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Hydrogel based on patch halloysite nanotubes: A rheological investigation



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<i>Keywords:</i> Halloysite nanotubes Rheology Hydrogel Patch halloysite	The rheological behaviour of Patch halloysite nanotubes (PT_Hal) was investigated here. Their peculiar morphology shows longer and thinner nanotubes and gives rise to the formation of gel-like systems that are not evidenced in halloysite from other natural sources. According to frequency sweep tests, PT_Hal possesses solid-like characteristics even at low concentrations, suggesting that the material is highly structured. Interestingly, flow ramp analysis evidenced two distinct behaviours based on the clay concentration: a yield stress was detected only from 0.75 wt%, indicating the sample's ability to resist deformation or breaking. Furthermore, the study investigated the influence of ionic strength, revealing that the addition of salt did not significantly affect the gel's properties of this clay. Accordingly, in this work we propose a new hydrogel system based on green nanoclays that can be suitable for industrial and biological applications as well as for cultural heritage.

1. Introduction

Halloysite is a naturally occurring nanoclay with a unique hollow tubular structure. Generally, the length of halloysite nanotubes is approximately 1 μ m, while the external and internal diameters vary within the ranges of 20 to 200 nm and 10 to 70 nm, respectively. Its properties and characteristics, such as the sizes, specific surface and polydispersity are influenced by the geological deposit from which they are sourced [1]. Among them, the Patch halloysite (PT_Hal), sourced from Western Australia, is acknowledged as a highly pure variety of halloysite (content of ca. 96 %) [2]. These nanotubes exhibit thinner tubes and an exceptional length extending up to 30 μ m. Moreover, they also stand up for their fibrous structure forming clusters that appear as patches. It is worth noting that the external and internal surfaces of halloysite have distinct chemical properties, with the former being negatively charged and the latter being positively charged across a wide pH range [3].

This characteristic enables the loading of negative molecules into the inner lumen and the interaction of positive active species with the outer surface. Halloysite also represents a versatile nanofiller that can be incorporated into biopolymeric matrices [4–6], obtaining green nanocomposite materials suitable for a wide range of applications, including medicine [7–10], cosmetics [11,12], packaging [13–15] and environmental remediation [16–25]. Moreover, halloysite nanotubes are

effective as catalytic supports [26–32], additives for metal batteries [33,34] and nanocarriers for functional molecules with biological and chemical activities [35–39].

Most research studies exploring the use of halloysite nanotubes as nanomaterials have focused on halloysite derived from one of the primary sources, typically New Zealand or USA, but its properties can differ based on its origin, owing to the distinct local conditions under which it was formed [40]. For instance, Makaremi et al. [40] demonstrated that shorter nanotubes showed superior capacity in encapsulating salicylic acid within their hollow structure compared to patchy and longer halloysite nanotubes as they require additional time and energy to have the active molecule loaded in the tubes. On the other hand, it is known that halloysite does not form gel itself. Actually, Luo et al. [41] proved that the sol–gel transition occurs at a concentration of ca. 40 wt% and the aqueous dispersion exhibits a pH-induced gelation when hydrochloric acid is added, due to the reduced repulsive electrostatic forces between the nanotubes.

Thus, our study wants to point out that, thanks to its peculiar fibrous structure, Patch has different rheological features with respect to halloysite from other natural sources. Accordingly, a rheological study on patch halloysite samples at different concentrations is presented here.

In addition, literature reports that the colloidal stability of nanoclays can be influenced by several factors, including the addition of polymers [42,43] as well as the ionic strength [44]. Moreover, changing the ionic

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strength in a system can also have significant effects on gel-like structures [45,46], such as swelling [47], increased stability [48] or viscosity due to stronger interactions with the components of the gel [49]. Since, according to the literature, NaCl concentration affects the settling velocity of halloysite dispersions, we also investigated its influence in the PT_Hal hydrogels in this study.

2. Materials and methods

2.1. Materials

Ultra HalloPure Halloysite (UP_Hal) was a gift by I-Minerals Inc. mined in the geological deposit of Latah County.

