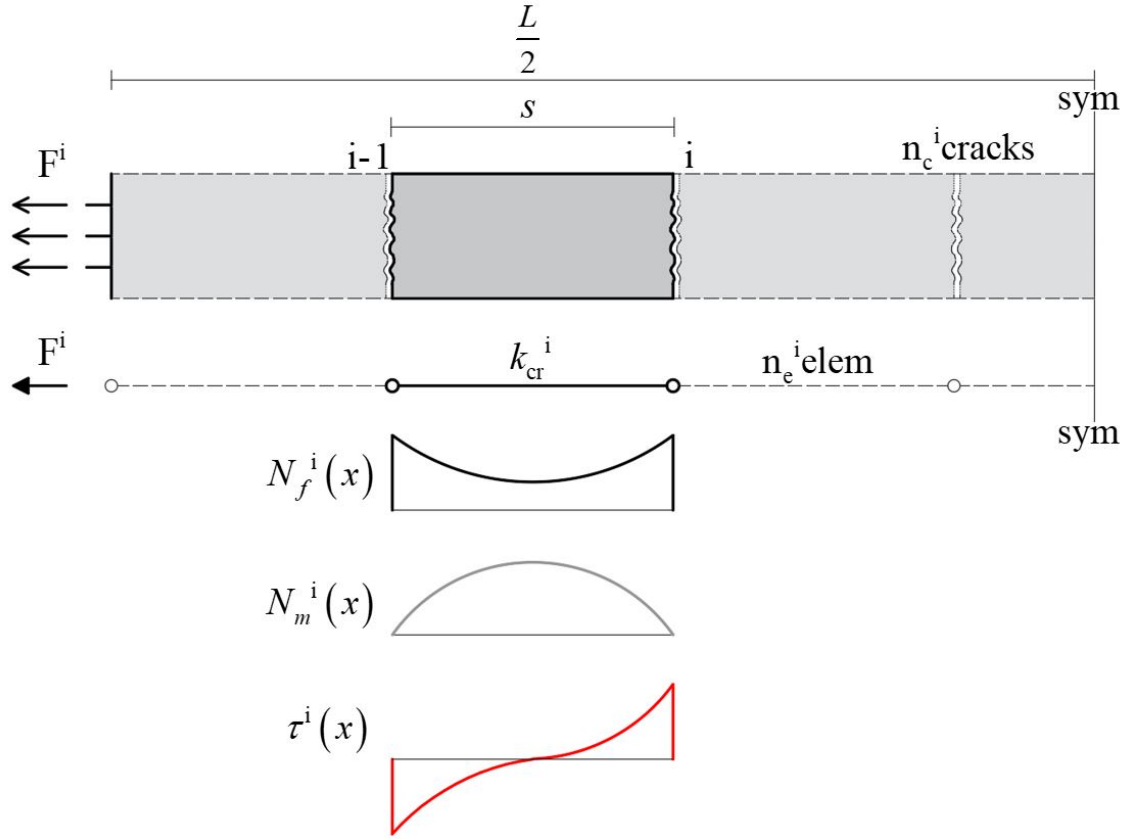


Figure 6: Series truss model

- 182 • the system is therefore made by $n_e^I = n_c^I - 1 = 1$ truss with stiffness equal to
- 183 $K^I = K_{cr}$ evaluated through Eq.19;
- 184 • the force carried by the system F^I is calculated through Eq.18, assuming that a new
- 185 crack is opened for $x = 0$;
- 186 • the elongation is therefore easily calculated $u^I = F^I/K^I$;
- 187 • the procedure is repeated from the first point, and for the general i -th step the fol-
- 188 lowing expressions can be considered:

$$n_e^i = 2^{(i-1)} \quad (20a)$$

$$n_c^i = n_e^i + 1 \quad (20b)$$

$$s^i = L/n_e^i \quad (20c)$$

$$\frac{1}{K^i} = \frac{n_e^i}{K_{cr}} \quad (20d)$$

- 189 • the procedure is interrupted when the force F^i exceeds the tensile strength of the
 190 textile or when the shear stress τ^i is greater than the maximum value assumed at the
 191 interface.

