

A computational and experimental investigation of the anchoring of organosilanes on the halloysite silicic surface

Lorenzo Lisuzzo^a, Marco Bertini^a, Giuseppe Lazzara^a, Chiara Ferlito^a,
Francesco Ferrante^{a,*}, Dario Duca^a

^a*Dipartimento di Fisica e Chimica "E. Segrè", Università degli Studi di Palermo,
Viale delle Scienze - 90128 Palermo, Italy*

Abstract

In this work, the effect of halloysite nanotubes alkali activation on its grafting efficiency with organosilanes was studied by Density Functional Theory and experimental investigations. In particular, computational analysis allowed to enlighten the structural properties of the organic molecules attached to the silanol groups on halloysite outer surface. The energetics of the reactions showed that the pretreatment with a base is crucial for the modification of the surface due to the appearance of a high number of active sites which led to thermodynamically favored exothermic processes. Experimental evidences were in good agreement with calculation hypothesis. For instance, the coating efficiency is higher after the alkali activation of the inorganic counterpart for both the investigated organosilanes. The findings here reported are important in order to improve any functionalization protocols for aluminosilicates without variations or loss of the hollow nanotubular morphological features and it paves the ground to halloysite based technological applications in many fields, from nanotechnology to catalysis.

Keywords: Halloysite nanotubes, Functionalization, Siloxanes, Heterogenous catalysis

*Corresponding author

1. Introduction

Clay minerals attract the interest of the scientific community due to their peculiar properties, which make them good platforms for the design of functional inorganic or hybrid materials to be exploited in many technological applications. Among clays, Halloysite Nanotubes (Hal) hold a certain importance as a result of both their surface chemistry and morphological features (Lvov et al., 2016). Hal are natural occurring 1:1 aluminum phyllosilicates with $\text{Al}_2\text{Si}_2\text{O}_5(\text{OH})_4 \cdot n\text{H}_2\text{O}$ as stoichiometric formula, where n can be 0 or 2 and it corresponds to the number of water molecules in the interlayer space, thus resulting in either the dehydrated 7Å-halloysite or in the hydrated 10Å-halloysite, respectively (Zsirka et al., 2017; Makó et al., 2020). Being structurally composed by the alternation of a tetrahedral Si–O based sheet which is overlapped to an octahedral Al–OH sheet, halloysite displays different groups in the external and internal surfaces (Hua et al., 2023; Ferrante et al., 2017). In particular, the kaolinite-like sheet rolls up, thus providing the nanoclay with its most distinctive property, i.e. its hollow nanotubular shape (Zhang et al., 2019a; Liu et al., 2016). As a result, the external silicon based surface is negatively charged and the inner aluminum based surface is positively charged in the pH range from 2 to 8 (Vergaro et al., 2012). Since it is extracted from natural deposits in different geographical areas, the dimensions of halloysite nanoparticles can be different (Lisuzzo et al., 2022). The length may vary from the submicron scale to 2-3 μm , the outer diameter from 15 to 200 nm whereas the inner diameter can range from 10 to 100 nm (Pاسبakhsh et al., 2013; Liu et al., 2021).

Several studies report about the biocompatibility and lack of cytotoxicity of halloysite nanotubes. (Rozhina et al., 2021; Kryuchkova et al., 2016) These aspects, together with the high specific surface area, eco-compatibility, abundant availability and low cost, are essential to burst their use in industrial domains. For instance, Hal have been widely investigated as inorganic fillers for nanocomposites manufacturing, showing their effect in improving thermal and mechanical properties of, among others, bioplastics and nanopapers (Lisuzzo et al.,

2020a). Also, they found application in food packaging, environmental remediation, building science, drug delivery, cultural heritage treatment and catalysis (Sadjadi et al., 2017; Sidorenko et al., 2018, 2021; Spepi et al., 2016; Liu et al., 2019; Panchal et al., 2020; Haw et al., 2020). Nonetheless, there are some issues to handle and to overcome before using Hal and exploiting their full potential. Halloysite does not display a long term colloidal stability, as the nanotubes tend to aggregate and settle at the bottom of aqueous dispersions, being the sedimentation driven by the high density of the nanoparticles (Lisuzzo et al., 2019). However, reaching a good stability is imperative for every application of halloysite, especially in liquid media, and it is needed as first condition in the synthetic protocol for the preparation of new materials. In this scenario, the chemical functionalization of halloysite surfaces showed to play a major role. Indeed, the inner positive surface can be functionalized with negative molecules, whereas the outer negative surface can be modified with positive species (e.g. polymers, surfactants, biopolymers) by electrostatic interactions (Zhang et al., 2019b). Moreover, the surface modification can also be conducted by chemical covalent grafting with organosilanes (Mehdizadeh et al., 2022). As a matter of fact, the functionalization of clays with silanes is important for polymer-clay nanocomposites, since it enhances the dispersion of the solid nanoparticles in the polymeric matrix thus improving the mechanical properties of the resulting materials. Literature reports about the grafting of halloysite inner surface, which bears hydroxyl groups (aluminols), with organosilanes and about the effects of thermal and evacuation pretreatments, where the former is carried out to remove physically adsorbed water and to improve the grafting efficiency (Yuan et al., 2008a). However, the silanization of the external surface is still an open issue because it consists of siloxane (Si–O–Si) groups with just few Si – OH groups at the edges of the nanotubes or in some defects (Ferrante et al., 2015). Undoubtedly, the presence of hydroxyl groups is the necessary condition to conduct any covalent modification. In order to improve the functionalization efficiency, halloysite nanotubes are very often pretreated or alkali activated (Wang et al., 2013; Shankar et al., 2018; De Silva et al., 2015).

62 In our previous work, we conducted computational and experimental inves-
63 tigations to enlighten the mechanisms of silanol groups formation on the outer
64 silicic surface of Hal as products of the reactions in alkaline aqueous dispersions
65 (Ferrante et al., 2023). Herein, it was found that the creation of different arrays
66 of Si–OH groups is a highly exothermic process in the presence of sodium hy-
67 droxide and the final activated nanotubes did not change their morphological
68 features, despite the increased number of active sites. Being the silanization of
69 the outer surface important in many technological applications, from the deliv-
70 ery of biologically active species (Liu et al., 2016; Massaro et al., 2022), to the
71 removal of pollutants from aqueous solutions (Cataldo et al., 2018; Zhu et al.,
72 2017), to the immobilization of metal nanoparticles for catalysis (Sadjadi & Atai,
73 2018), its optimization and deep understanding are crucial. The present work is
74 aimed at investigating the role of alkaline activation on the efficiency of silanes
75 grafting on halloysite silicon-based surface. The formation of Si–OH groups and
76 their reactions with organosilanes, namely the (3-aminopropyl)triethoxysilane
77 (APTES) and the N-3-(trimethoxysilyl)propyl ethylenediamine (AEAPTMS;
78 see Figure 1), will be focused by a density functional theory based computa-
79 tional approach and the results will be corroborated by experimental analysis to
80 provide meaningful insights about the effects of chemical activation and about
81 the grafting mechanism and efficiency. It should be noted that, among the
82 different applications, the silanization of halloysite outer surface is a challenge
83 for industrial and catalytic purposes. The termination of the silane chain can
84 bear different atoms, each one with a peculiar affinity towards metal particles.
85 This factor, together with halloysite hollow nanotubular proper morphology, can
86 lead to the design of multicatalytic system with different active sites in isolated
87 environments, each one catalyzing a specific reaction or process.

88 2. Models and Computation Details

89 All the calculations in this work were carried out by using the Gaussian 16
90 software (Frisch et al., 2016). The geometries of halloysite functionalized struc-

91 tures were optimized within the Density Functional Theory (DFT) framework,
92 by using the M06-L exchange-correlation functional (Zhao & Truhlar, 2006) to-
93 gether with the resolution of identity approximation (Eichkorn et al., 1997), in
94 order to reduce the computational cost for investigating systems with large di-
95 mensions. The SVP valence double-zeta plus polarization basis set of Ahlrichs
96 and co-workers (Schaefer et al., 1992) was employed and the corresponding
97 auxiliary functions needed for RI approximation were generated at run time.
98 A cluster approach was used throughout the whole investigation. The starting
99 pristine halloysite surface was tailored from a nanotube model built and inves-
100 tigated through a Density Functional Tight Binding approach (Ferrante et al.,
101 2015, 2017). Herein, all the oxygen atoms placed at the cutting edges were
102 kept frozen in their positions during geometry optimization, thus maintaining
103 the surface curvature. Aimed at investigating the functionalization of halloysite
104 surface with organosilanes, APTES and AEAPTMS were considered and their
105 attachments to the clay surface were investigated by exploiting their most stable
106 conformations obtained in vacuum conditions. In particular, in the optimized
107 geometry of APTES the alkyl chain is bent as a result of very weak interactions
108 between the nitrogen atom and the methylenic hydrogen, while the geometry
109 of the alkyl chain in AEAPTMS is ruled by the intramolecular NH \cdots N hydro-
110 gen bond. For what concerns halloysite, the Hal-Si(2, 1) $^-$, the Hal-Si(4, 2) $^{2-}$,
111 the Hal-Si(6, 3) $^{3-}$ and the Hal-Si-v $_1^-$ structures taken from (Ferrante et al.,
112 2023) represent the starting points for the simulated functionalization reactions.
113 The resulting modified structures will be labeled as Hal-S x A y , where S indicates
114 the silicic surface and A = APTES or AEAPTMS; x is the number of silanol
115 groups on the starting surface model (or $x = v$ for the system derived from
116 Hal-Si-v $_1^-$, having a silicon vacancy) and y is the number of -OR groups (R
117 = Et or Me) left on A after the functionalization.