(NW Idaho, North America) and Patch Halloysite (PT_Hal) (Kalgoorie, Western Australia) is a kind donation provided by Dr. Keith Norrish from his collection and research on Patch (CSIRO Soils, Adelaide). NaCl (>99 %, CAS 7647–14-5) and NaOH pellets (\geq 98 %, CAS 1310–73-2) were Sigma Aldrich products.

2.2. Patch halloysite purification

The separation of coarse impurities was performed by following a procedure reported elsewhere for PT_Hal samples [1,2]. In particular, 5 g of PT_Hal were grounded in a mortar. The obtained powder was dispersed in deionized water and the pH was set between 7.5 and 8 by using a sodium hydroxide solution 0.1 M. Several cycles of sonication and washing in 1.5 L of water were carried out to purify the sample. After that, the dispersion was filtered by a peristaltic pump to exclude coarse residuals and dried at 50 °C for 48 h.

2.3. Hydrogel preparation

Patch halloysite, after purification, was dispersed in deionized water at different concentrations: 0.005, 0.1, 0.5, 0.75, 1 and 2 wt%. Then, samples were sonicated for 15 min and stirred for 48 h to avoid the formation of aggregates.

The same procedure was performed to prepare UP_Hal samples for comparison at 1 wt%.

Patch halloysite nanotubes were added to NaCl aqueous solutions at four different concentrations $(10^{-1}, 10^{-2}, 10^{-3} \text{ and } 10^{-4} \text{ M})$ and the obtained dispersions were sonicated for 15 min and magnetically stirred for 2 days.

2.4. Methods

2.4.1. Thermogravimetric analysis (TGA)

Thermogravimetric experiments were carried out using a Q5000 IR apparatus (TA Instruments) under inert atmosphere using nitrogen flows of 25 and 10 cm³ min⁻¹ for the sample and the balance, respectively. Samples were heated in a platinum pan from room temperature to 800 °C with a scanning rate of 20 °C min⁻¹.

2.4.2. X-ray Fluorescence (XRF)

X-ray Fluorescence (XRF) Spectrometry was performed using an Olympus Innov X DS-2000 Delta Standard Alloy XRF Handheld Analyzer operating in the Alloy Plus analysis mode.

2.4.3. ζ-Potential

 ζ -Potential measurements were performed by means of a Zetasizer NANO-ZS (Malvern Instruments) under isothermal conditions at 25.0 \pm 0.1 °C. The tests were carried out by using a disposable folded capillary cell and the concentration of PT_Hal dispersion was 0.001 wt%. The ζ -potential was calculated using the Smoluchowski approximation, which is considered an accurate approximation for particles larger than 0.2 μm in aqueous solutions. All experiments were replicated three times, and the average values were reported.

2.4.4. X-ray diffraction (XRD)

The patterns were obtained from an X-ray diffractometer (Rigaku, MiniFlex) with a CuKa radiation source including a nickel filter and working at 40 kV and 15 mA. The wavelength of the X-ray beam was 1.5406 Å, and the layer spacing of the samples was calculated by the Bragg's equation, which can be expressed as

$$\lambda = 2dsin\theta$$
 (1)

where θ is the angle that the outgoing beam forms with the crystalline layer, λ is the wavelength of the radiation, d is the distance between two adjacent layers and n can be 1,2 or 3.

The angle for scanning ranged from 2° to 70° with a rate of 20° min⁻¹ and a step of 0.02° .

2.4.5. Scanning electron microscopy

The ultra-high resolution field emission scanning electron microscope (FE-SEM, Hitachi SU8010) was used to observe the morphologies of halloysite nanotubes. A thin layer of platinum was applied to coat the samples in order to avoid electrostatic charging. Additionally, the morphology of halloysite nanotubes was also observed using the transmission mode (STEM) of the same FE-SEM instrument with a voltage of 30 kv.