192 The use of Eq.20 together with the generalized expressions of the force Eq.18 and
 193 stiffness Eq.19 allows to evaluate the post-cracking tensile response of FRCM strips and
 194 to assess the number of cracks at failure.

195 Fig.7 shows two examples of application of the proposed model for a strip length
 196 of $L = 200$ mm, considering the two cases mentioned above of stiff (Fig.7(a)) and de-
 197 formable interface (Fig.7(b)). The response is reported in terms of normalised axial force
 198 F/F_{mt} as per Eq.18, while the axial elongation is normalised with respect to the ultimate
 199 displacement of the mortar $\epsilon_{mt}L$. This kind of representation is due to the fact that it de-
 200 pends only by the parameters β_f , β_m and by the length of the strip L . The first point of
 201 the curve corresponding to the formation of the first crack is calculated by the well known
 202 expressions

$$F_{1stcr} = f_{mt} \left(A_m + \frac{E_f}{E_m} A_f \right) \quad (21a)$$

$$u_{1stcr} = \frac{F_{1stcr}}{E_m A_m + E_f A_f} L \quad (21b)$$

203 It is clear to observe that the behaviour of the two samples is completely different.
 204 The example with stiff interface (Fig.7(a)) shows a progressive hardening response, due

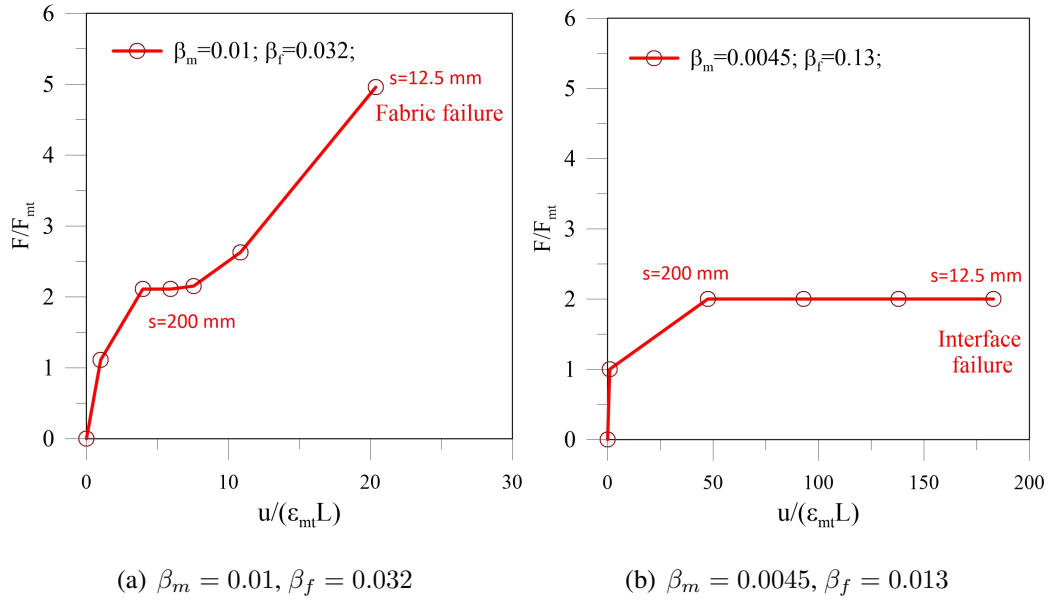


Figure 7: Normalised axial load vs. normalised elongation response $L=200$ mm

205 to the stiffening behaviour provided by the mortar. In this case, as the number of cracks
 206 increases, the system tends to become stiffer. The axial load increases up to the maximum
 207 tensile capacity of the fabric. Differently, the example with deformable interface (Fig.7(b))
 208 shows a limited tensile capacity, due to the fact that the contribution of the matrix can be
 209 considered negligible. Consequently, the system tends to face larger displacements with
 210 negligible contribution in terms of force and failure is reached due to the excessive slip
 211 between fabric and matrix.

212 3.1. Calibration of the interface stiffness

213 As discussed in the previous section, the representation of the tensile constitutive be-
 214 haviour in dimensionless form is dependent only by the numerical parameters β_f and β_m .
 215 However, a reliable calculation of the tensile load vs. elongation curve needs a calibration
 216 of the dimensional parameters needed for calculating β_f and β_m - i.e. A_f, A_m and k .