118

[Figure 1 about here.]

119 **3. Materials and Methods**

120 *3.1. Materials*

121 Halloysite (specific surface area of $65 \text{ m}^2 \text{ g}^{-1}$ and specific gravity of 2.53
122 g cm^{-3}), (3-Aminopropyl)triethoxysilane (APTES, $\text{H}_2\text{N}(\text{CH}_2)_3\text{Si}(\text{OC}_2\text{H}_5)_3$, 99%),
123 N-3-(trimethoxysilyl)propyl ethylenediamine (AEAPTMS, $(\text{CH}_3\text{O})_3\text{Si}(\text{CH}_2)_3\text{NHCH}_2\text{CH}_2\text{NH}_2$,
124 97%), sodium hydroxide ($\geq 97.0\%$) and toluene (anhydrous, 99.8%) were pur-
125 chased from Sigma-Aldrich and used without any further purification. Absolute
126 ethanol ($\geq 99.8\%$) is a Honeywell product.

127 *3.2. Alkaline modification of Halloysite*

128 The alkaline activation of halloysite was carried out as reported in literature
129 (Ferrante et al., 2023). Briefly, a dispersion containing 2 g of the nanoclay in 40
130 cm^3 of NaOH 4M was prepared and kept under stirring for 30 minutes at 80°C ,
131 before being separated by centrifugation (5000 rpm, 10 min), washed and dried
132 overnight at 60°C . The sample will be referred to as alkaline Hal (aHal).

133 *3.3. APTES functionalization of Halloysite*

134 Grafting of Hal outer surface with γ -Aminopropyltriethoxysilane was con-
135 ducted as described in literature by Yuan et al. (Yuan et al., 2008a). Briefly,
136 0.6 g of the clay was added to a solution composed of 2 cm^3 APTES in 25 cm^3
137 of toluene. The dispersion was sonicated for 30 min and refluxed for 20 h at
138 120°C under constant stirring. The organosilane functionalized clay was sep-
139 arated by centrifugation and washed six times with toluene before being dried
140 overnight. Both pristine and alkaline halloysite were functionalized according
141 to this procedure, resulting in the Hal-APTES and aHal-APTES samples.

142 *3.4. AEAPTMS functionalization of Halloysite*

143 Grafting of Hal outer surface with N-3-(trimethoxysilyl)propyl ethylenedi-
144 amine was performed using a procedure similar to that reported in literature
145 by Takahara et al. (Takahara et al., 2005). In particular, 0.5 g of the clay
146 was added to 50 cm^3 of solution containing AEAPTMS 0.4 M in toluene. The

147 dispersion was sonicated and magnetically stirred for 16 h at room temper-
148 ature. The aminoalkylated clay was separated by centrifugation and washed
149 with ethanol before being dried for 4 h. Both pristine and alkaline halloysite
150 were functionalized according to this procedure, resulting in the Hal-AEAPTMS
151 and aHal-AEAPTMS samples.

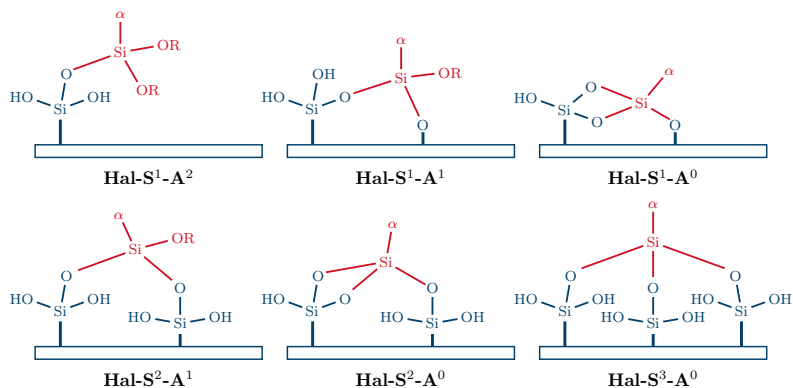
152 3.5. Experimental analysis

153 Thermogravimetry (TGA) was conducted by using a Q5000 IR (TA Instru-
154 ments) apparatus operating with N_2 flow at $25 \text{ cm}^3 \text{ min}^{-1}$ for the sample and
155 $10 \text{ cm}^3 \text{ min}^{-1}$ for the balance, respectively. The investigated temperature range
156 was set from room temperature to $800 \text{ }^\circ\text{C}$, the heating rate was $20 \text{ }^\circ\text{C min}^{-1}$.
157 The calibration was previously carried out on the basis of the Curie tempera-
158 tures of standards (Blanco et al., 147-153). Fourier transform infrared inves-
159 tigation was performed through a Frontier FTIR spectrometer (PerkinElmer),
160 working at room temperature using 64 scans in the range between 4000 and 500
161 cm^{-1} with 2 cm^{-1} spectral resolution. The experiments were conducted on KBr
162 pellets with an amount of milled sample $<2 \text{ wt}\%$. X-ray Fluorescence (XRF)
163 Spectrometry was carried on by means of an Olympus Innov X DS-2000 Delta
164 Standard Alloy XRF Handheld Analyzer operating in the Alloy Plus analysis
165 mode. The X-Ray Diffraction (XRD) analysis was conducted through a Rigaku
166 (MiniFlex) diffractometer based on a copper $K\alpha$ radiation source including a
167 nickel filter. The patterns were recorded in the range $2\text{-}70^\circ$ with a rate of
168 $20^\circ \text{ min}^{-1}$ and a step of 0.02° . The voltage was 40 kV and the current 15 mA.

169 4. Results and Discussion

170 4.1. Halloysite functionalization DFT analysis

171 All the investigated functionalization processes are collected in Table 1, to-
172 gether with the corresponding global reaction energies taken by starting from
173 the pristine halloysite model as reactant. The partial reaction energies corre-
174 sponding to the attachment of APTES and AEAPTMS on the functionalized



Scheme 1: Pictorial representation of the six $\text{Hal-S}^m\text{A}^{3-n}$ systems ($m, n = 1, 2, 3$) derived by functionalization of the Hal model bearing silanol groups protruding from the silicic surface. The blue atomic centers belong to the Hal silicic surface, which is represented by the rectangle; the parts in red color are the fragments of APTES ($\alpha = (\text{CH}_2)_3\text{NH}_2$, $\text{R} = \text{CH}_2\text{CH}_3$) or AEAPTMS ($\alpha = (\text{CH}_2)_3\text{NH}(\text{CH}_2)_2\text{NH}_2$, $\text{R} = \text{CH}_3$) remaining after the functionalization.

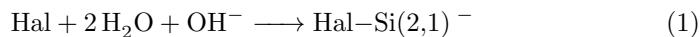
175 halloysite models, instead, are reported in Table 2. A look at these tables re-
 176 veals that the differences, in term of reaction energies, between APTES and
 177 AEAPTMS functionalizations are not relevant, so that in the following a uni-
 178 fied discussion can be reported. Scheme 1 shows the conceptualization of the 6
 179 $\text{Hal-S}^x\text{-A}^y$ systems, whose structural and energetic properties will be detailed
 180 in the following.

181 [Table 1 about here.]

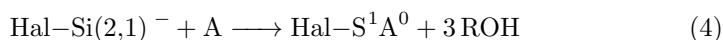
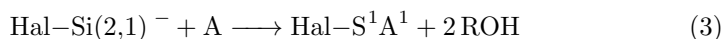
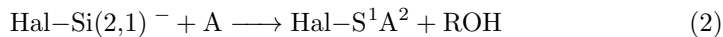
182 [Table 2 about here.]

