## **RESEARCH**



# **Multifunctional 3D‑printed composites based on biopolymeric matrices and tomato plant (***Solanum lycopersicum***) waste for contextual fertilizer release and Cu(II) ions removal**

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Received: 23 November 2023 / Revised: 7 May 2024 / Accepted: 8 May 2024 / Published online: 21 May 2024 © The Author(s) 2024

#### **Abstract**

The production of tomatoes faces significant challenges, including the high amount of waste generated during the harvest stage and copper-contaminated soil due to pesticide use. To address these issues and to promote a more sustainable agriculture, innovative biodegradable green composites for contextual controlled soil fertilization and Cu removal were produced by 3D-printing technology. These composites were made by incorporating NPK fertilizer flour and tomato plant waste particles (SLP) into three different biodegradable polymeric matrices: polylactic acid (PLA); a commercial blend of biodegradable co-polyesters (Mater-Bi®, MB) and their blend (MB/PLA, 50:50). Rheological characterization suggested the potential processability of all of the composites by FDM. Morphological analysis of printed samples confirmed the good dispersion of both filler and fertilizer, which also acted as reinforcement for MB and MB/PLA composites. SLP and NPK moduli were evaluated by powder nanoindentation and, for almost composites, the theoretical Halpin-Tsai model satisfactorily fitted the actual tensile moduli. The decrease in NPK fertilizer release rate and the increase in Cu(II) removal efficiency were achieved using whole 3D-printed composites. By selecting the appropriate matrix and incorporating SLP particles, it was possible to tune the NPK release rate and achieve copper absorption efficiency. Notably, MB samples containing SLP particles displayed the fastest release and the highest Cu(II) removal efficiency.

**Keywords** Hybrid composites · Bio-composites · Natural filler · Fertilize release · Cu removal · 3D printing

# **1 Introduction**

Water, soil, and food contamination is currently one of the major global problems [[1](#page-13-0)]. The use of copper-containing fertilizers, pesticides and fungicides has rapidly increased in the Mediterranean area over the years [[2,](#page-13-1) [3](#page-13-2)]. Tomato cultivation is widespread worldwide due to its nutritional and economic value [\[4](#page-13-3)]. However, the excessive use of pesticides, including copper-based compounds, in tomato cultivations poses a significant risk to both the environment and human health [\[5\]](#page-13-4). Although pesticides may effectively control pests and diseases, they can cause long-term environmental

 $\boxtimes$  Roberto Scaffaro roberto.scafaro@unipa.it damage by altering soil microbial communities and disrupting the balance of soil ecosystems [[6\]](#page-13-5). Moreover, the mutagenic and carcinogenic effects of this element raise profound concerns about the environment and human health, making them one of the serious worldwide ecological threat [[7](#page-13-6), [8](#page-13-7)]. In addition to the risks posed by copper-based pesticides, tomato harvest also generates significant amounts of waste due to large unused portions of the crops. These scraps, if not properly managed, can lead to negative environmental impacts, including soil contamination and greenhouse gas emissions primarily resulting from their improper burning directly on the soil [[4,](#page-13-3) [9](#page-13-8)]. Therefore, there is an urgent need to address the environmental risks associated with excessive copper use in tomato plant cultivations and the consequent management of their waste.

Among the biopolymeric matrices, polylactic acid (PLA) is one of the most commonly used [[10,](#page-13-9) [11](#page-13-10)], particularly to obtain green composites [[12](#page-13-11)]. Additionally, Mater-Bi® (MB)—a family of biopolymers widely used for applications that require biodegradability—has recently emerged

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as a promising option for green composite production [[13,](#page-13-12) [14\]](#page-13-13). Several organic biomasses, such as *Opuntia ficus indica* flour [[15–](#page-13-14)[17](#page-13-15)], *Posidonia oceanica* leaves [[18](#page-13-16), [19](#page-13-17)], and banana fibers [\[20,](#page-13-18) [21](#page-14-0)], have been successfully used as filler in biopolymeric matrices obtaining improvements in tensile properties and biodegradability. Recently, also *Solanum lycopersicum* (i.e., tomato plant, SLP) was added to MB aiming to produce 3D-printed samples [[13\]](#page-13-12). The addition of up to 10% of SLP allowed optimal processability, with satisfactory dispersion of the filler in the matrix and improved tensile strength upon increasing the filler content.

Adding natural scraps to biopolymeric matrices reduces plastic usage, thus decreasing costs and promoting more ecofriendly processes. Moreover, it accelerates matrix biodegradability while enhancing mechanical performance [\[22](#page-14-1), [23](#page-14-2)].

3D-printing and, more in detail, fused deposition modeling (FDM) attracted interest as an advantageous technique to produce green composites [\[24\]](#page-14-3), due to its ability to create highly complex geometries while significantly reducing production time and costs [[25](#page-14-4)–[27\]](#page-14-5). FDM structures have been already successfully produced for different applications: aerospace and automotive [[28\]](#page-14-6), drug release devices [\[29\]](#page-14-7), fertilizer release devices [\[30](#page-14-8)], and metal ions capture [[31\]](#page-14-9). FDM's potential to create complex and customized structures with a specific composition has been the key to its success: by appropriately selecting process parameters such as infill and raster angle, the porosity of the devices can be tailored, thereby enabling control over release and removal kinetics [[30\]](#page-14-8). Porosity, in fact, strongly influences particle delivery and metal ions capture [\[32](#page-14-10)].