217 In the current study, the area of the transverse cross section of the fabric A_f is assumed

218 to be equal to the conventional equivalent area of a lamina of dry textile. This assumption,
 219 despite simplified, is coherent with the common conventional models adopted in the tech-
 220 nical codes, especially for FRP composites. It is also observed that this assumption is an
 221 approximation of the area of the yarns, which makes possible to calculate the area of the
 222 fabric based on dimensional parameters. Coherently with this hypothesis, the thickness of
 223 the interface t_i is assumed equal to twice the width of the sample.

224 Finally, the model requires the definition of the interface stiffness k . It should be noted
 225 that this parameter can be considered as a function of the nature of the fiber and of the
 226 matrix. In fact, in general, the stiffness of a zero-thickness interface between two layers
 227 can be assessed as

$$k = \frac{G}{t_p} \quad (22)$$

228 being G the lower shear modulus of the two layers and t_p is the thickness of the process
 229 zone of the interface. In the context of the tensile behaviour of FRCM strips, the process
 230 zone of the interface can be considered equal to the cortical thickness of yarns embedded
 231 by the matrix. Obviously, t_p can be considered a function of the type of fiber and of its
 232 external treatment (coated or uncoated). For this reason, the parameter k is here calibrated
 233 as a function of the mechanical ratio of fabric ω , defined as

$$\omega = \frac{A_f f_t}{A_m f_{mt}} \quad (23)$$

234 being f_t the tensile strength of the fabric.

235 This parameter includes indirectly the shear deformability of the matrix through its
 236 tensile strength f_{mt} and the number of layers through the value of A_f . The value of the
 237 interface stiffness is therefore calibrated experimentally by evaluating the trend of k as
 238 a function of ω for different types of fiber. In particular, for each experimental data the
 239 value of k was evaluated as the optimal value which minimizes the difference between
 240 the experimental $E_{cr,exp}$ and the theoretical $E_{cr,th}$ value of dissipated energy during the

241 post-cracking stage, this last calculated as the area under the tensile force vs. elongation
 242 response after first cracking. Table 1 shows the overall experimental database adopted

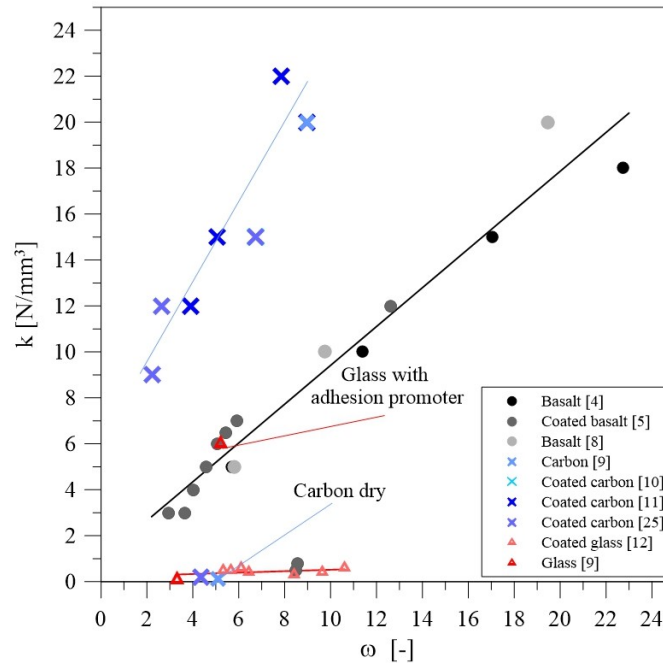


Figure 8: Calibration of k as a function of ω

242

243 for calibrating the value of k . The database includes the mechanical properties of fiber,
 244 matrix and the corresponding values of ω . The result of the calibration is shown in terms
 245 of ratio $E_{cr,exp}/E_{cr,th}$, and the resulting value of k . Basalt, carbon and glass FRCM were
 246 considered with different yarn treatment and for a wide range of ω . It is worth noting that
 247 PBO FRCM was not included in this database due to the different behaviour with the other
 248 types of fibres.