183 The alkaline activation of halloysite surface after treatment with a strong
 184 base was investigated in our previous study. Herein, it is reported about the for-
 185 mation of $\text{Hal-Si}(n,m)^{m-}$ systems and the appearance of surface silanol groups
 186 through the reaction of halloysite with water and hydroxide ions in the ratio

187 1:n:m (Ferrante et al., 2023). In particular, the following reaction:



resulted in the formation of a structure, which consisted of a $\text{Si}(\text{OH})_3$ group protruding from the external surface of the clay, two SiOH moieties on the silicic array and a negative charge on an interlayer oxygen center. Its formation occurred with an energy release of ca. 700 kJ mol^{-1} and, as a consequence, it was strongly favored from the thermodynamic point of view. As a matter of fact, the appearance of silanol groups on Hal silicic surface is crucial since they represent the anchor points for the attachment of the organosilanes in the functionalization reactions. The processes that were investigated at first are the following ones:



188 The optimized structures of functionalized Hal produced by the reaction (2)
189 above are reported in Figure 2. It should be noted that from the structural
190 point of view there are no sensible differences between the outcomes of APTES
191 or AEAPTMS functionalizations: both geometries show the formation of a hy-
192 drogen bond between the OH from the surface and the OR moiety from the
193 organosilane. Further, in both cases the N-alkyl chain is elongated perpendic-
194 ularly to the silicic surface. Reaction (2) occurs with the release of only 31
195 kJ mol^{-1} (Table 2), but it becomes largely exothermic (Table 1) if coupled with
196 reaction (1). The product of process (3) forms after two hydrolysis reactions
197 involving one Si-OH from the silanol group and the OH moiety left on the sur-
198 face after the $\text{Hal-Si}(2,1)^-$ formation, with a global release of energy slightly
199 higher than the one calculated for reaction (4). In the optimized geometry
200 of the $\text{Hal-S}^1\text{A}^1$ system, showed in Figure S1 and S2 of Supporting Informa-
201 tion the two other groups bonded to the silicon center of APTES are forced
202 to lie almost parallel to the silicic surface; this is true also for the first part

203 of the $-(\text{CH}_2)_3\text{NH}(\text{CH}_2)\text{NH}_2$ chain in AEAPTMS, while the second fragment
 204 stretches upwards. A sort of destabilization effect in $\text{Hal}-\text{S}^1\text{A}^1$ is likely due
 205 to deformation of the bonding configuration occurring between the A fragment
 206 and the protruding silanol; in fact, all the configuration tension seems to be
 207 located on the latter, being this group more mobile than the surface hydroxyl,
 208 and the results are a sensibly wider Si–O–Si angle (increased by 15°) and longer
 209 Si–O bond (by ca. 0.05 \AA) with respect to the corresponding configuration in
 210 $\text{Hal}-\text{S}^1\text{A}^2$. In process (4) two OH groups from the raised $\text{Si}(\text{OH})_3$ tetrahedron
 211 and one OH group from the OH moiety are the active sites for the nucleophilic
 212 attack on the silicon center, with the resulting $\text{Hal}-\text{S}^1\text{A}^0$ system showing no OR
 213 group left on A. In this case the functionalization with both the organosilanes
 214 is a highly endothermic process, as it needs an energy of ca. 190 kJ mol^{-1} to
 215 occur. This effect can be most likely related to the poor availability of active
 216 Si–OH sites on the surface, necessarily leading to the formation of a system
 217 showing an innatural Si–O–Si–O ring, which provides ring tension to the final
 218 optimized structures (see Figure S3 and S4 of Supporting Information). Tak-
 219 ing into account the exothermicity of $\text{Hal}-\text{Si}(2,1)^-$ formation (Ferrante et al.,
 220 2023), the global reaction energies calculated starting from pristine halloysite
 221 are -509.4 and $-507.9 \text{ kJ mol}^{-1}$, but it is clear that this situation is not a likely
 222 one: when the Hal surface is sparsely populated by silanol groups the func-
 223 tionalization would occur through many partial hydrolysis processes on a large
 224 number of sites instead of total hydrolysis on few sites.

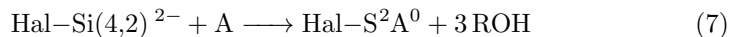
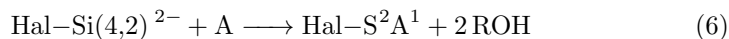
225 [Figure 2 about here.]

226 A step forward is moved by investigating the functionalization of the $\text{Hal}-\text{Si}(4,2)^{2-}$
 227 structure, which is the result of the following reaction:



which shows two $\text{Si}(\text{OH})_3$ groups protruding from the surface of the halloysite
 model, four $\text{Si}(\text{OH})$ groups embedded in the silicic array and two negative
 charges as a consequence of the alkaline treatment. In this case, the com-

putational study was focused on the energetic of the following processes:



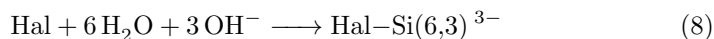
228 The optimized structures of the products of reaction 6, with A = APTES and
229 AEAPTMS, are reported in Figure 3. In these structure a network of hydro-
230 gen bonds exists also after the functionalizations, involving the surface and the
231 silanolic OH groups. The geometry of the surface anchoring sites, along with
232 the position of OR moiety, make the N-alkyl chain to tilt sensibly with respect
233 to the surface normal. The reaction step 6 shows only low exothermicity, which
234 obviously is more than compensated by the energy release characterizing reac-
235 tion 5, making the global process to show an exothermicity larger than 1100
236 kJ mol^{-1} . The fact that this value is well below than the double of the energy
237 released by the combined 1 and 2 processes is due to the issue, already hypoth-
238 esized in (Ferrante et al., 2023), regarding the energetic preference toward the
239 presence of silanol groups sparse on the Hal surface with respect to the for-
240 mation of clustered configurations, a situation, this last, that would occur at
241 extreme alkaline conditions.

242 [Figure 3 about here.]

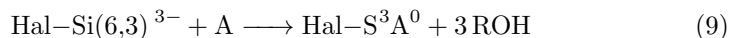
243 Bearing the surface of Hal-Si(4,2)^{2-} two tetrahedra, there is no room for
244 interactions with surface OH groups, so the nucleophilic attack on the Si center
245 of the silanes is conducted by two OH groups from one Si(OH)_3 and one OH
246 group from a second Si(OH)_3 . As happened in the case of $\text{Hal-S}^1\text{A}^0$ forma-
247 tion, the presence of a tensioned Si-O-Si-O ring in the $\text{Hal-S}^2\text{A}^0$ product
248 (see Figure S5 and S6 of Supporting Information) makes reaction (7) highly
249 endothermic, both for the APTES and AEAPTMS functionalizations. Never-
250 theless, the loss in exothermicity in this case, when reactions (6) and (7) are
251 compared and global reaction energies are considered, is equal to 160 kJ mol^{-1}
252 in the average, while the same quantity was equal to ca. 210 kJ mol^{-1} for the
253 corresponding processes occurring in the Hal-Si(2,1)^- ancestor, represented by

254 equations (2)-(4). This can be considered another hint that a larger availability
255 of active Si-OH sites on the Hal surface is beneficial to the functionalization
256 reactions.

257 In order to have more insights about the effect of the alkaline pretreatment on
258 the efficiency of functionalization, we carried out further computational stud-
259 ies by considering a third structure, namely Hal-Si(6,3)³⁻, produced by the
260 reaction:



It shows three Si(OH)₃ tetrahedra protruding from the surface of halloysite, three negative charges and high configuration flexibility as a consequence of the alkaline treatment. In this case, the computational study was focused on the following process:



261 Here, the nucleophilic attack on the Si center of the silanes is conducted by
262 three OH, each one of them coming from one of the three different tetrahedra
263 raised above the halloysite surface. The optimized structures are reported in
264 Figure 4. It was found that the best geometrical arrangement for the anchoring
265 portion in this system is not a symmetrical one, as one could imagine by mentally
266 drawing the Si atom of the functionalizing organosilane at the center of a triangle
267 whose vertex are occupied by the silanol groups of the surface. As a matter of
268 fact, the optimized geometry of the portion above can be viewed as originated
269 by two almost equivalent silanol groups plus a third group which adjusts its
270 position driven by forces which, other than the main geometric constraints,
271 include the formation of hydrogen bond between the silanols, the asymmetry
272 of the underneath surface and even by the interaction with the amino group(s)
273 of A (see inset in Figure 4). This configuration, that can be considered as
274 representative of this kind of situations, allows to minimize steric hindrance
275 and to fit almost perfectly the A residue between the three silanol groups, as
276 witnessed by the geometrical parameters of the newly formed -SiO₃ fragment
277 (Si-O bonds lengths in the range 1.65-1.67 Å and O-Si-O bond angles in

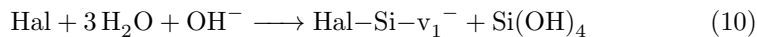
278 the range 107-110°), which resulted essentially equivalent to those of the same
279 fragment in other part of the pristine surface.

280 [Figure 4 about here.]