This study aims to investigate the effectiveness of innovative biodegradable bio-composites for contextual controlled soil fertilization and Cu removal. The composites were fabricated using 3D-printing technology, adding NPK fertilizer flour and ground particles of tomato plant waste (SLP) to three different biodegradable polymeric matrices: polylactic acid (PLA); a commercial blend of biodegradable co-polyesters (Mater-Bi®, MB); their blend (MB/PLA, 50:50 ratio). By using more cost-effective biodegradable polymers and combining them with agricultural waste, it is possible to decrease the amount of polymer required. This strategy has the potential to reduce effectively and significantly the amount of plastic used and decrease the cost of the final device. Furthermore, the addition of natural fillers or agricultural scraps to biodegradable polymers can enhance mechanical property [[33](#page-14-11), [34\]](#page-14-12) of the device and accelerate biodegradability [\[35](#page-14-13)]. Ultimately, the aim of this work is to develop a device capable of controlling the release rate of NPK, selected as a model fertilizer compound, and contextually removing copper ions resulting from excessive fertilization, thus promoting sustainable agriculture. To validate the accomplishment of this goal, evaluations were conducted in order to test the Cu removal and NPK release ability of the devices. The Peppas-Korsmeyer mathematical model was used in order to analyze release test data.

# **2 Experimental section**

## **2.1 Materials and methods**

#### **2.1.1 Materials**

In this study, Mater-Bi $\otimes$  EF51L (MB; density = 1.22 g/cm<sup>3</sup>; melt flow index =  $3.5 \text{ g} / 10 \text{ min}$ , produced by Novamont SpA (Novara, Italy) and PLA 2003D (density =  $1.25$  g/cm<sup>3</sup>; melt flow index =  $6 \text{ g} / 10 \text{ min}$ ; 4.3% content of D-lactic acid monomer), purchased from NatureWorks® (Minnetonka, MN, USA) were used, both neat and in combination, to prepare green composites. MB and PLA were melt-mixed together using an internal mixer (Brabender, Germany;  $T=160$  °C, rotor speed=64 rpm,  $t=3$  min) aiming to produce MB/PLA (50:50) blend. Both neat matrices and the prepared blend were vacuum-dried overnight at 60 °C before processing. *Solanum lycopersicum* plant waste (SLP) was generously provided by a local farm (Palermo, Italy). The plants were mowed after tomato harvesting. In this study, only the stems and leaves of the tomato plant were used; the roots were removed prior to the following process. The tomato plants were firstly cleaned by immersing them in water at room temperature. After cleaning, the plants were dried under sunlight for 7 days and then further oven-dried overnight at 40 °C to ensure complete drying. The whole plants were then ground. NPK fertilizer with 12% of nitrogen content, 12% of phosphorus content and 17% of potassium content, purchased from Flortis, Orvital S.p.A., was used. SLP and NPK were separately ground for 3 min and sieved (Controls, USA), selecting a mesh size under 150  $\mu$ m aiming to achieve a smooth printing, taking into account the 0.4 mm diameter of the nozzle (see also Table [2\)](#page-2-0). According to our previous studies [[13,](#page-13-12) [30,](#page-14-8) [36](#page-14-14), [37](#page-14-15)], this particle size was chosen to avoid nozzle clogging during the 3D printing process. Prior to processing, both SLP and NPK collected powders oven dried at 40 °C overnight to reduce polymeric matrices hydrolytic scission phenomena.

# **2.1.2 Preparation of MB, PLA and MB/PLA neat and composites filaments**

Aiming to obtain a good dispersion of the fillers and achieve good printability, specific amounts of SLP and NPK were chosen to prepare hybrid biocomposites, not overcoming a total amount of 20 wt% of filler added. More in detail, nine different formulation were prepared. In detail, a Brabender internal mixer was used in order to melt mix 10% of SPL and (or) 10% of NPK to all the selected matrices ( $T = 160$  °C for MB and MB/PLA series and *T* = 190 °C for PLA series, rotor speed=64 rpm,  $t=3$  min). The three neat matrices (MB, PLA, and MB/PLA) were also processed under identical conditions for the purpose of comparison. The obtained formulations were then extruded using a Haake Polylab, Germany singlescrew extruder equipped with a cylindrical nozzle (thermal profile of 130–140–150–160 °C for MB and MB/PLA series and 160–170–180–190 °C for PLA). A conveyor belt system (6 m/min) was used to draw the output extrudate aiming to obtain filaments with a diameter of  $\sim$  1.75 mm. The filaments were named based on the selected matrix and their filler contents. Formulations of the matrices and the produced composites, together with their code names, are reported in Table [1.](#page-2-1)

#### **2.1.3 Fabrication of 3D‑printed samples**

The design of printed samples was performed following the procedures already reported in our previous work [\[30](#page-14-8)] and briefly described in Section-1 of SI. Sharebot, Next Generation (Italy) 3D printer was used and the adopted FDM parameters are summarized in Table [2.](#page-2-0)

## **2.2 Characterizations**

## **2.2.1 Morphological analysis**

Morphological analysis of SLP and NPK powder cross sections surfaces (cryo-fractured) of filaments and 3D printed samples was performed through Phenom ProX (Phenom-World, The Netherlands) scanning electron microscope (SEM). For more information about the operative condition, please see Section-1 of supplementary material.