Fig.8 shows the results of the correlation between k and ω for all the considered data,
 grouped for kind of fibers and referring literature work. The graph shows that all the sets
 of experimental data can be fitted with good accuracy by a linear regression model, which

equation depends on the kind of fiber. The application of the least squares approximation of the trend leads to the following results

$$k = 1.743\omega + 6.098 \text{ for coated carbon} \quad (24a)$$

$$k = 0.827\omega + 1.251 \text{ for basalt} \quad (24b)$$

$$k = 0.048\omega - 0.057 \text{ for glass} \quad (24c)$$

249 The reliability of these best fitting curves is confirmed by the values of the coefficient
250 of determination R^2 reported in Fig.8. Values over 90% are obtained for all the cases,
251 confirming that ω can be considered the independent variable for the calculation of k and
252 given kind of fiber. It is observed that only two results were in contrast with the linear
253 trend, as highlighted in the graph due to the application of a treatment on the yarns. Eq.24
254 are therefore valid only for coated carbon and fibers without adhesion promoter at the
255 interface. Further experimental studies need to be addressed for verifying the application
256 of a linear regression model on these systems.

257 **4. Comparison with experimental results**

258 The model was validated against the experimental results shown in Table 1. In the
259 following only some examples are shown in graphical form for the sake of brevity.

260 Fig.9 shows the comparison between experimental data and theoretical predictions for
261 basalt FRCM specimens with the experimental data achieved by Larrinaga et al. [4] and
262 D'Anna et al. [8]. It is clear to observe that good agreement is generally obtained be-
263 tween predictions and experimental results. The model is able to consider the effect of
264 the number of layers and it tends to slightly underestimate the response for the data of
265 D'Anna et al. [8] especially for greater number of layers, while a negligible overestima-
266 tion is observed for low number of layers for the data of [4]. It is also worth observing that