281 Despite the reactions of Hal–Si(6,3)³⁻ with both APTES and AEAPTMS
282 release a small quantity of energy ($\Delta E = -15.0$ and -43.3 kJ mol⁻¹, respec-
283 tively), the global reactions energies, which are calculated starting from pristine
284 halloysite nanotubes and by taking into account the formation energy (-1395.4
285 kJ mol⁻¹) of Hal–Si(6,3)³⁻, are -1410.4 kJ mol⁻¹ for A = APTES and -1438.8
286 kJ mol⁻¹ for A = AEAPTMS. Then, these reactions are thermodynamically the
287 most favored among those investigated, in agreement with the high availability
288 of Si–OH active sites on the surface of the nanoclay model and the disappear-
289 ance of the ring tensions which affected some of the previous structures.

290 At this point of the discussion it is worth to note that, due to the lacking
291 of cooperativity characterizing the formation of silanol groups (Ferrante et al.,
292 2023), the most favourable situation is the one where a large number of func-
293 tionalization reactions take place through a single hydrolysis process, like (2) and
294 (6), localized in regions of the outer Hal surface which are distant from each
295 other. Reactions like (9) would occur instead when silanol groups are clustered on
296 the surface, which could happen after the treatment at high alkaline conditions.
297 It is to say that this last arrangement would be auspicious, since it could lead
298 to the realization of a surface with a high concentration of functionalization,
299 hence a high density of anchoring point, useful, for example, to stabilize metal
300 nanoparticles.

301 Literature also reports that, as an extremization of the alkaline treatment,
302 the breaking of a fourth Si–O bond can be caused together with the extraction
303 of orthosilicic acid, Si(OH)₄, from the surface of halloysite. The corresponding
304 reaction is reported below:



305 where the silicic surface presents a vacancy surrounded by OH groups. Since this
306 property can be important for the functionalization of Hal with organosilanes,

307 whose alkoxyethyl moieties could be incorporated in the surface, the following
308 reactions were simulated:

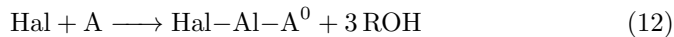


309 In this case, the nucleophilic attack is carried out from three OH left at the
310 center of the vacancy, after the extraction of the orthosilicic acid. The optimized
311 structures of the $\text{Hal-S}^v\text{A}^0$ species are reported in Figure 5.

312 [Figure 5 about here.]

313 Reactions (11) occur with an energy consumption of 53.0 and 58.9 kJ mol^{-1} for
314 $\text{A}=\text{APTES}$ and AEAPTMS , respectively. The reactions global energies, calcu-
315 lated from pristine halloysite nanotubes and taking into account the formation
316 of the Hal-Si-v_1^- species ($-650.4 \text{ kJ mol}^{-1}$) are -597.4 and $-591.5 \text{ kJ mol}^{-1}$,
317 respectively. It is clear that these reactions are less thermodynamically favored
318 than the functionalizations on silicic surfaces bearing silicon-based tetrahedra
319 before any extraction of orthosilicic acid occurred. In the $\text{Hal-S}^v\text{A}^0$ structures,
320 the silicon atom from A, substituting the one lost as $\text{Si}(\text{OH})_4$, is not equiva-
321 lent to the other Si centers on the surface. As a matter of fact, none of these
322 last could form bonds with any molecule protruding outwards, since their SiO_3
323 tetrahedra have a cavity facing up and the fourth Si-O bond is formed with
324 the interlayer; conversely, the Si originating from A shows the SiO_3 tetrahedron
325 with the cavity facing downwards and is located just above the interlayer oxygen
326 atom bearing the negative charge, at a distance of 2.84 Å from it. Consequently,
327 the three Si-O-Si bond angles that realize on the edge of the functionaliza-
328 tion cannot have the typical value of 132° which characterize the other centers
329 on the surface, but their values (166° on the average) witness a much linear
330 arrangement. Likely for these reasons reactions (11) are endothermic.

331 Finally, the functionalization of halloysite inner lumen was also studied by
332 simulating the product of the following reactions, which would occur on the
333 untreated aluminum-based surface:



334 The two organosilanes react with three OH groups of the inner surface of hal-
335 loysite, resulting in its silanization. The optimized structures are reported in
336 Figure 6. It is to be noted that the silicon atom from A fits almost naturally on
337 the AlO hexagonal arrangement, with Si–O bond lengths equal to 1.68 Å on
338 the average, *i.e.* only 0.04 Å longer than the usual; further, the oxygen atoms
339 that coordinate the functionalization only slightly deviate from their bridged
340 position among two Al centers, showing a closure of 6° of the Al–O–Al angle
341 caused by an averaged Al–O bond elongation of 0.05 Å. In the case of APTES
342 functionalization a H₂N---HO hydrogen bond is formed with the aluminic sur-
343 face, while AEAPTMS prefers to maintain its intramolecular H-bond.

344 [Figure 6 about here.]

345 Both reactions (12) occur with a small release of energy, being $\Delta E = -99.9$
346 kJ mol^{-1} for A = APTES and $-61.3 \text{ kJ mol}^{-1}$ for A = AEAPTMS (which is lower
347 than the APTES case due to the absence of hydrogen bonds with the surface),
348 and there is, in alkaline environment, no other reaction with which they could
349 couple. By comparing these values with the reaction energies characterizing
350 the functionalizations occurring at the outer silicic surface, it is possible to
351 assess that the attachment of both APTES and AEAPTMS is preferred on the
352 modified Si external array. It is worth to note, however, that in the aluminum
353 surface the functionalization with APTES and AEAPTMS could occur directly
354 on the pristine Hal, without pretreatment.

355 4.2. *Experimental study of the functionalization efficiency*

356 The section on the computational investigations above report the results
357 about the selective functionalization of halloysite silicic surface, which exploit
358 the formation of active sites with hydroxyl groups, as a consequence of the alka-
359 line hydrolysis. It was showed that the silanization with 3-aminopropyltriethoxysilane
360 (APTES) and N-3-(trimethoxysilyl)propyl-ethylenediamine (AEAPTMS) of the
361 pretreated nanoclay is a thermodynamically favored reaction which allows one
362 to infer high efficiency. These aspects were investigated experimentally. Ther-
363 mogravimetric curves for untreated and treated Hal modified with APTES and

364 AEAPTMS are reported in Figure 7, whereas the thermogravimetric parameters
365 are reported in Table 3.

366 [Figure 7 about here.]

367 [Table 3 about here.]

368 As it can be observed, the mass residues at 800 °C of all the functionalized sam-
369 ples are lower than the final mass of pristine Hal and aHal (see Figure S7 of
370 Supporting Information). Since the organic matter undergoes thermal degrada-
371 tion in this range of temperature, the mass residue at 800 °C is decreasing with
372 increasing the amount of organic counterpart in each sample. Therefore, it can
373 be assessed that the functionalization with the organosilanes occurred. More-
374 over, MR_{800} for aHal-APTES and aHal-AEAPTMS are lower than the values for
375 Hal-APTES and Hal-AEAPTMS. These findings are related to a more efficient
376 grafting on the surface of the nanotubes after the treatment with the base. In
377 order to provide a more detailed evaluation of the grafting efficiency, the loading
378 of APTES on both Hal-APTES and aHal-APTES was calculated and it resulted
379 to be 9.5 and 10.9 wt%, respectively. Similarly, the grafting efficiency was cal-
380 culated to be 8.6 and 11.0 wt% for Hal-AEAPTMS and aHal-AEAPTMS. As a
381 consequence, the treatment of halloysite with sodium hydroxide effectively im-
382 proved the functionalization of its surface. Experimental evidences are in good
383 agreement with the computational results.

384 The attachment of the organosilane molecules onto halloysite was also inves-
385 tigated by FTIR spectroscopy. Figure 8 reports the spectra of Hal and alkaline
386 pretreated Hal before and after functionalization with APTES, as an example
387 (see Figure S8 of Supporting Information for the AEAPTMS case). It is worth
388 to observe that the positions of the typical bands of halloysite at 3695 and 3620
389 cm^{-1} are not affected by the grafting. Since these two bands are related to the
390 OH-stretching vibrations of the Al_2OH groups inside the lumen (Lisuzzo et al.,
391 2020b), any interactions arising between the aluminols groups and the organic
392 molecules could be disregarded. Most importantly, after the silanization is car-

393 ried out, some new bands appear in the 1800-1200 cm^{-1} range (magnifications
394 in Figure 8).

395 [Figure 8 about here.]