#### **2.2.2 Rheological behaviors**

The study of composite rheological behaviors can provide preliminary information about filament printability. Aiming to observe the influence of SLP particles and/or fertilize on the

<span id="page-2-1"></span>**Table 1** FDM process parameters

FDM parameter	Value for MB, PLA, and MB/PLA series	
Nozzle temperature	160, 190, 160 °C	
Nozzle diameter	$0.4 \text{ mm}$	
Bed temperature	60, 90, 60 °C	
Infill rate	100%	
Infill pattern	Rectilinear	
Layer thickness	$0.1 \text{ mm}$	
Extrusion width	$0.4 \text{ mm}$	
Raster angle	$+45^{\circ}$	
Printing speed	$50 \text{ mm/s}$	
Perimeter shells	4	

<span id="page-2-0"></span>**Table 2** Formulations of used polymeric matrices and prepared biocomposites

Sample code name		MB [wt%] PLA [wt%] SLP [wt%]		$NPK$ [wt%]
MВ	100	$\Omega$	0	$\theta$
PLA	$\Omega$	100	0	$\theta$
<b>MB/PLA</b>	50	50	0	$\theta$
<b>MB-SLP</b>	90	$\Omega$	10	$\theta$
<b>MB-NPK</b>	90	$\Omega$	0	10
<b>MB-SLP-NPK</b>	80	$\Omega$	10	10
<b>PLA-SLP</b>	$\Omega$	90	10	$\theta$
PLA-NPK	$\theta$	90	0	10
<b>PLA-SLP-NPK</b>	$\Omega$	80	10	10
MB/PLA-SLP	45	45	10	$\theta$
<b>MB/PLA-NPK</b>	45	45	0	10
MB/PLA-SLP- NPK	40	40	10	10

viscosity of the polymeric matrices and their potential impact on processability [\[38\]](#page-14-16), rheological behaviors of the obtained filaments were analyzed, using a rotational rheometer (ARES-G2, TA Instruments, New Castle, PA, USA). The tests were conducted in frequency sweep mode in the range 0.1–100 rad/s, a constant stress of 1 Pa and using a 25-mm parallel-plate geometry. A temperature of 160 °C was used for MB an MB/ PLA series while 190 °C was adopted for PLA series.

## **2.2.3 Thermal properties**

Differential scanning calorimetry (DSC) analysis was performed on the neat polymeric matrices using a Chip-DSC 10 (Linseis Messgeraete GmbH, Selb, Germany). The samples were heated to 200 °C at a heating rate of 20 °C/min. Nitrogen was used as a shielding gas to avoid polymer degradation. Trace analysis was carried out using Linseis and GraphPad Prism 9 software.

#### **2.2.4 Nanoindentation of powder**

The elastic modulus of SLP and NPK particles were tested using an Anton Paar NHT<sup>2</sup> nanoindenter (CSM Instruments, Switzerland) equipped with a Berkovich diamond indenter, following Baraldi et al.'s method [[39](#page-14-17)]. In brief, SLP and NPK particles suitable for the test were selected using an optical microscopy (Leica MS5 stereo microscope, Wetzlar, Germany) and glued on a dedicated square glass slide. The indentation experiments were performed using the following protocol: a loading ramp of 30 s, a 30 s dwell period at maximum load (15.00 mN), and a final unloading step lasting 30 s. Measures were repeated at least six times, the average values and the standard deviations were calculated.

#### **2.2.5 Density measurements**

Density measurements of powders and polymeric matrices were carried out by using a Thermo Pycnomatic Helium Pycnometer (Pycnomatic ATC, Thermofisher, USA). The measurements were performed at 25 °C, using pure Helium. Measures were repeated at least six times, the average values and the standard deviations were calculated.

#### **2.2.6 Mechanical characterization and modeling**

Tensile properties of the 3D printed samples were investigated using an Instron 3365 machine (Norwood, MA, USA), following the same method already reported in our previous work [\[30](#page-14-8)]. For more information about the operative condition please see Section-1 of supplementary material.

Data obtained underwent statistical analysis with an unpaired Student t-test using GraphPad Prism 9. Statistical significance was attributed to differences between datasets when the *p*-value obtained was lower than 0.05.

Tensile test results were compared with the outcomes predicted by the Halpin-Tsai model. In a ternary composite, the Halpin-Tsai equation can be rewritten according to Eq. ([1\)](#page-3-0) [[40\]](#page-14-18):

$$
E_{C,HT} = \frac{3}{8}E_{L} + \frac{5}{8}E_{T}
$$
 (1)

For more information about model implementation please see Section-1 of supplementary material.

## **2.2.7 Release of NPK fertilizer and modeling**

The material's ability to release NPK fertilizer was investigated performing conductively measurement following the protocol already reported in our previous work [[30](#page-14-8)]. The Peppas-Korsmeyer model was used in order to fitted the release data. For more information about release test and the modeling of the obtained data, please see Section-1 of supplementary material.

## **2.3 Evaluation of Cu(II) adsorption**

 $CuSO<sub>4</sub>·5H<sub>2</sub>O$  (Merck, 99.0%) was used to prepare the Cu(II) ion solutions. The metal ion adsorption capability was assessed by immersing the 3D-printed devices in 50 mL of copper sulfate aqueous solutions (1000 mg/L) at a room temperature for 30 days. The metal concentration in the collected aliquots was assessed by using UV/vis spectrophotometer (model UVPC 2401, Shimadzu Italia s.r.l., Milan, Italy). Previously, a series of aqueous solutions containing a known amount of  $CuSO<sub>4</sub>$  were analyzed in order to obtain a calibration line. The maximum absorbance band of  $CuSO<sub>4</sub>$  water solution was detected at 810 nm. The removal efficiency (%) of the samples was calculated according to Eq. ([2\)](#page-3-1) [[41](#page-14-19)]:

<span id="page-3-1"></span>Removal efficiency = 
$$
RE(\%) = \frac{C_i - C_f}{C_i} \times 100
$$
 (2)

where  $C_i$  (mg/L) and  $C_f$  (mg/L) are the concentration of copper sulfate before and after adsorption tests. All the experiments were carried out in triplicate.