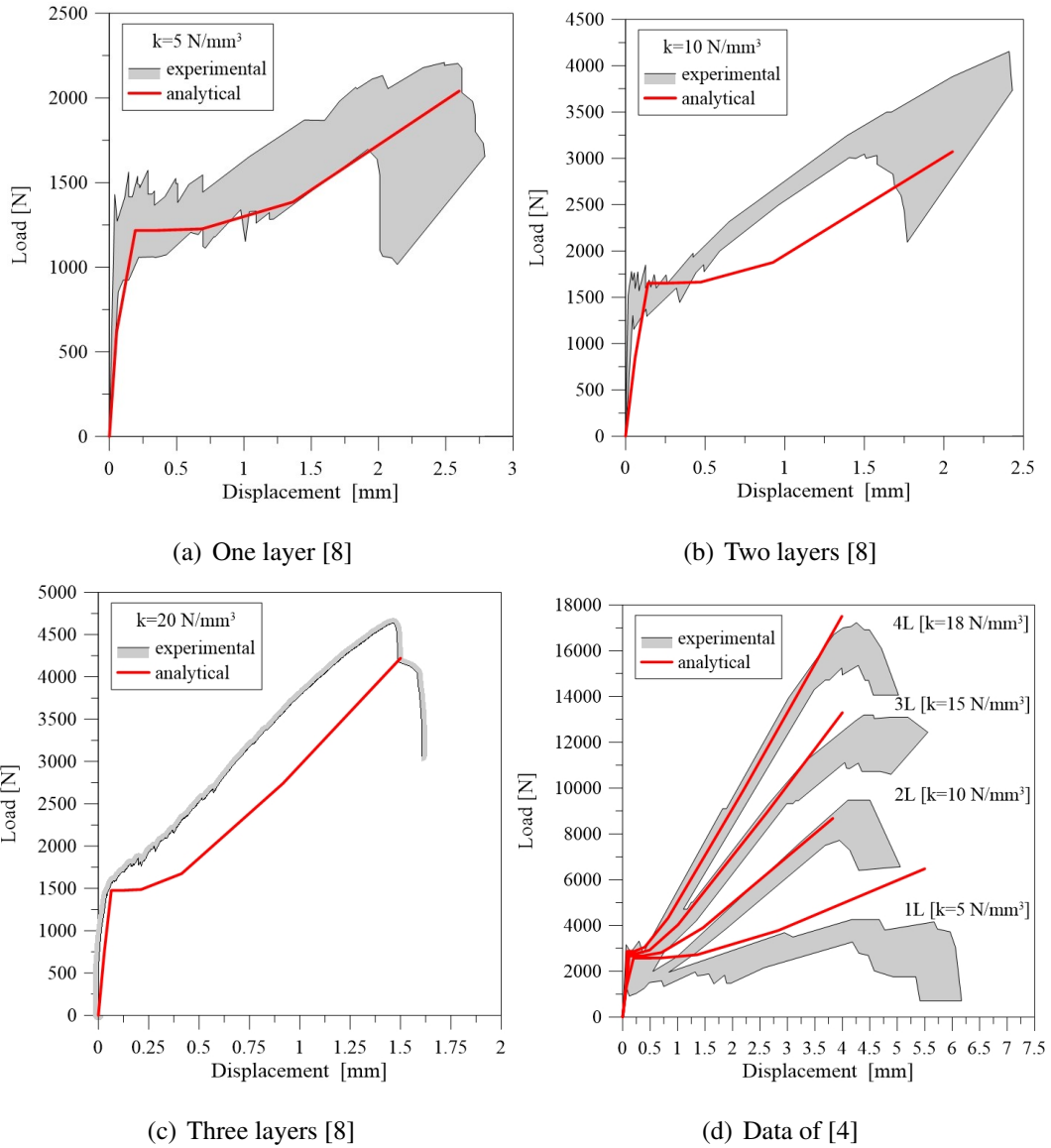


Figure 9: Comparisons with experimental data for basalt FRCCM

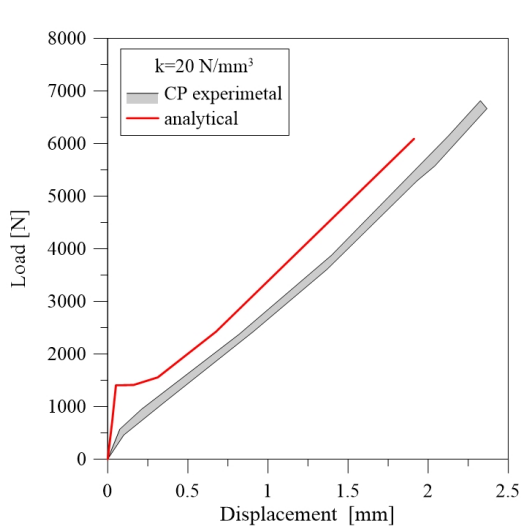
267 for specimens tested by D'Anna et al. [8] the model provides an estimated step of cracks
268 equal to about 31 mm, 25 mm and 19 mm for one, two and three layers respectively, while
269 the experimental values of cracks spacing are 37 mm, 16 mm, and 11 mm for one, two and
270 three layers respectively.

271 Similarly Fig.10 shows the comparison between predictions and experimental data
272 available in the literature [9][10][11] for carbon FRCM specimens. Also in these cases
273 the proposed procedure is capable of catching the average experimental response for dif-
274 ferent number of layers and for different sources of the experimental data. It is also worth
275 observing that as expected the value of k increases for greater number of layers. Fig.11
276 shows similar comparisons for Glass FRCM specimens. Experimental data refer to the
277 works of Bellini et al. [9], Bramato et al. [12] and Donnini et al. [26] . It is observed that
278 lower values of k are associated to the tensile behaviour of the specimens, which usually
279 shows a bilinear response with a negligible second stage. The model overestimates the
280 first branch of the response for the cases of Fig.11(a) and 11(b), probably due to the un-
281 certainties about the mechanical properties of the constituent materials, but fits well the
282 experimental response in the last stage for all the data.