2.4.6. Rheology

The rheological characterization of PT_Hal dispersions was studied using a Discovery HR-1 rheometer (TA Instruments) with a parallel plate geometry (800–1000 μ m gap and 20 mm diameter). Amplitude sweep, frequency sweep and flow ramp stress tests were performed after conditioning the sample at 25 °C. The amplitude sweep tests were conducted at a constant angular frequency of 100 rad s⁻¹ while increasing the oscillation strain from 0,001 % and 100 %. For frequency sweep tests, the angular frequency was ranged from 0.1 to 100 rad s⁻¹ at a fixed strain amplitude of 0.1 %. Flow ramp tests were instead performed increasing the stress from 0 to 50 Pa.

3. Results and discussion

3.1. Patch purification

XRF measurement allowed to obtain the chemical composition of UP_Hal and PT_Hal before and after the purification. As reported in Table 1, PT_Hal not purified (PT_Hal_NP) presents lower silicon content than the same sample after purification (PT_Hal_P) which has an enrichment of 6 % in silicon. Moreover, PT_Hal_P shows a decrease of the amount of Ti, Mg, Cr and Mn. On the other hand, iron content is nearly constant after purification, and it is present in a minimal amount.

It has been previously reported [2,50] that patch halloysite samples usually contain less than 1 % of iron. Actually, we found a value of 2.52 % in PT_Hal_NP that can be attributed to the different levels of iron oxide contamination. As a matter of fact, Churchman et al. [50] stated that, even if iron is not part of the structure, it can influence the size of halloysite particles by restraining crystal growth. The substitution of the smaller Al^{3+} ion with the larger Fe^{3+} ion in the octahedral sheet of

Table 1

Chemical composition of UP_Hal and PT_Hal samples before and after the purification.

-							
Sample	Al/wt %	Si/wt %	Mg/ wt%	Fe/wt %	Ti/wt %	Cr/wt %	Mn∕ wt%
UP_Hal	$\begin{array}{c} 41.7 \pm \\ 0.4 \end{array}$	$\begin{array}{c} 52.1 \\ \pm \ 0.4 \end{array}$	1.4 ± 0.6	$\begin{array}{c} 3.89 \\ \pm \ 0.09 \end{array}$	$\begin{array}{c} \textbf{0.8} \pm \\ \textbf{0.1} \end{array}$	ND	$\begin{array}{c} 0.08 \\ \pm \ 0.02 \end{array}$
PT_Hal_P	$\begin{array}{c} 41.1 \ \pm \\ 0.3 \end{array}$	$\begin{array}{c} 53.5 \\ \pm \ 0.3 \end{array}$	$\begin{array}{c} 1.2 \\ \pm \ 0.4 \end{array}$	$\begin{array}{c} 2.99 \\ \pm \ 0.06 \end{array}$	$\begin{array}{c} \textbf{0.47} \\ \pm \textbf{0.09} \end{array}$	$\begin{array}{c} 0.18 \\ \pm 0.02 \end{array}$	ND
PT_Hal_NP	$\begin{array}{c} 41.05 \\ \pm \ 0.4 \end{array}$	$\begin{array}{c} 46.8 \\ \pm \ 0.4 \end{array}$	$\begin{array}{c} 2.2 \\ \pm \ 0.6 \end{array}$	$\begin{array}{c} 2.52 \\ \pm \ 0.07 \end{array}$	$\begin{array}{c} \textbf{6.7} \pm \\ \textbf{0.2} \end{array}$	$\begin{array}{c} 0.20 \\ \pm 0.03 \end{array}$	$\begin{array}{c} 0.07 \\ \pm \ 0.01 \end{array}$

halloysite can also happen with high levels of Fe_2O_3 (ca. 4 %) leading to more planar layers as in kaolinite. This is reflected by the fact that halloysite with long tubular particles tends to contain low Fe content whereas shorter and approximately platy particles often contain high levels of structural Fe, as demonstrated by the results reported in Table 1 for UP_Hal. Moreover, besides the amount of iron, it showed very similar chemical composition compared to PT_Hal. In general, the obtained XRF results are also in agreement with literature data reported elsewhere [51] for halloysite nanotube samples.