The surface composition of 3D-printed samples with adsorbed Cu(II) ions was collected and analyzed by energy dispersive X-ray (EDX) analysis by using an EDX probe during SEM imaging to detect the presence of Cu(II).

# **3 Result and discussion**

#### **3.1 Morphology of SPL and NPK powders**

<span id="page-3-0"></span>After the drying process, small pieces of tomato plant stems were examined using SEM to evaluate their internal morphological structure. SEM images (Fig. [1a](#page-4-0) and b) revealed that the tomato plants used in this work are characterized by circular and square-shaped voids regularly distributed along the entire length of the stem. These features could be particularly useful to achieve infiltration of the polymeric matrices into the filler, thereby potentially obtaining a mechanical reinforcing effect with the formation of an interphase [[42,](#page-14-20) [43](#page-14-21)]. After this preliminary characterization, the plant was then ground into fine particles and the obtained powder was sieved, selecting the fraction under 150 µm. Relevant SEM micrographs of SLP powder are shown in Fig. [1](#page-4-0)c and d. SLP flour is characterized by an hybrid morphology: some particles showed a fibrous shape (yellow arrows), while others had a flakes-like geometry (red arrows) due to the fact that whole tomato plant was ground as received without separating the different parts: the fibrous-shaped particles likely belong to the stems, while the flakes-like ones belong to the leaves. Moreover, fibrous-shaped particles are characterized by lengths greater than 150 μm but with high aspect ratio. This feature allows them to pass through the sieve when aligned perpendicularly to the mesh. SEM images in Fig. [1d](#page-4-0) and e (closeup view and arrow) revealed that the porous structure of the stem was partially preserved even after grinding, and that some skein-shaped particles (green circle) were also sporadically present.

Relevant micrographs of ground NPK fertilizer particles are shown in Fig. [1f](#page-4-0) and g. Although they displayed some inhomogeneity in shape and size, they all exhibited a rounded geometric shape. Furthermore, NPK and SLP particles have distinct morphologies.

<span id="page-4-0"></span>**Fig. 1** SEM micrographs of tomato plant steam (**a**) and close-up view (**b**); SLP powder (**c**, **d**) and close-up view (**e**) and NPK powder (**f**, **g**). Highlighted areas in **b** and **e** provide a detailed examination of the material's microstructure



# **3.2 Processing and morphological analysis of filaments**

First, all the different formulations (included neat MB and PLA) were melt-mixed for 2 min  $(T = 160$  °C for MB and MB/PLA series and *T* = 190 °C for PLA series). The obtained mixtures were ground into pellets and then extruded into filaments using a cylindrical nozzle and a conveyor belt system. Melt mixing was performed aiming to improve the dispersion of the fillers in the polymeric matrices. This preliminary mixing enables the production of composite filaments with a homogeneous dispersion of the filler and a rounded shape, improving properties and printing performance of them. Both neat matrices and SLP and/or NPK-containing filament (diameter: ~ 1.75 mm) were successfully extruded, their morphology was analyzed, and relative optical images and SEM micrograph of their cryofractured cross section are reported in Fig. [2.](#page-4-1)

SEM micrographs confirm that all the filament showed a diameter of  $\sim$  1.75 mm and an almost regular cylindrical shape (Fig. [2](#page-4-1)a–c; e–g; i–k) but -SLP-NPK ones that exhibited



<span id="page-4-1"></span>**Fig. 2** SEM micrograph of MB (**a**); MB-SLP (**b**); MB-NPK (**c**); MB-SLP-NPK (**d**); PLA (**e**); PLA-SLP (**f**); PLA-NPK (**g**); PLA-SLP-NPK (**h**); MB/PLA (**i**); MB/PLA-SLP (**j**); MB/PLA-NPK (**k**); MB/PLA-SLP-NPK (**l**) flaments cross section

a non-totally uniform and quite rough surface (Fig. [2d](#page-4-1), h, l). More in detail, MB (Fig. [2a](#page-4-1)), PLA (Fig. [2](#page-4-1)e) and MB/PLA (Fig. [2](#page-4-1)) neat filaments showed a very regular cylindrical shape. Furthermore, when blend together, MB and PLA, form an apparently homogeneous structure, at least at this scale, as shown in Fig. [2i](#page-4-1). When just SLP is added to the different polymeric matrices (Fig. [2](#page-4-1)b, f, and j), the regular cylindrical shape and homogeneous dispersion of filler are almost preserved. SLP exhibited a good adhesion with the matrices and limited presence of voids can be noted in all three filament samples. On the other hand, the addition of NPK (Fig. [2](#page-4-1)c, g, and k) had no adverse effect on the geometric regularity of the filament. Moreover, in all filament samples, an almost homogeneous dispersion of the fertilizer with very limited presence of voids could be noted and the interfacial adhesion between the particles and the polymeric matrices seems to be good. MB-SLP-NPK (Fig. [2d](#page-4-1)), PLA-SLP-NPK (Fig. [2](#page-4-1)h) and MB/PLA-SLP-NPK (Fig. [2l](#page-4-1)) exhibit uneven edges with some voids along them. This can be explained taking into account that in this latter case the total amount of fillers (i.e., SLP and NPK) is 20%, and no more 10%. However, the cylindrical shape is sufficiently preserved in order to be 3D printed, and both the filler and the fertilizer are homogeneously dispersed throughout the filaments. The neat matrices and the SLP and(or) NPK-containing formulations showed morphology potentially suitable for FDM printing process.