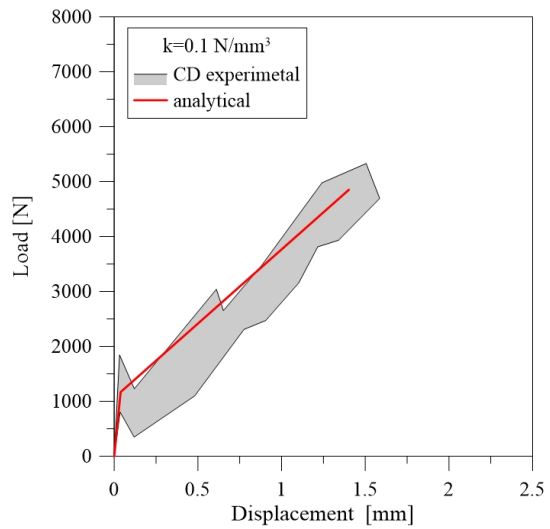
283 Tensile behaviour requires to take into account complex mechanisms. However, the
284 proposed simplified theoretical model is able to simulate fairly well the behaviour of
285 FRCM composites. It should be observed that the same experimental data were used
286 for the calibration of k and therefore a more reliable validation should be made with new
287 experimental results outside from the considered database.

288 **5. Conclusions**

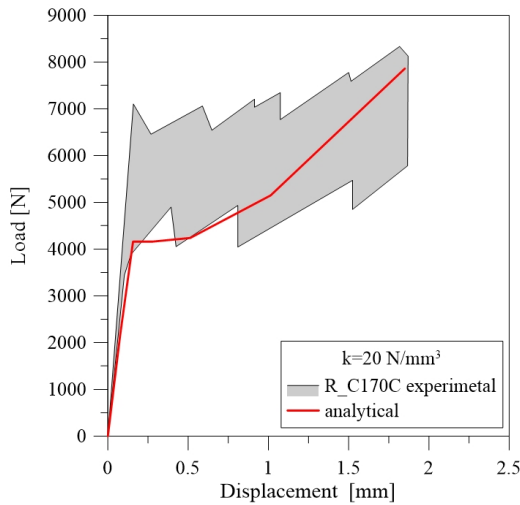
289 This paper presented the proposal of a truss element for modelling the tensile response
290 of FRCM strips. The model was formulated by studying the mechanics of force trans-
291 mission between fabric and matrix within the crack spacing, assuming that the two phases



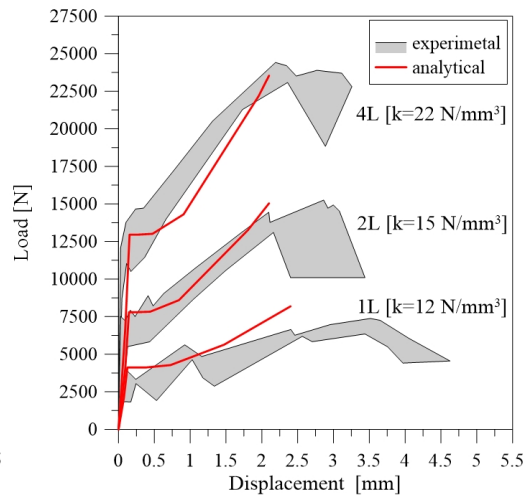
(a) Data of [9] carbon fibers with adhesive promoter