Thermogravimetric curves of UP_Hal, purified and not purified PT_Hal are reported in Fig. 1. All the samples exhibited the two mass losses typical of halloysite: the first one until 150 °C and the second one in the range between 350 and 550 °C, respectively related to physically adsorbed water molecules and the dehydroxylation of aluminium inner sheets. The residual mass at 800 °C and the mass losses at each step (30-150 °C and 350-550 °C) are reported in Supporting Information (Table S1). Concerning the first mass loss, UP Hal showed the lowest value (1.17 %) compared to the PT Hal samples. Furthermore, the main difference between the two samples of Patch halloysite is related to the percentage of the first mass loss and consequently the residual mass at 800 °C. It is equal to 82 % for the purified PT Hal and 79 % for the not purified PT Hal. During the purification protocol, impurities have been washed away and consequently, we observed a lower value of residual mass in PT_Hal_P. On the other hand, the second mass loss is comparable for the three investigated samples.

3.2. Comparison between PT_Hal and UP_Hal

Before deeply analysing the rheological behavior of these peculiar clays, we focused our attention on the differences occurring between them.

A morphological characterization of PT_Hal and UP_Hal is presented in Fig. 2. SEM micrographs reveal that PT_Hal exhibits much greater length and thinness of the tubes. They are characterized by outer diameters ranging from 40 to 55 nm, inner diameters between 12 and 22 nm and lengths ranging from 200 to 30,000 nm. Moreover, PT_Hal tubes are entangled and bent, forming a structure similar to a bird nest. Since the walls are very thin, they are susceptible to breakage into shorter fragments, but still intact tubes can be observed as bundles of tightly packed parallel tubes.

On the other hand, UP_Hal, possesses shorter and stubby tubes ranging in length from 100 nm to $2 \mu m$, while the outer diameter ranges between 100 and 200 nm and the inner diameter between 10 and 70 nm (Fig. 2.(g)).

X-ray diffraction patterns of both UP_Hal and PT_Hal exhibited the typical diffraction pattern of dehydrated halloysite (Fig. S1). The



Fig. 1. Thermogravimetric curves of UP_Hal, non-purified (PT_Hal_NP) and purified (PT_Hal_P) Patch Halloysite.

dehydration of Patch was previously reported [52] by following the continuous collection of XRD patterns whilst in situ heating and it was showed that the dehydration produces some irregular d-spacings probably due to the interstratification of hydrated and dehydrated halloysite layers and the interference between the regular flattened tubes with ca. 8 Å d-spacing. In particular, peaks were observed at $2\theta = 12.0$, 19.9, 24.7, 35.0, 38.3, 54.9 and 62.4 corresponded to (001), (100), (002), (110), (003), (210) and (300) planes, respectively [53]. The (001) plane corresponds to a basal spacing of 0.72 nm, which identifies these samples as halloysite-(7 Å).

Traces of quartz were identified only in the PT_Hal sample, corresponding to a sharp peak at $2\theta=26^\circ$ and $60^\circ.$

A qualitative evaluation of the gel formation is shown in Fig. 3. We reported the images of PT_Hal and UP_Hal 1 wt% before and after the inversion. Despite the same concentration of PT_Hal, it is stable to the tube inversion test showing a gel-like structure compared to the same or even higher concentrations of UP_Hal in which it is not evidenced.

3.3. Rheological characterization of PT_Hal

Aimed to characterize the viscoelastic behavior of the clay, amplitude sweep tests were performed by measuring the strain amplitude dependence of the storage and loss moduli (G', G").

In Fig. 4 we reported two concentrations as examples of the more comprehensive concentration range investigated in this work, just to show the linear viscoelastic regions. Namely, below 10 % strain, G' is higher than G", for both PT_Hal at 1 and 2 wt%, suggesting a gel-like structure. As a matter of fact, in this region both moduli remain constant, indicating the presence of the linear viscoelastic region. The crossover strain points are clearly displayed after 10 % strain and their values were calculated determining the point at which G'=G". As shown in Fig. 4, the crossing points between G' and G" moduli are shifted to higher oscillation strain by increasing the clay concentration. Specifically, they are at 16.30 % and 45.52 % for PT_Hal 1 and 2 wt%, respectively.