# **3.3 Rheological behaviors and thermal properties of filaments**

Filaments printability by FDM technology is strongly related to their rheological behavior [[44](#page-14-22)]. Therefore, rheological analyses were investigated by using rotational rheometer and results are reported in Fig. [3a](#page-5-0)–c.



<span id="page-5-0"></span>**Fig. 3** Complex viscosities of MB (**a**), PLA (**b**), and MB/PLA (**c**) series performed at 160 °C, 190 °C, and 160 °C, respectively, and DSC curves (d) of the polymeric matrices

Based on our previous works [[15](#page-13-14), [38\]](#page-14-16), all the formulations based on MB, PLA, and MB/PLA displayed viscosity values that are potentially suitable for be 3D-printed. More in detail, MB (Fig. [3a](#page-5-0)) shows an almost Newtonian behavior at lower frequencies and a shear-thinning behavior at higher ones. The presence of SLP or NPK lead to an increase in viscosity values across the whole range of frequency, and a more pronounced non-Newtonian behavior. When compared to MB-NPK, MB-SLP displayed higher viscosity value, possibly due to the porous and spongy structure of SLP particles that allows the polymeric matrix to infiltrate it. This tendency further increases when both SLP and NPK were present in the composite and the onset of yield stress phenomena could be observed. In this latter case, it is necessary to consider that the total amount of the dispersed phase was 20% (10% SLP nd 10% NPK)and no more 10% (SLP or NPK only) like in the previous analyzed systems. On the other hand, neat PLA (Fig. [3](#page-5-0)b) displayed its typical [[45,](#page-14-23) [46\]](#page-14-24) Newtonian behavior at lower frequencies and a slight shear-thinning in the higher ones region. The addition of SLP or NPK into the PLA matrix significantly lowered the complex viscosity over the entire investigated frequencies range (0.1–100 rad/s), as already reported for similar composite systems [[44\]](#page-14-22), suggesting a decrease of the molecular weight  $(M_w)$  due to a deterioration of the polymer phase upon processing. In the PLA composite filament (Fig. [3b](#page-5-0)), a slight change in the shape of the curve was observed: PLA-SLP, PLA-NPK, and PLA-SLP-NPK exhibited a non-Newtonian behavior at lower angular frequencies and pronounced shear-thinning behavior at higher frequencies. Moreover, for PLA-SLP-NPK the presence of yield stress phenomena could be observed. The presence of yield stress, in fact, is typical of composite systems. In Fig. [3c](#page-5-0), the complex viscosities of the MB/PLA series are reported. The rheological behavior of MB/PLA blend was mostly influenced by the PLA phase. In the case of MB/PLA series, in fact, it should be considered that the temperature of 160 °C (used for filament preparation and sample printing), was chosen in order to limit MB thermal degradation. Therefore, at this temperature, the PLA phase was not completely molten. Due to the presence of this partially molten phase, neat MB/PLA exhibited a non-Newtonian behavior characterized by a shear-thinning behavior at high angular frequencies and the onset of yield stress phenomena at lower frequencies. The presence of SLP or NPK leads to a decrease in viscosity values across the entire frequency range, but almost no change in the shape of the curve was observed. When both SLP and NPK were present in the composite (MB/PLA-SLP-NPK), a more pronounced non-Newtonian behavior and yield stress phenomenon could be noted.

The results suggest that the rheological behaviors of all the composite filaments were compatible with the 3D printing process, especially those of the MB/PLA series. Shear-thinning properties are fundamental for lowering viscosity values, aiming to improve filament printability for 3D printing.

In order to deeply investigate the behavior of the blend, DSC measurements were performed on the neat matrices and the obtained results are presented in the form of thermogram plots in Fig. [3d](#page-5-0). MB and PLA exhibited the same glass transition temperature of about 75 °C and melting point of 130 °C and 165 °C, respectively. MB/PLA showed almost the same glass transition temperature of the single polymeric component (about 75 °C) and two different melting points: the first at about 125 °C followed by a second at about 160 °C likely related to the melting of the PLA phase. Considering that MB is a blend of aromatic and aliphatic biodegradable co-polyesters reasonably including PLA [\[14](#page-13-13)], the total melting enthalpy of the different matrices could be compared in order to obtain information about the crystallinity of the materials. The thermal properties of PLA confirm a typical almost amorphous behavior of PLA, while MB exhibited a much higher melting enthalpy value showing similar results to other research studies [[14\]](#page-13-13). When MB and PLA are mixed together, a higher melting enthalpy is observed reasonably due to the occurrence of a nucleation effect as already reported for similar systems [[47\]](#page-14-25).

## **3.4 3D printing and morphological analysis of composites**

MB, PLA, and MB/PLA composites filament with 10% of SLP and/or NPK powders were successfully extruded obtaining a diameter of about 1.75 mm. Pure MB, PLA, and MB/PLA filaments were also produced using identical procedure for comparison. All filaments were easily 3D printed for FDM as was predicted considering SEM analysis (Fig. [2\)](#page-4-1) and their rheological behaviors (Fig. [3a](#page-5-0)–c). The FDM process, in fact, proceeded smoothly, without any interruption, (also for -SLP-NPK ones despite the non-uniform surface of the filaments) and no nozzle clogging occurred. All the composites formulations could be actually printed without varying the temperature normally used for MB and PLA pure matrices (160 °C and 190 °C, respectively). Neat MB/ PLA and its composites have been fluidly printed at 160 °C. Specimens of  $60 \times 10 \times 1$  mm have been successfully printed with all the prepared formulations (Fig. S1).