(b) Data of [9] dry carbon



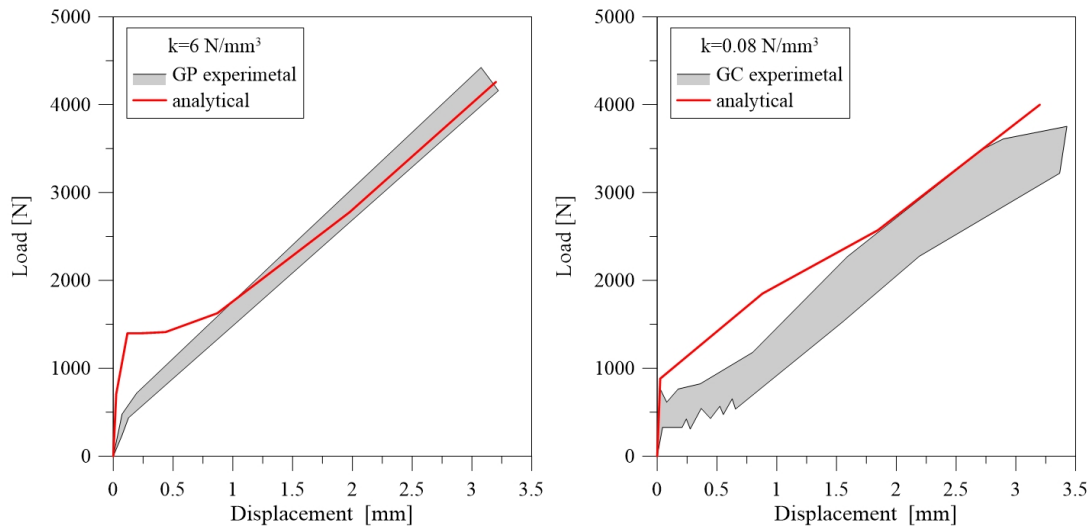
(c) Data of [10] coated carbon



(d) Data of [11] coated carbon

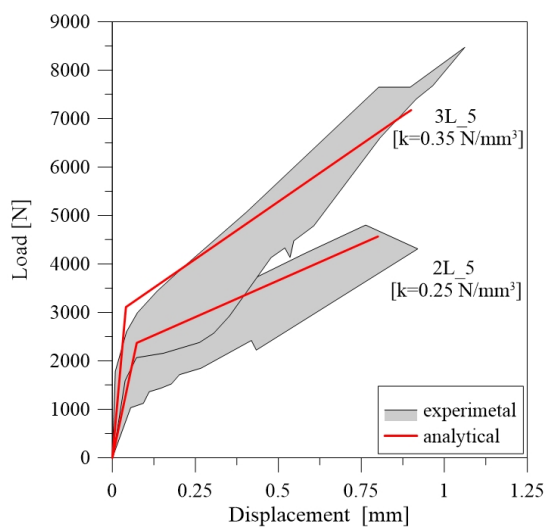
Figure 10: Comparisons with experimental data for Carbon FRCM

292 can be studied as truss elements connected through a shear interface. The definition of the
 293 coupled truss allows to set a procedure for obtaining the tensile response of the composite
 294 material, under the assumption of linear behaviour of fabric, matrix and interface.

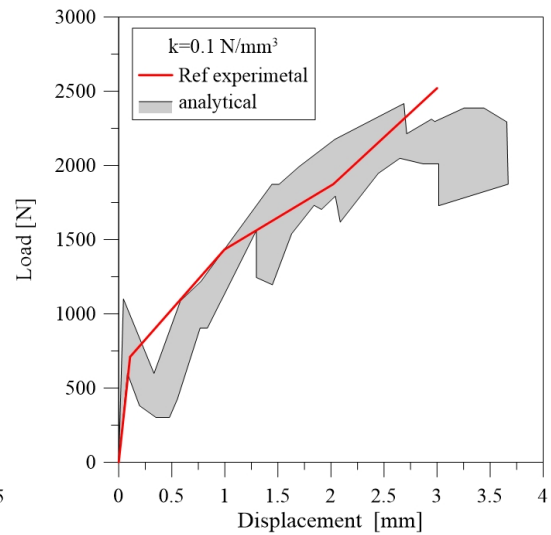


(a) Data of [9] glass fibers with adhesive promoter

(b) Data of [9] coated glass fibers



(c) Data of [12] coated glass fibers



(d) Data of [26] coated glass fibers

Figure 11: Comparisons with experimental data for Glass FRCM

295 On the basis of the obtained results and for the range of analyzed variables, the follow-
 296 ing conclusions can be drawn:

- 297 • the numerical parameters β_m and β_f rule the force transmission mechanism between

298 matrix and fabric. Force in the fabric and interface shear stress tend to intensify in
299 correspondence of the cracks for greater values of k (i.e. stiffer interface);

300 • the normalised representation of the tensile response of FRCM strips F/F_{mt} vs.
301 $u/(\epsilon_{mt}L)$ proved to be dependent only by the numerical parameters β_m , β_f and
302 from the length of the strip L ;

303 • it should be considered that previous studies showed that the form of textile rein-
304 forcement cross-section has a substantial influence on the load carrying capacity of
305 FRCM. However the parameter k can be considered a function of the mechanical
306 ratio of fabric reinforcement ω and from the nature of the fiber. The experimental
307 calibration of k allows to find a linear variation of k as a function of ω for differ-
308 ent kinds of fiber. However, this correlation proved to be affected by the treatment
309 of the yarns and different results can be achieved for fibers applied with adhesion
310 promoter;

311 • the proposed procedure was validated against experimental results available in the
312 literature, proving to be reliable. It should be noted that the validation was made
313 against experimental data included in the database for the calibration of k and for this
314 reason, further experimental investigation should be made for a broader validation
315 of the model.