Having pinpointed the linear viscoelastic regions, angular frequency sweep tests were performed, imposing a constant strain on the systems at variable frequency. Usually, frequency sweep curves give a rheological description of the product behavior during storage and application [54-56]. The obtained rheological data are displayed in Fig. 5a), showing G' and G" moduli as a function of the angular frequency. As a general result, the storage modulus (G') is larger than the loss modulus (G") in the entire range of frequency. Namely, for a gel-like material G' and G" curves are constant, remarking that PT Hal behaves like a solid and suggesting that the material is highly structured starting from a concentration of 0.75 wt%. High frequencies correspond to short time scales and low frequencies to longer time scales. This means that the gellike structure is time-independent [57,58], highlighting their stability over time. Moreover, Fig. 5b) shows that increasing the concentration both moduli increase and the G' medium values are higher than G" from 0.005 wt% along the angular frequency range investigated.

We calculated the average tan δ values within the angular frequency range between 0.1 and 100 rad s⁻¹ by G''/G' ratios. The obtained data (see Supporting Information) confirmed the solid-like behavior of the hydrogels being that tan δ is lower than 0.1 for all the investigated dispersions. We detected that tan δ is nearly constant throughout the entire concentration range highlighting the predominance of the elastic contribution.

The viscosity as a function of the shear rate can provide relevant information about its performance and processing. With this in mind, flow curves are reported in Fig. 6. The viscosity displays higher values increasing the concentration of the clay. Accordingly, the most considerable difference is evidenced for PT_Hal 2 wt%, which presents an order of magnitude of increment with respect to 1 wt%. It is clear that the microstructure plays a crucial role in determining the rheological behavior of the system. In this case, two phenomena should be taken



Fig.2. Morphological properties of pure Patch and UltraPure Halloysite. (a-d) Scanning electron microscopy micrographs of PT_Hal_P (e-f) Scanning transmission electron microscopy micrographs showing patchy entangled halloysite nanotubes of PT_Hal_P (g) Scanning transmission electron microscopy micrographs of UP_Hal.



Fig. 3. Inversion test of PT_Hal and UP_Hal 1 wt%.

into consideration to explain the rheology of Patch halloysite.

The first is that increasing the concentration, the interactions between the nanotubes are enhanced promoting a higher viscosity of the system. Another crucial factor is represented also by the presence of a bird nest structure in PT_Hal in which the nanotubes are entangled with each other and bent[40], reducing the free water diffusion between the nanotubes. In this regard, geometrical considerations can give a more comprehensive view about the influence of the tube's morphology on the rheology. We can consider a contact distance given by the average length of the nanotubes and assume a simple cubic model to calculate the critical overlapping volume fraction φ^* :

$$\rho^* = \pi R^2 / L^2 \tag{2}$$

where R and L are the external radius and the length of the nanotubes, respectively. Volume fraction values of ca. 4–10 % for aqueous dispersions of halloysite from Dragon Mine have been previously reported [3].

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Fig. 4. Amplitude sweep tests of PT_Hal. Full and empty symbols indicate Storage modulus (G') and Loss modulus (G''), respectively.



Fig. 5. A) results of angular frequency sweep tests on pt_hal at 2, 1 and 0.75 wt %. Full and empty symbols indicate Storage modulus (G') and Loss modulus (G'), respectively. b) Storage and Loss moduli as a function of the clay concentration.

Here, the calculated ϕ^* , by considering average values for R and L [2], are 3.1 % for UP_Hal and 0.03 % for PT_Hal. This means that the volume required to generate the nanotubes entanglement is 100 times lower for PT_Hal than UP_Hal. Namely, the critical concentration for the hydrogel formation is reached at lower concentration for PT_Hal samples with respect to shorter and stubby halloysite clay nanotubes.