Morphological characterization was conducted on the cryofractured cross section of 3D printed composite samples, see SEM micrographs in Fig. [4.](#page-7-0)

A relatively uniform dispersion of both natural filler and fertilizer was observed in all the composites (Fig. [4](#page-7-0)a–i). In MB (Fig. [4](#page-7-0)a–c) and PLA-based composites (Fig. [4d](#page-7-0)–f),

imperfect adhesion was observed between the matrices and both SLP and NPK particles, as noted by the presence of many voids in the fracture surface. This behavior was more pronounced with SLP particles. On the other hand, both SLP and NPK particles appeared to be well-embedded in the MB/ PLA matrix (Fig. [4g](#page-7-0)-i) and the adhesion between matrices and particles seemed to be good, especially for SLP particles (see red circle and arrow in Fig. [4g](#page-7-0) and i, respectively). In this latter case, the matrix penetrated inside the SLP particle pores, anchoring the particles to the matrix and achieving excellent adhesion. Moreover, all FDM-printed samples exhibited a porous structure created by the separation points among adjacent raster layers due to the selected raster angle  $(\pm 45^{\circ})$ . The production of porous devices is really interesting because it allows water transport across the samples, enhancing fertilizer release and pollutant removal [[48](#page-14-26), [49](#page-14-27)].

#### **3.5 Nanoindentation of SLP and NPK powders**

Elastic modulus (E) of SLP and NPK particles have been investigated using nanoindentation. The nanoindentation test showed that the elastic modulus of NPK fertilizer is significantly higher than that of SLP. Specifically, Young's modulus of NPK was found to be  $12,708(\pm 8.7)$  MPa, while SLP one was  $171(\pm 0.1)$  MPa. This difference in stiffness between NPK and SLP can be attributed to the material's nature: NPK, in fact, is composed of a mixture of nitrogen, phosphorus, and potassium salts, which are known for their hardness and strength. On the other hand, SLP is made up of organic compounds, which tend to be relatively soft.

## **3.6 Mechanical properties of 3D printed composites and Halpin‑Tsai model**

The mechanical performance of 3D-printed composite samples was investigated through tensile tests and the outcomes of composite elastic modulus, normalized to those of each matrix, are shown in Fig. [5](#page-8-0). The values of elastic modulus (E), tensile strength (TS), and elongation at break (EB) are reported in Table S2. The addition of SLP particles to MB (Fig. [5](#page-8-0)a, first group of bars) did not lead to any statistically relevant variation in the *E* value (see Fig. S2). To address this behavior, the fractured cross-section of MB-SLP was analyzed by SEM. As shown in Fig. [5b](#page-8-0), during the fracture phase, SLP particles were pulled out of the polymeric matrix limiting their reinforcing effect. On the other hand, the addition of NPK to MB (MB-NPK and MB-SLP-NPK) led to a slight but statistically significant (see Fig. S2) increase in E, going from 89 MPa of neat MB to 94 and 95 MPa of MB-NPK and MB-SLP-NPK respectively.

Regarding the PLA series (Fig. [5](#page-8-0)a, second group of bars), the addition of SLP or NPK particles, or both contextually, to PLA led to a slight but statistically significant (see Fig. S2) <span id="page-7-0"></span>**Fig. 4** SEM micrograph of MB-SLP (**a**); MB-NPK (**b**); MB-SLP-NPK (**c**); PLA-SLP (**d**); PLA-NPK (**e**); PLA-SLP-NPK (**f**); MB/PLA-SLP (**g**); MB/PLA-NPK (**h**); and MB/ PLA-SLP-NPK (**i**) cross section of 3D printed samples



decrease in *E*, going from 891 MPa of neat PLA to 852, 820 and 850 MPa of PLA-SLP, PLA-NPK, and PLA-SLP-NPK, respectively. These results appear to be in full agreement with those obtained during rheological characterization. MB/PLA blend showed a different mechanical behavior (Fig. [5a](#page-8-0), third group of bars). The addition of SLP particles led to an increase in *E*, going from 503 MPa of neat MB/ PLA to 635 MPa of MB-SLP. In fact, the filler is currently well-integrated into the polymeric matrix, acting as a reinforcement (see SEM images in Fig. [4\)](#page-7-0). This is evident from the observed fracture behavior during tensile testing (Fig. [5c](#page-8-0)), where SLP particles tend to break rather than pull out from the matrix. These results suggest that the fillermatrix interface is strong, allowing efficient load transfer and improved mechanical properties of the composite, in accordance with the morphological analysis, (Fig. [4](#page-7-0)h). The same trend was also observed when NPK is added to the matrix, reaching a Young's modulus of 674 MPa. Similarly, the hybrid composite (MB/PLA-SLP-NPK) exhibited the highest increase in Young's modulus, exhibiting an *E* value of 707 MPa due to the fact that the total amount of particles (SLP and NPK) is 20%. For MB/PLA series, all variations were statistically significant (see Fig. S2) and the tensile behaviors seems to be mainly influenced by the reinforcing effect given by the fillers that stiffen the composites reducing their flexural deformability (see Table S1).