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Table 1: Experimental database

Ref.	ID sample	Fiber		Mortar		k [N/mm ³]	$\frac{E_{cr,exp}}{E_{cr,th}}$	ω
		Type	E_f [GPa]	Type	f_{mt} [MPa]			
[4]	TB1	Basalt	67	cement-based	2.48	5	1.06	5.7
	TB2	Basalt	67	cement-based	2.48	10	0.98	11.4
	TB3	Basalt	67	cement-based	2.48	15	0.99	17.1
	TB4	Basalt	67	cement-based	2.48	18	1.07	22.7
[5]	FRCM 1	Coated basalt	110	cement-based	0.80	0.8	0.85	8.6
	FRCM 1	Coated basalt	116	cement-based	1.50	4	1.09	4.0
	FRCM 1	Coated basalt	111.5	cement-based	1.30	6	0.99	5.1
	FRCM 2	Coated basalt	111.5	lime-based	1.20	6.5	1.16	5.4
	FRCM 2	Coated basalt	107	lime-based	1.00	12	1.03	12.6
	FRCM 3	Coated basalt	71.3	lime-based	1.30	7	1.12	5.9
	FRCM 4	Coated basalt	89	lime-based	0.50	0.5	0.97	8.5
	FRCM 4	Coated basalt	51	lime-based	0.50	3	1.06	3.0
	FRCM 4	Coated basalt	45.3	lime-based	1.00	3	0.96	3.7
	FRCM 4	Coated basalt	45.3	lime-based	0.80	3	1.07	4.6
[8]	1L	Basalt	83	cement-based	2.00	5	1.24	5.8
	2L	Basalt	83	cement-based	2.00	10	1.17	9.6
	3L	Basalt	83	cement-based	2.00	20	1.21	19.5
[9]	CP	Carbon ^(*)	240	hydraulic lime-based	1.92	20	1.14	9.0
	CD	Carbon	240	hydraulic lime-based	2.91	0.1	1.36	5.1
	GP	Glass ^(*)	65	hydraulic lime-based	1.92	6	0.94	5.2
	GC	Coated glass	70	hydraulic lime-based	2.00	0.08	0.88	3.3
[10]	R- C170CC	Coated carbon	219	cement-based	3.35	15	1.14	6.7
[11]	L1	Coated carbon	196.4	cement-based	2.50	12	0.79	3.91
	L2	Coated carbon	196.4	cement-based	2.50	15	0.96	5.1
	L4	Coated carbon	196.4	cement-based	2.50	22	1.01	7.8
[12]	1L	Coated glass	108	lime-based	1.00	0.3	1.19	10.6
	2L-3	Coated glass	108	lime-based	2.00	0.55	1.06	12.9
	3L-3	Coated glass	108	lime-based	2.20	0.55	1.05	13.2
	2L-5	Coated glass	108	lime-based	2.50	0.25	0.81	5.7
	3L-5	Coated glass	108	lime-based	2.50	0.35	0.93	6.9
	2L-10	Coated glass	108	lime-based	1.10	0.25	0.99	6.4
	3L-10	Coated glass	108	lime-based	1.50	0.3	0.96	5.3
[25]	FRCM 6	Coated carbon	187	cement-based	3.35	12	1.26	2.6
	FRCM 6	Coated carbon	219	cement-based	4.00	9	1.12	2.22
[26]	Ref	Coated glass	67.58	cement-based	0.80	0.1	0.84	4.3
	S-40	Coated glass	67.87	cement-based	0.80	0.1	0.90	4.0
	A-40	Coated glass	69.37	cement-based	0.90	0.08	0.87	2.9
	FT	Coated glass	66.21	cement-based	0.80	0.09	0.86	4.0

(*) Adhesion promoter