396 The deformation (scissoring) of NH_2 at 1561/1571 cm^{-1} , the deformation (scis-
397 soring) of CH_2 at 1490/1494 cm^{-1} , the deformation (wagging) of CH_2 at 1385/1387
398 cm^{-1} and the deformation (scissoring) of Si-CH at 1335/1339 cm^{-1} can be
399 clearly identified in the spectra of both Hal-APTES and aHal-APTES. Also,
400 APTES-modified samples exhibit some new FTIR peaks at around 2930 cm^{-1} ,
401 related to the CH_2 stretching vibration. These results confirm that the func-
402 tionalization was achieved (Yuan et al., 2008b).

403 Aimed at quantitatively evaluating the silanization of the the nanoclays,
404 XRF experiments were carried out and the Si/Al ratios for the samples before
405 and after the APTES and AEAPTMS functionalization were measured (result
406 reported in Table 4).

407 [Table 4 about here.]

408 It is clear that the Si/Al ratio increases due to the functionalization with the
409 organosilanes, which brings more Si atoms on the surface of the nanotubes. It
410 is worth to note that, in agreement with thermogravimetric data, the increase
411 is higher for aHal-APTES compared to Hal-APTES, and for aHal-AEAPTMS
412 compared to Hal-AEAPTMS. These results represent a further proof that the
413 functionalization of halloysite surface with the organosilanes is improved after
414 the alkaline pretreatment of the nanoclay with sodium hydroxide.

415 Finally, X-Ray Diffraction analysis was conducted in order to evaluate any
416 possible variations on the structure of halloysite nanotubes. Diffractograms are
417 reported in Figure 9.

418 [Figure 9 about here.]

419 The pattern of pristine Hal shows the typical reflections at $2\theta = 12.0, 20.4$ and
420 24.5° corresponding to the (001) basal spacing of 0.73 nm, (020)/(110) basal

spacing of 0.44 nm and (002) basal spacing of 0.35 nm, respectively (Barot et al., 2020). The presence of the first peak, which depends on the distance between the different layers of the nanotubes, indicates that the clay is in the 7Å-form and it is consequently dehydrated. After the treatment with NaOH and the functionalization with both APTES and AEAPTMS, the reflections are not displaced. This finding confirms that no intercalation occurred and that the functionalization of halloysite was carried out maintaining its morphological properties and its peculiar hollow nanotubular shape, in agreement with the literature (Ferrante et al., 2023).

5. Conclusions

In this work, a computational investigation based on density functional theory is conducted in order to have detailed atomistic insights about both the structural properties of halloysite nanotubes functionalized with organosilanes and also on the energetics of the reactions involved. It was found that the alkaline activation of the nanoclay, by pretreatment with a base, has deep effects on the grafting of organic molecules. Computational results suggested that the interaction with halloysite outer surface is thermodynamically favored when the nanotubes display silanols groups, which act as active sites for the nucleophilic attack on the silicon center of the organosilanes. In particular, the breaking of three Si–O bonds in three different Si centers of the silicic array provided the best conditions (i.e. number of active sites and high configurational flexibility) for the reaction with APTES and AEAPTMS, being the global process energies -1410.4 and -1438.8 kJ mol⁻¹, respectively. Moreover, this study also showed that the functionalization of the inner aluminum based surface of halloysite is less favored, by a thermodynamic point of view, being the calculated reaction energies -99.9 and -61.3 kJ mol⁻¹ for APTES and AEAPTMS attachment on the internal lumen. Aimed at supporting the computational findings, experimental analysis were also carried out. Thermogravimetry showed that the functionalization efficiency improved after the alkaline pretreatment of Hal, reaching the

450 highest values of 10.9 and 11.0 wt% for aHal-APTES and aHal-AEAPTMS.
451 The successful functionalization was confirmed by FTIR spectroscopy, which
452 allowed to identify the appearance of new bands typical of the organosilanes on
453 the spectra of the functionalized materials. X-Ray Fluorescence analysis fur-
454 therly proved that the amount of silicon atoms on halloysite surface increased
455 after the functionalization and, in particular, at a higher extent for the alkali ac-
456 tivated clay compared to the pristine Hal. Finally, XRD measurements showed
457 that the grafting did not alter the morphological properties of halloysite nan-
458 otubes, whose hollow nanotubular shape is preserved without any intercalation
459 in the interlayer space. On the whole, these results prove that the functional-
460 ization of halloysite is enhanced as a consequence of the alkaline pretreatment
461 and due to the creation of silanol active groups on the surface. Since the chem-
462 ical modification of halloysite surface is very often the first step before its use
463 in any technological application, this work opens up new perspectives about
464 the optimization of reactions and processes where aluminosilicates are involved,
465 especially in catalysis.

466 **Acknowledgements**

467 Funding are gratefully acknowledged by “SiciliAn MicronanOTecH Research
468 And innovation Center - SAMOTHRACE” (MUR, PNRR-M4C2, ECS_0000002),
469 spoke 3, Università degli Studi di Palermo, “S2-COMMs-Micro and Nanotech-
470 nologies for Smart & Sustainable Communities”. LL thanks the European Union
471 for co-financing within the FESR e FSE, PON “Ricerca e Innovazione 2014-2020
472 - DM 1062/2021”

References

Barot, T., Rawtani, D., & Kulkarni, P. (2020). Physicochemical and bio-
logical assessment of silver nanoparticles immobilized halloysite nanotubes-
based resin composite for dental applications. *Heliyon*, *6*, e03601.
doi:<https://doi.org/10.1016/j.heliyon.2020.e03601>.

- Blanco, I., Cicala, G., Latteri, A., Saccullo, G., El-Sabbagh, A. M. M., & Ziegmann, G. (147-153). Thermal characterization of a series of lignin-based polypropylene blends. *Journal of Thermal Analysis and Calorimetry*, *127*, 2017. doi:<https://doi.org/10.1007/s10973-016-5596-2>.
- Cataldo, S., Lazzara, G., Massaro, M., Muratore, N., Pettignano, A., & Riela, S. (2018). Functionalized halloysite nanotubes for enhanced removal of lead(ii) ions from aqueous solutions. *Appl. Clay Sci.*, *156*, 87–95. doi:<https://doi.org/10.1016/j.clay.2018.01.028>.
- De Silva, R. T., Pasbakhsh, P., Lee, S. M., & Kit, A. Y. (2015). ZnO deposited/encapsulated halloysite–poly(lactic acid) (PLA) nanocomposites for high performance packaging films with improved mechanical and antimicrobial properties. *Applied Clay Science*, *111*, 10–20. doi:<https://doi.org/10.1016/j.clay.2015.03.024>.
- Eichkorn, K., Weigend, F., Treutler, O., & Ahlrichs, R. (1997). Auxiliary basis sets for main row atoms and transition metals and their use to approximate coulomb potentials. *Theor. Chem. Acc.*, *97*, 119–124. doi:<https://doi.org/10.1007/s002140050244>.
- Ferrante, F., Armata, N., Cavallaro, G., & Lazzara, G. (2017). Adsorption studies of molecules on the halloysite surfaces: A computational and experimental investigation. *J Phys. Chem. C*, *121*, 2951–2958. doi:<https://doi.org/10.1021/acs.jpcc.6b12876>.
- Ferrante, F., Armata, N., & Lazzara, G. (2015). Modeling of the halloysite spiral nanotube. *J Phys. Chem. C*, *119*, 16700–16707. doi:<https://doi.org/10.1021/acs.jpcc.5b04281>.
- Ferrante, F., Bertini, M., Ferlito, C., Lisuzzo, L., Lazzara, G., & Duca, D. (2023). A computational and experimental investigation of halloysite silicic surface modifications after alkaline treatment. *Appl. Clay Sci.*, *232*, 106813. doi:<https://doi.org/10.1016/j.clay.2022.106813>.