Nevertheless, it should be considered that the tensile properties of these types of systems can be optimized if a raster angle of  $0^{\circ}$  is chosen. In fact, in a previous work [\[13](#page-13-12)], we prepared green composites based on MB and tomato plant, and we printed them with a  $0^{\circ}$  or  $\pm 45^{\circ}$  raster angle in order to evaluate their influence on the tensile properties of the composites. Tensile test result reveals that the 0° raster angle definitively optimizes the tensile properties, confirming data from the literature, obtained on similar systems, where 0° raster angle usually optimizes tensile properties [[34](#page-14-12)]. However, in this case, a raster angle of  $\pm 45^{\circ}$  was chosen aiming to produce a more porous structure by creating separation points among adjacent



<span id="page-8-0"></span>**Fig.** 5 Reduced elastic modulus of the composites  $(E_c, i.e., normal$ ized to those of each matrix  $E<sub>m</sub>$ ), dashed line refers to neat matrices value of the reduced property (**a**); SEM micrograph of SLP particles

behavior during tensile fracture in MB-SLP (**b**) and MB/PLA-SLP (**c**) printed samples; **b** theoretical and experimental elastic modulus of MB (**d**), PLA (**e**) and MB/PLA (**f**) composites

raster layers. This promotes NPK release and Cu absorption, generates channels along the devices, and allows water to be conveyed inside the sample.

The experimental data obtained in tensile tests and those predicted by Halpin–Tsai model, according to Eq. ([1\)](#page-3-0), were analyzed.  $E_{SLP}$  and  $E_{NPK}$  were determined by nanoindentation, as reported in the experimental part. Density measurements were performed on matrices, SLP, and NPK, and the relative results are reported in Table S2. Figure [5](#page-8-0)d–f provides a comparison between the experimentally measured tensile moduli of MB, PLA, and MB/PLA printed composites and those predicted by the Halpin-Tsai model. For MB based composites, the theoretical model satisfactorily fitted the actual tensile moduli. On the other hand, for PLA composites, the theoretical model almost fitted the tensile modulus of PLA-SLP but overestimated PLA-NPK and PLA-SLP-NPK ones. This behavior could be attributed to the imperfect adhesion between the matrix and filler and the variable aspect ratio of the fillers, especially SLP. For MB/PLA blend composites a similar behavior to that observed for PLA was noticed for MB/PLA-SLP and MB/ PLA-NPK. However, for MB/PLA-SLP-NPK, the theoretical model clearly underestimated the actual tensile modulus. This issue could be addressed by considering that some

SLP fibers were uniaxially oriented in tensile direction (see Fig. [6](#page-9-0)), while the used Halpin-Tsai model fitted composites reinforced with randomly oriented fibers only.

The performances of SLP-filled composites in the tensile test were strongly influenced by uniaxially oriented fibers. Moreover, the presence of SLP, NPK or both during the preparation of the MB/PLA composites blend led to the production of matrices that were slightly different from the neat one, allowing to obtain a finer dispersion of one polymeric phase in the other (see whiter sphere in Fig. S3).

#### **3.7 NPK release and Peppas‑Korsmeyer model**

Conductivity measurements were used to determine the release profile of NPK from the 3D-printed samples. More in detail, electrolyte resistance was measured and correlated to the total concentration of electrolytes in solution. All tests were performed in water at 25 °C. It should be noted that this evaluation method, performed in water, may overestimate the NPK release compared to what typically occurs in soil. As a control, conductivity measurements were also performed for free NPK and for composites that do not contain NPK (see Figs. [7](#page-9-1)a and S4). When immersed in water, free NPK immediately achieved a plateau region, releasing all

<span id="page-9-0"></span>

fertilizer within the first 24 h. Devices without NPK (MB, MB-SLP, PLA, PLA-SLP, MB/PLA, MB/PLA-SLP) did not release any significant concentration of electrolytes after 30 days of immersion.

Conductivity tests were also conducted on NPK-filled composites to determine their release kinetics and the results are shown in Fig. [7](#page-9-1)b. The NPK release kinetics of all the 3D-printed composites were characterized by three phases: a burst release in the initial 24 h of immersion, a subsequent phase of progressive and slower release, and a final plateau region observed after 15 days of immersion, in agreement with already reported similar systems [\[30,](#page-14-8) [50](#page-14-28)].



<span id="page-9-1"></span>**Fig. 7** Fertilizer release as function of time for free NPK, composites that not contain NPK (**a**) and 3D-printed devices (**b**); comparison of release rate of MB-SLP-NPK with other CRFs device obtained with diferent methods (**c**); logarithmic plots of the Peppas-Korsmeyer model power law applied to the release data collected in the burst region,  $M_t/M_\infty$  < 0.6 was highlighted by colored portion of the plot (**d**); pictorial description of NPK release mechanism (**e**)

The sustained release of NPK fertilizer was achieved for both composites (-NPK) and hybrid composites (SLP-NPK) prepared using the three different matrices. All composites showed a significant reduction in the amount of released NPK and therefore in conductivity values, especially in the first 24 h, compared to free NPK. Moreover, the NPK release rate could be modulated by adding SPL particles. In detail, MB-SLP-NPK showed the fastest delivery with about 80% of NPK released in the first week of the test, followed by PLA-SLP-NPK with 70% of NPK released. Other composites exhibited a lower release of NPK: after 7 days, MB/PLA-SLP-NPK released about 60% of NPK while all the composited that not contain SLP released about 55% of it. In fact, when the samples were immersed in water, SLP particles were also released into the aqueous medium, generating channels along the devices and allowing the water to be conveyed inside the sample.

This suggests that NPK particles, found within the interior of the samples, may have a higher potential to be released. Moreover, a weak adhesion between matrix and filler would promote filler and fertilize release, thus justifying the more inhibited release of MB/PLA systems. Therefore, the results imply that, if a faster and complete release is required, SLP particles should be added to the composites and MB should be preferred as polymeric matrix. MB-SLP-NPK system, in fact, release 100% of NPK after 30 day of immersion, achieving a complete release of the fertilize. The quantity of NPK released for each sample after 30 day is reported in Table S3. When compared to other controlled-release fertilizer devices mentioned in the scientific literature, the devices produced in this study displayed a remarkable capacity to reduce the release rate of NPK, as evident in Fig. [7](#page-9-1)c [\[51](#page-15-0)[–53\]](#page-15-1).