Moreover, the presence of this network-like structures in PT_Hal, reduces the fluid flow, contributing to the higher viscosity of the system [59,60].

On the other hand, the viscosity below 0.75 wt% assumes low values



Fig. 6. Flow ramp tests showing the viscosity as a function of the shear rate.

pointing out that the network is not structured as in 1 wt%. As a result of this transition, the material undergoes a change from an apparently solid state to a liquid state, allowing it to flow easily. All the samples, starting from a shear stress of 0.1 s^{-1} exhibited shear thinning, as represented by the decrease of viscosity with increasing the shear rate. This phenomenon is requested in hydrogel for biomedical applications since they should exhibit viscous flow under shear stress and self-healing properties as reported elsewhere [61,62].

It should be noted that PT_Hal 0.75, 1 and 2 wt% at very low shear rate exhibited an increase of the viscosity up to a critical value, beyond which it decreases. Thus, this indicates the presence of yield stress that is better evidenced in Fig. 7, where the shear stress as a function of the shear rate is reported.

Accordingly, the yield stress reflects the resistance of the sample structure to deformation or break and it is detected starting from PT_Hal 0.75 wt%. The flow begins when an applied stress is strong enough to disturb the gel-like structure of the material, resulting in a significant decrease in viscosity, as previously shown in Fig. 6. This is better highlighted by looking at the inset in Fig. 7, where the yield stress is not evidenced up to a concentration of 0.5 wt% and these curves start from 0 Pa, meaning that the material flows and it is easily deformable. On the contrary, the yield stress of PT_Hal 2 wt% is remarkable as it is significantly higher than less concentrated samples, reaching a value of ca. 28 Pa (Table 2).

Thus, we can state that PT_Hal is a non-Newtonian fluid and specifically, it can be classified as a Pseudoplastic fluid at low concentration and as Bingham Plastic fluid at concentrations greater than 0.5 wt%.

To better investigate the rheological properties of Patch halloysite, the Carreau (Eq. (3)) and Carreau-Yasuda models (Eq. (4)) were applied to the experimental flow curves obtaining the fitting parameters reported in Table S3.

$$\frac{\eta - \eta_{\infty}}{\eta_0 - \eta_{\infty}} = \frac{1}{[1 + (\mathbf{k}\dot{\mathbf{y}})^2]^{\frac{n}{2}}}$$
(3)

$$\frac{\eta - \eta_{\infty}}{\eta_0 - \eta_{\infty}} = \left[1 + (\mathbf{k}\dot{\mathbf{y}})^a\right]^{\frac{p-1}{a}} \tag{4}$$

Here, η_0 is the viscosity at zero shear rate, η_∞ is the viscosity at infinite shear rate, k represents the consistency index or relaxation time and n and a are two dimensionless parameters. The Carreau-Yasuda model is an extension of the Carreau model. In particular, the main difference between the two models is linked to the ability to capture shear thinning or thickening behaviours. The Carreau model is usually referred to as shear thinning, meaning that the fluid viscosity decreases as the shear rate increases. In contrast, the Carreau Yasuda model is used for both shear thinning and thickening behaviours [63]. The presence of yield



Fig. 7. Shear stress as a function of the shear rate. The inset shows a magnification highlighting different types of flow behaviour.

Sample	Yield Stress σ_0 /Pa	
PT_Hal 0.005 wt%	0	
PT_Hal 0.1 wt%	0	
PT_Hal 0.5 wt%	0	
PT_Hal 0.75 wt%	3.07	
PT_Hal 1 wt%	4.69	
PT_Hal 2 wt%	28.08	

stress led to the Carreau Yasuda model to better fit the analysed curves as reported in Supporting Information (Table S3).

An increase in zero-shear rate viscosity and consistency with increasing PT_Hal concentration is detected and showed in Fig. 8.