- Frisch, M. J., Trucks, G. W., Schlegel, H. B., Scuseria, G. E., Robb, M. A., Cheeseman, J. R., Scalmani, G., Barone, V., Petersson, G. A., Nakatsuji, H., Li, X., Caricato, M., Marenich, A. V., Bloino, J., Janesko, B. G., Gomperts, R., Mennucci, B., Hratchian, H. P., Ortiz, J. V., Izmaylov, A. F., Sonnenberg, J. L., Williams-Young, D., Ding, F., Lipparini, F., Egidi, F., Goings, J., Peng, B., Petrone, A., Henderson, T., Ranasinghe, D., Zakrzewski, V. G., Gao, J., Rega, N., Zheng, G., Liang, W., Hada, M., Ehara, M., Toyota, K., Fukuda, R., Hasegawa, J., Ishida, M., Nakajima, T., Honda, Y., Kitao, O., Nakai, H., Vreven, T., Throssell, K., Montgomery, J. A., Jr., Peralta, J. E., Ogliaro, F., Bearpark, M. J., Heyd, J. J., Brothers, E. N., Kudin, K. N., Staroverov, V. N., Keith, T. A., Kobayashi, R., Normand, J., Raghavachari, K., Rendell, A. P., Burant, J. C., Iyengar, S. S., Tomasi, J., Cossi, M., Millam, J. M., Klene, M., Adamo, C., Cammi, R., Ochterski, J. W., Martin, R. L., Morokuma, K., Farkas, O., Foresman, J. B., & Fox, D. J. (2016). Gaussian 16 Revision C.01.
- Haw, T. T., Hart, F., Rashidi, A., & Pasbakhsh, P. (2020). Sustainable cementitious composites reinforced with metakaolin and halloysite nanotubes for construction and building applications. *Appl. Clay Sci.*, *188*, 105533. doi:<https://doi.org/10.1016/j.clay.2020.105533>.
- Hua, Y., Guo, T., Li, F., Fu, L., & Yang, H. (2023). The structure evolution of halloysite nanotubes during the acid leaching process: A molecular dynamics study. *Appl. Clay. Sci.*, *242*, 107021. doi:<https://doi.org/10.1016/j.clay.2023.107021>.
- Kryuchkova, M., Danilushkina, A., Lvov, Y., & Fakhrullin, R. (2016). Evaluation of toxicity of nanoclays and graphene oxide in vivo: A paramecium caudatum study. *Environ. Sci.*, *3*, 442–452. doi:<https://doi.org/10.1039/C5EN00201J>.
- Lisuzzo, L., Cavallaro, G., Milioto, S., & Lazzara, G. (2020a). Effects of halloysite content on the thermo-mechanical performances

- of composite bioplastics. *Applied Clay Science*, *185*, 105416. doi:<https://doi.org/10.1016/j.clay.2019.105416>.
- Lisuzzo, L., Cavallaro, G., Milioto, S., & Lazzara, G. (2022). Halloysite nanotubes as nanoreactors for heterogeneous micellar catalysis. *J. Colloid Interface Sci.*, *608*, 424–434. doi:<https://doi.org/10.1016/j.jcis.2021.09.146>.
- Lisuzzo, L., Cavallaro, G., Parisi, F., Milioto, S., & Lazzara, G. (2019). Colloidal stability of halloysite clay nanotubes. *Ceramics International*, *45*, 2858–2865. doi:<https://doi.org/10.1016/j.ceramint.2018.07.289>.
- Lisuzzo, L., Wicklein, B., Dico, G. L., Lazzara, G., Real, G. d., Aranda, P., & Ruiz-Hitzky, E. (2020b). Functional biohybrid materials based on halloysite, sepiolite and cellulose nanofibers for health applications. *Dalton Trans.*, *49*, 3830–3840. doi:<https://doi.org/10.1039/C9DT03804C>.
- Liu, M., Chang, Y., Yang, J., You, Y., He, R., Chen, T., & Zhou, C. (2016). Functionalized halloysite nanotube by chitosan grafting for drug delivery of curcumin to achieve enhanced anticancer efficacy. *J. Mater. Chem. B*, *4*, 2253–2263. doi:<https://doi.org/10.1039/C5TB02725J>.
- Liu, M., Fakhrullin, R., Novikov, A., Panchal, A., & Lvov, Y. (2019). Tubule nanoclay-organic heterostructures for biomedical applications. *Macromol. Biosci.*, *19*, 1800419. doi:<https://doi.org/10.1002/mabi.201800419>.
- Liu, T., Zhang, J., Ouyang, P., Fu, L., & Yang, H. (2021). The relation between nanotube diameter, length and surface area and pore volume of multi-walled spiral halloysite nanotubes: A theoretical study. *Appl. Clay Sci.*, *215*, 106303. doi:<https://doi.org/10.1016/j.clay.2021.106303>.
- Lvov, Y., Wang, W., Zhang, L., & Fakhrullin, R. (2016). Halloysite clay nanotubes for loading and sustained release of functional compounds. *Advanced Materials*, *28*, 1227–1250. doi:<https://doi.org/10.1002/adma.201502341>.
- Makó, E., Dódon, I., Pekker, P., Pósfai, M., Kovács, A., Ható, Z., & Kristóf, T. (2020). Nanoscale structural and morphologi-

- cal features of kaolinite nanoscrolls. *Appl. Clay Sci.*, *198*, 105800. doi:<https://doi.org/10.1016/j.clay.2020.105800>.
- Massaro, M., Licandro, E., Cauteruccio, S., Lazzara, G., Liotta, L. F., Notarbartolo, M., Raymo, F. M., Sánchez-Espejo, R., Viseras-Iborra, C., & RIELA, S. (2022). Nanocarrier based on halloysite and fluorescent probe for intracellular delivery of peptide nucleic acids. *J. Colloid Interface Sci.*, *620*, 221–233. doi:<https://doi.org/10.1016/j.jcis.2022.03.151>.
- Mehdizadeh, M., Sadjadi, S., Poater, A., Mansouri, A., & Bahri-Laleh, N. (2022). Molecular modelling aided catalyst design for pao oils hydrofinishing. *J. Mol. Liq.*, *352*, 118675. doi:<https://doi.org/10.1016/j.molliq.2022.118675>.
- Panchal, A., Rahman, N., Konnova, S., Fakhrullin, R., Zhang, D., Blake, D., John, V., Ivanov, E., & Lvov, Y. (2020). Clay nanotube liquid marbles enhanced with inner biofilm formation for the encapsulation and storage of bacteria at room temperature. *ACS Appl. Nano Mater.*, *3*, 1263–1271. doi:<https://doi.org/10.1021/acsanm.9b02033>.
- Pasbakhsh, P., Churchman, G. J., & Keeling, J. L. (2013). Characterisation of properties of various halloysites relevant to their use as nanotubes and microfibre fillers. *Applied Clay Science*, *74*, 47–57. doi:<https://doi.org/10.1016/j.clay.2012.06.014>.
- Rozhina, E., Batasheva, S., Miftakhova, R., Yan, X., Vikulina, A., Volodkin, D., & Fakhrullin, R. (2021). Comparative cytotoxicity of kaolinite, halloysite, multiwalled carbon nanotubes and graphene oxide. *Appl. Clay Sci.*, *205*, 106041. doi:<https://doi.org/10.1016/j.clay.2021.106041>.
- Sadjadi, S., & Atai, M. (2018). Ternary hybrid system of halloysite nanotubes, polyacrylamides and cyclodextrin: An efficient support for immobilization of pd nanoparticles for catalyzing coupling reaction. *Appl. Clay Sci.*, *153*, 78–89. doi:<https://doi.org/10.1016/j.clay.2017.12.013>.

- Sadjadi, S., Hosseinejad, T., Malmir, M., & Heravi, M. M. (2017). Cu@furfural imine-decorated halloysite as an efficient heterogeneous catalyst for promoting ultrasonic-assisted a₃ and ka₂ coupling reactions: A combination of experimental and computational study. *New J. Chem.*, *41*, 13935–13951. doi:<https://doi.org/10.1039/C7NJ02272G>.
- Schaefer, A., Horn, H., & Ahlrichs, R. (1992). Fully optimized contracted gaussian-basis sets for atoms li to kr. *J. Chem. Phys.*, *97*, 2571–2577. doi:<https://doi.org/10.1063/1.463096>.
- Shankar, S., Kasapis, S., & Rhim, J.-W. (2018). Alginate-based nanocomposite films reinforced with halloysite nanotubes functionalized by alkali treatment and zinc oxide nanoparticles. *International Journal of Biological Macromolecules*, *118*, 1824–1832. doi:<https://doi.org/10.1016/j.ijbiomac.2018.07.026>.
- Sidorenko, A. Y., Kravtsova, A. V., Aho, A., Heinmaa, I., Volcho, K. P., Salakhutdinov, N. F., Agabekov, V. E., & Murzin, D. Y. (2018). Acid-modified halloysite nanotubes as a stereoselective catalyst for synthesis of 2H-chromene derivatives by the reaction of isopulegol with aldehydes. *ChemCatChem*, *10*, 3950–3954. doi:<https://doi.org/10.1002/cctc.201800974>.
- Sidorenko, A. Y., Kurban, Y. M., Aho, A., Ihnatovich, Z. V., Kuznetsova, T. F., Heinmaa, I., Murzin, D. Y., & Agabekov, V. E. (2021). Solvent-free synthesis of tetrahydropyran alcohols over acid-modified clays. *mol. Catal.*, *499*, 111306. doi:<https://doi.org/10.1016/j.mcat.2020.111306>.
- Spepi, A., Duce, C., Pedone, A., Presti, D., Rivera, J.-G., Ierardi, V., & Tiné, M. R. (2016). Experimental and DFT characterization of halloysite nanotubes loaded with salicylic acid. *J. Phys. Chem. C*, *120*, 26759–26769. doi:<https://doi.org/10.1021/acs.jpcc.6b06964>.
- Takahara, Y. K., Ikeda, S., Ishino, S., Tachi, K., Ikeue, K., Sakata, T., Hasegawa, T., Mori, H., Matsumura, M., & Ohtani, B. (2005). Asym-

- metrically modified silica particles: A simple particulate surfactant for stabilization of oil droplets in water. *J. Am. Chem. Soc.*, *127*, 6271–6275. doi:<https://doi.org/10.1021/ja043581r>.
- Vergaro, V., Lvov, Y. M., & Leporatti, S. (2012). Halloysite clay nanotubes for resveratrol delivery to cancer cells. *Macromol. Biosci.*, *12*, 1265–1271. doi:<https://doi.org/10.1002/mabi.201200121>.
- Wang, Q., Zhang, J., & Wang, A. (2013). Alkali activation of halloysite for adsorption and release of ofloxacin. *Appl. Surf. Sci.*, *287*, 54–61. doi:<https://doi.org/10.1016/j.apsusc.2013.09.057>.
- Yuan, P., Southon, P. D., Liu, Z., Green, M. E. R., Hook, J. M., Antill, S. J., & Kepert, C. J. (2008a). Functionalization of halloysite clay nanotubes by grafting with γ -aminopropyltriethoxysilane. *J. Phys. Chem. C*, *112*, 15742–15751. doi:<https://doi.org/10.1021/jp805657t>.
- Yuan, P., Southon, P. D., Liu, Z., Green, M. E. R., Hook, J. M., Antill, S. J., & Kepert, C. J. (2008b). Functionalization of halloysite clay nanotubes by grafting with γ -aminopropyltriethoxysilane. *J. Phys. Chem. C*, *112*, 15742–15751. doi:<https://doi.org/10.1021/jp805657t>.
- Zhang, H., Cheng, C., Song, H., Bai, L., Cheng, Y., Ba, X., & Wu, Y. (2019a). A facile one-step grafting of polyphosphonium onto halloysite nanotubes initiated by Ce(IV). *Chem. Commun.*, *55*, 1040–1043. doi:<http://dx.doi.org/10.1039/C8CC08667B>.
- Zhang, Y., Bai, L., Cheng, C., Zhou, Q., Zhang, Z., Wu, Y., & Zhang, H. (2019b). A novel surface modification method upon halloysite nanotubes: A desirable cross-linking agent to construct hydrogels. *Appl. Clay Sci.*, *182*, 105259–105267. doi:<https://doi.org/10.1016/j.clay.2019.105259>.
- Zhao, Y., & Truhlar, D. G. (2006). A new local density functional for main-group thermochemistry, transition metal bonding, thermochemical

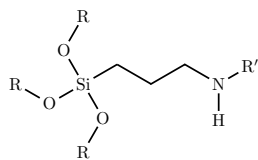
kinetics, and noncovalent interactions. *J. Chem. Phys.*, *125*, 194101.
doi:<https://doi.org/10.1063/1.2370993>.

Zhu, K., Duan, Y., Wang, F., Gao, P., Jia, H., Ma, C., & Wang, C. (2017). Silane-modified halloysite/Fe₃O₄ nanocomposites: Simultaneous removal of Cr(VI) and Sb(V) and positive effects of Cr(VI) on Sb(V) adsorption. *Chem. Eng. J.*, *311*, 236–246. doi:<https://doi.org/10.1016/j.cej.2016.11.101>.

Zsirka, B., Horváth, E., Szabó, P., Juzsakova, T., Szilágyi, R. K., Fertig, D., Makó, E., Varga, T., Kónya, Z., Kukovecz, A., & Kristóf, J. (2017). Thin-walled nanoscrolls by multi-step intercalation from tubular halloysite-10Å and its rearrangement upon peroxide treatment. *Appl. Surf. Sci.*, *399*, 245–254. doi:<https://doi.org/10.1016/j.apsusc.2016.12.053>.

List of Figures

1	The (3-aminopropyl)triethoxysilane (APTES) and the N-3-(trimethoxysilyl)propyl ethylenediamine (AEAPTMS) molecules.	28
2	The DFT optimized geometries of the Hal-S ¹ A ² system, with A = APTES (a) and A = AEAPTMS (b), representing the product of process (2). For the sake of visualization, only the functionalized silicic portion of the model is shown in the main picture, while the whole molecular model is reported in stick representation in the upper corner. Color code: H-pink, C-grey, N-blue, O-red, Si-green. Hydrogen bonds are represented by dashed lines.	29
3	The DFT optimized geometries of the Hal-S ² A ¹ system, with A = APTES (a) and A = AEAPTMS (b), representing the product of process (6). For the sake of visualization, only the functionalized silicic portion of the model is shown in the main picture, while the whole molecular model is reported in stick representation in the upper corner. Color code: H-pink, C-grey, N-blue, O-red, Si-green. Hydrogen bonds are represented by dashed lines.	30
4	The DFT optimized geometries of the Hal-S ³ A ⁰ system, with A = APTES (a) and A = AEAPTMS (b), representing the product of process (9). For the sake of visualization, only the functionalized silicic portion of the model is shown in the main picture, while the whole molecular model is reported in stick representation in the upper corner. Color code: H-pink, C-grey, N-blue, O-red, Si-green. Hydrogen bonds are represented by dashed lines. The peculiar SiO configuration is depicted in the inset, with the Si originating from the functionalization evidenced in red.	31
5	The DFT optimized geometries of the Hal-S ^v A ⁰ system, with A = APTES (a) and A = AEAPTMS (b), representing the product of process (11). For the sake of visualization, only a portion of the functionalized silicic surface is shown. Color code: H-pink, C-grey, N-blue, O-red, Si-green.	32
6	The DFT optimized geometries of the Hal-Al-A ⁰ system, with A = APTES (a) and A = AEAPTMS (b), representing the product of process (12). For the sake of visualization, only a portion of the functionalized aluminic surface is shown. Color code: H-pink, C-grey, N-blue, O-red, Al-cyan. In the case of APTES the picture is enlarged to allow appreciating the details of the anchoring configuration.	33
7	Thermogravimetric curves of APTES and AEAPTMS functionalized Hal and alkaline pretreated Hal.	34
8	FTIR spectra of Hal and alkaline pretreated Hal before and after functionalization with APTES.	35
9	Diffraction patterns of Hal, aHal, Hal-APTES, aHal-APTES, Hal-AEAPTMS and aHal-AEAPTMS.	36



APTES: R'=H R=Et

AEAPTMS: R'=(CH₂)₂NH₂ R=Me

Figure 1: The (3-aminopropyl)triethoxysilane (APTES) and the N-3-(trimethoxysilyl)propyl ethylenediamine (AEAPTMS) molecules.

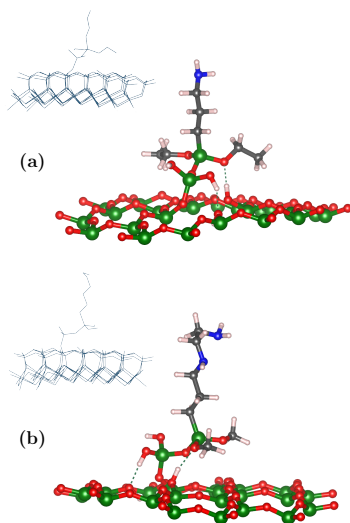


Figure 2: The DFT optimized geometries of the Hal-S¹A² system, with A = APTES (a) and A = AEAPTMS (b), representing the product of process (2). For the sake of visualization, only the functionalized silicic portion of the model is shown in the main picture, while the whole molecular model is reported in the upper corner. Color code: H-pink, C-grey, N-blue, O-red, Si-green. Hydrogen bonds are represented by dashed lines.

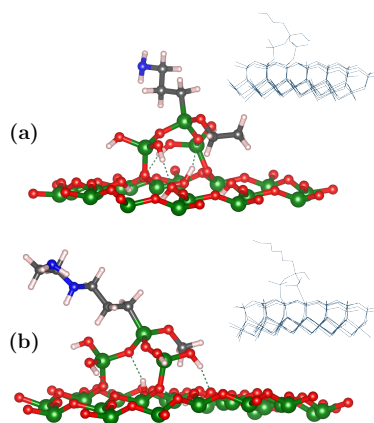


Figure 3: The DFT optimized geometries of the Hal-S²A¹ system, with A = APTES (a) and A = AEAPTMS (b), representing the product of process (6). For the sake of visualization, only the functionalized silicic portion of the model is shown in the main picture, while the whole molecular model is reported in stick representation in the upper corner. Color code: H-pink, C-grey, N-blue, O-red, Si-green. Hydrogen bonds are represented by dashed lines.