3.4. Effect of ionic strength

The ionic strength can have a strong impact on gel-like structures, and it was widely reported in literature [64,65]. For example, it has been demonstrated that the addition of NaCl can weaken the gel structure of cellulose nanocrystal suspensions and decrease the viscosity as well [66]. With this in mind, the effect of ionic strength on the investigated gels was evaluated. Accordingly, flow ramp tests performed on PT_Hal 1 wt% in water and at different NaCl concentrations are reported in Fig. 9. It can be noted that the gel-like network is not destroyed by adding the







Fig. 9. Flow ramp tests. Viscosity as a function of the shear rate for PT_Hal 1 wt % in water and NaCl at different concentrations.

salt and with respect to water, lower concentrations of NaCl do not affect the rheology properties. As a matter of fact, the viscosity curves of PT_Hal in water and in NaCl 10^{-4} M are overlapped. There is a small effect of ionic strength increasing the NaCl concentration evidenced by a minimal increase of viscosity.

 ζ potential measurements were carried out in order to have information about the surface charge of the nanotubes. As reported by Cavallaro et al.[44], although Hal from different natural sources can show similar responses to salt addition, they present differences in the effective charge that can influence their colloidal stability. The surface charge (σ) was estimated from the ζ potential values as established by the Grahame equation:

$$\sigma = (8RT\varepsilon\varepsilon_0 C)^{\frac{1}{2}} \sinh\left(\frac{Ze\zeta}{2k_bT}\right)$$
(5)

where ε is the solvent relative dielectric constant, ε_0 is the vacuum dielectric constant, C is the bulk concentration of electrolyte, e is the electron elemental charge, and Z is 1 for NaCl.

As shown in Table 3, the addition of NaCl generated a slight decrease of the absolute value of ζ potential for PT_Hal. These results can be explained by considering the screening effect of Na⁺ ions towards the negative surface charge of PT_Hal. Namely, the presence of the

Table 3

ζ Potential PT Hal samples in water and NaCl.

Sample	ζ Potential/mV	Surface charge $\sigma/mC~m^{-2}$
PT_Hal PT_Hal/NaCl 10 ⁻⁴ M PT_Hal/NaCl 10 ⁻² M	$-17.5 \\ -15.5 \\ -14.0$	$^{/}$ 1.13·10 ⁻⁵ 1.02·10 ⁻⁴

electrolyte in the aqueous medium caused a reduction of the Debye length inducing a decrease of ζ potential absolute value. As expected, the screening effect was enhanced by increasing the ionic strength of the aqueous dispersion. Similar results were observed for halloysite nanotubes from other sources (Matauri Bay and Dragon Mine) [44].

4. Conclusions

This work aims to characterize PT Hal clay samples from a rheological point of view, highlighting that clay microstructure plays a key role on it. As a matter of facts, their different morphology led to peculiar physico-chemical properties that are not evidenced in UP Hal. PT Hal shows thinner and longer tubes creating a structure similar to a bird nest allowing the formation of gel-like structures. It was proved by dynamic viscoelastic tests that PT_Hal is a non-Newtonian fluid and G' is higher than G", meaning that the system is highly structured. Flow curves provided information about the viscosity related to shear rates, which is crucial in the production field where the product is subjected to stirring and pumping processes and namely different shear rates. Accordingly, two different behaviours were evidenced depending on the clay concentration: from 0.75 wt% a yield stress is detected, reflecting the sample's ability to withstand deformation, or breaking. The effect of ionic strength was also evaluated, showing that the addition of salt doesn't affect the properties of the gel since there isn't a significant change in both ζ potential and viscosity of PT Hal samples.

CRediT authorship contribution statement

Martina Maria Calvino: Investigation, Data curation, Writing – original draft. Giuseppe Cavallaro: . Pooria Pasbakhsh: Data curation, Writing – original draft. Giuseppe Lazzara: Conceptualization, Supervision. Stefana Milioto: Funding acquisition, Resources.

Declaration of Competing Interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

Data availability

Data will be made available on request.

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Appendix A. Supplementary material

Supplementary data to this article can be found online at https://doi.org/10.1016/j.molliq.2023.123721.

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