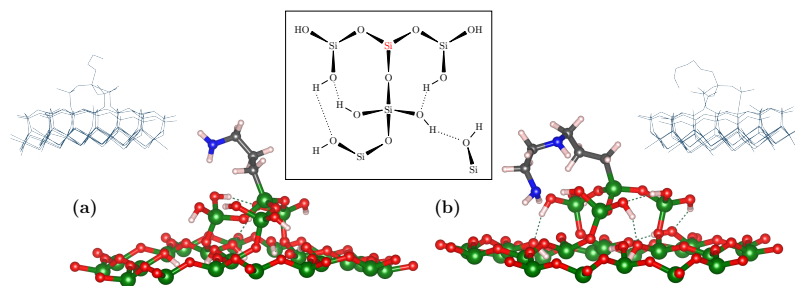


Figure 4: The DFT optimized geometries of the Hal-S³A⁰ system, with A = APTES (a) and A = AEAPTMS (b), representing the product of process (9). For the sake of visualization, only the functionalized silicic portion of the model is shown in the main picture, while the whole molecular model is reported in stick representation in the upper corner. Color code: H-pink, C-grey, N-blue, O-red, Si-green. Hydrogen bonds are represented by dashed lines. The peculiar SiO configuration is depicted in the inset, with the Si originating from the functionalization evidenced in red.

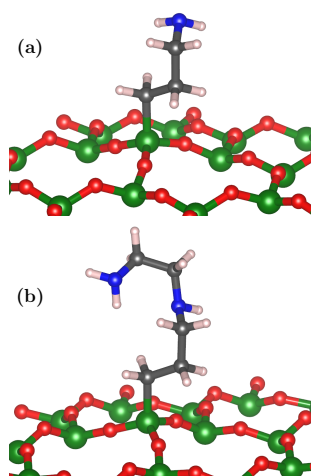


Figure 5: The DFT optimized geometries of the $\text{Hal-S}^v\text{A}^0$ system, with $\text{A} = \text{APTES}$ (a) and $\text{A} = \text{AEAPTMS}$ (b), representing the product of process (11). For the sake of visualization, only a portion of the functionalized silicic surface is shown. Color code: H-pink, C-grey, N-blue, O-red, Si-green.

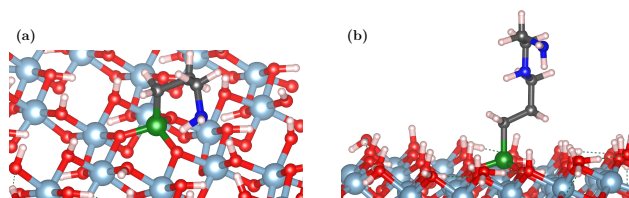


Figure 6: The DFT optimized geometries of the Hal-Al-A⁰ system, with A = APTES (a) and A = AEAPTMS (b), representing the product of process (12). For the sake of visualization, only a portion of the functionalized aluminic surface is shown. Color code: H-pink, C-grey, N-blue, O-red, Al-cyan. In the case of APTES the picture is enlarged to allow appreciating the details of the anchoring configuration.

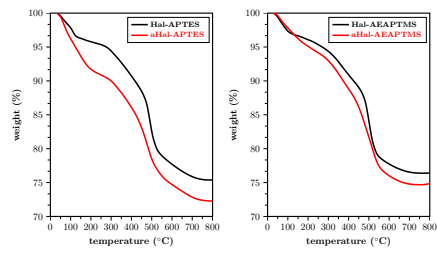


Figure 7: Thermogravimetric curves of APTES and AEAPTMS functionalized Hal and alkaline pretreated Hal.

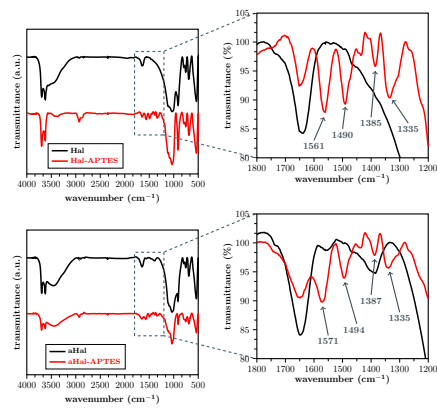


Figure 8: FTIR spectra of Hal and alkaline pretreated Hal before and after functionalization with APTES.

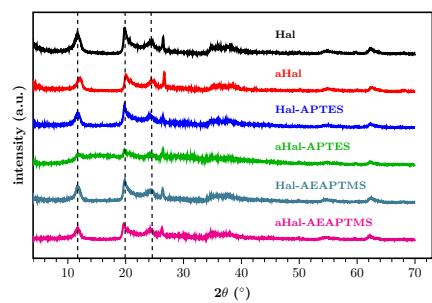


Figure 9: Diffractograms of Hal, aHal, Hal-APTES, aHal-APTES, Hal-AEAPTMS and aHal-AEAPTMS.

List of Tables

1	Global energies (ΔE) for all the investigated functionalization reactions. R = Et for APTES, R = Me for AEAPTMS.	38
2	Reaction energies for the $\text{Hal-Si}(2\text{ m,m})^{\text{m}^-} + \text{A} \longrightarrow \text{Hal-S}^{\text{m}}\text{A}^{3-\text{n}} + \text{nROH}$ functionalization.	39
3	Thermogravimetric parameters (wt%) of pristine Hal and alkaline pretreated Hal functionalized with APTES and AEAPTMS.	40
4	Si/Al ratios for Hal and alkaline pretreated Hal before and after functionalization with APTES and AEAPTMS.	41

Table 1: Global energies (ΔE) for all the investigated functionalization reactions. R = Et for APTES, R = Me for AEAPTMS.

reaction	$\Delta E/\text{kJ mol}^{-1}$	
	A=APTES	A=AEAPTMS
$\text{Hal} + 2 \text{H}_2\text{O} + \text{OH}^- + \text{A} \longrightarrow \text{Hal-S}^1\text{A}^2 + \text{ROH}$	-727.5	-714,4
$\text{Hal} + 2 \text{H}_2\text{O} + \text{OH}^- + \text{A} \longrightarrow \text{Hal-S}^1\text{A}^1 + 2 \text{ROH}$	-732.5	-719.9
$\text{Hal} + 2 \text{H}_2\text{O} + \text{OH}^- + \text{A} \longrightarrow \text{Hal-S}^1\text{A}^0 + 3 \text{ROH}$	-509.3	-507.8
$\text{Hal} + 4 \text{H}_2\text{O} + 2 \text{OH}^- + \text{A} \longrightarrow \text{Hal-S}^2\text{A}^1 + 2 \text{ROH}$	-1125.8	-1130.7
$\text{Hal} + 4 \text{H}_2\text{O} + 2 \text{OH}^- + \text{A} \longrightarrow \text{Hal-S}^2\text{A}^0 + 3 \text{ROH}$	-972.5	-964.3
$\text{Hal} + 6 \text{H}_2\text{O} + 3 \text{OH}^- + \text{A} \longrightarrow \text{Hal-S}^3\text{A}^0 + 3 \text{ROH}$	-1410.4	-1438.8
$\text{Hal} + 3 \text{H}_2\text{O} + \text{OH}^- + \text{A} \longrightarrow \text{Hal-S}^v\text{A}^0 + \text{Si}(\text{OH})_4 + 3 \text{ROH}$	-587.4	-591.5
$\text{Hal} + \text{A} \longrightarrow \text{Hal-Al-A}^0 + 3 \text{ROH}$	-99.9	-61.3

Table 2: Reaction energies for the $\text{Hal-Si}(2\text{ m,m})^{\text{m-}} + \text{A} \longrightarrow \text{Hal-S}^{\text{m}}\text{A}^{3-\text{n}} + \text{nROH}$ functionalization.

Reaction energy/ kJ mol^{-1}						
A = APTES				A = AEAPTMS		
m	n			n		
	1	2	3	1	2	3
1	-31.0	-36.0	187.2	-17.9	-23.4	188.8
2		-6.0	147.3		-11.0	155.4
3			-15.0			-43.3

Table 3: Thermogravimetric parameters (wt%) of pristine Hal and alkaline pretreated Hal functionalized with APTES and AEAPTMS.

Sample	ML₁₅₀	MR₈₀₀	MD₈₀₀
Hal ^a	2.2	83.8	10.0
aHal ^a	3.4	83.4	13.2
Hal-APTES	3.7	75.3	21.0
aHal-APTES	6.3	72.3	21.4
Hal-AEAPTMS	3.3	76.4	20.2
aHal-AEAPTMS	3.8	74.8	21.5

^a See (Ferrante et al., 2023)

Table 4: Si/Al ratios for Hal and alkaline pretreated Hal before and after functionalization with APTES and AEAPTMS.

Sample	Si/Al ratio
Hal	1.28 ^a
aHal	1.27 ^a
Hal-APTES	1.49
aHal-APTES	1.58
Hal-AEAPTMS	1.43
aHal-AEAPTMS	1.65

^a See (Ferrante et al., 2023)