The obtained NPK release data was evaluated by plotting logarithmic graphs as a function of time (Fig. [7d](#page-9-1)) to gain a better understanding of the fertilizer release mechanism. During the firsts 24 h of release test, burst release was observed. This phase can be divided in two sub-phases: at first the fertilizer available on composites surface was rapidly delivered (within the first 15 min) and subsequently the delivery was driven by diffusive phenomena (colored portion of the plot).

As the release progressed, the release curves showed an increase in slope likely due to the occurrence of swelling phenomena across the polymeric matrix possibly caused by the higher porosity induced by the printing process that reasonably led to a relevant increase in water uptake for FDM structures [\[30\]](#page-14-8). In the final stage, fertilizer delivery was reasonably governed by both swelling and diffusive phenomena. Peppas-Korsmeyer mathematical model [[54\]](#page-15-2) was used to fit the obtained release data and confirm the hypotheses on NPK release mechanism [\[55\]](#page-15-3). This model has previously been successfully used to describe fertilizer release kinetics from CRFs devices [[30](#page-14-8), [56](#page-15-4)]. The values of *K* and *n* of the different formulations were determined and reported Table [3](#page-10-0).

Analyzing the calculated *n* value is possible to make assumption about the release mechanism: *n* approximately 0.5 indicate a diffusion-driven release; when  $n = 1$  swelling occurs and when *n* values are between 0.5 and 1 both swelling and diffusive phenomena occurs [\[32](#page-14-10)]. A value of  $n < 0.5$  may indicate that the matrix hinders the diffusion of the particles. As seen previously, the release curves of all composites are characterized by a burst region that represent two different effect:

- (i) NPK particles readily available on the device's surface are immediately delivered in the aqueous media  $(t=15 \text{ min})$
- (ii) The real capability of the 3D-printed devices to perform a controlled release

In this second region, all the systems displayed *n <*0.5 (see *n* value in Table [3\)](#page-10-0); therefore, the fillers could indeed slow the fertilizer release. In general, in the first 24 h, all the composites showed a hindered diffusion of the NPK, especially PLA-NPK, MB/PLA-NPK, and MB/PLA-NPK  $(n =$ 0.33, 0.32, 0.36, respectively). CRFs composites showed an almost diffusive release mechanism, unlike NPK grains. In fact, in this latter case, NPK is immediately released for erosive dissolution. Figure [7](#page-9-1)e provides a pictorial representation of the release mechanism of the CRFs composites described above.

<span id="page-10-0"></span>**Table 3** Values of slopes (*n*) and intercepts (*k*) of ftting of power law applied to the release data collected in the burst region ( $M_t / M_\infty$  < 0.6)

Sample	K	n
<b>MB-NPK</b>	0.46	0.42
<b>MB-SLP-NPK</b>	0.41	0.43
PLA-NPK	0.48	0.33
PLA-SLP-NPK	0.59	0.41
MB/PLA-NPK	0.48	0.32
MB/PLA-SLP-NPK	0.53	0.36

Morphology of CRF devices was analyzed also after 30 days of soaking in water. Optical image (Fig. [8](#page-10-1)a-f) and SEM micrograph (Fig. [8](#page-10-1)a'–f') of the 3D-printed samples before (Fig. [8](#page-10-1)a–c and a'–c') and after (Fig. 8d-f and d'-f') NPK release were reported. Before the release, all the samples exhibited a satisfying adhesion of adjacent layer especially MB/PLA one (Fig. [8c](#page-10-1) and c') that also showed an excellent print definition. After soaking in water for 30 days, instead, PLA sample (Fig. [8](#page-10-1)e and e') showed a clear detachment of the layers, which was even more marked for MB one (Fig. [8](#page-10-1)d and d'). This behavior testifies the occurrence of swelling of the polymeric matrices confirming the supposed mechanism of release. Moreover, the PLA samples appear discolored after release, likely due to filler loss, as observed by comparing the optical images in Fig. [8b](#page-10-1) and e. With regard to the MB/PLA samples, the swelling phenomenon was less evident compared to MB and PLA ones. From the SEM micrograph in Fig. [8e](#page-10-1)', it is possible to observe that the layers do not show any detachment, but on the contrary, they appear to be better adherent to each other. Once again, this different behavior may be caused by the greater crystallinity of the blend compared to one of the single components. Moreover, this behavior reasonably explains the slower NPK release obtained for the MB/PLA samples compared to that recorded for MB and PLA ones.



<span id="page-10-1"></span>**Fig. 8** Optical image (a–f) and SEM micrograph (a'–f') of good adhesion between layers in FDM samples before NPK release (a–c and a'–c') and layers detachment in FDM samples after soaking in water for 30 days due to occurrence of swelling (d–f and d'–f')





<span id="page-11-0"></span>Fig. 9 Removal efficiency of the devices (a) and SEM micrographs (b–j) and optical images (c', f', i') of the green composites after the immersion in copper sulfate aqueous solutions for 30 days

## **3.8 Cu(II) absorption**

All the 3D-printed green composites were separately immersed in copper sulfate aqueous solutions for 30 days. UV spectroscopy was employed to test the Cu(II) ion removal efficiency of the samples, calculated according to Eq. (4), and the outcomes are reported in Fig. [9](#page-11-0)a. All the green composites exhibited a significant increment in the removal efficiency over the neat MB, PLA, and MB/ PLA that were 14, 10, and 8%, respectively. Despite plastics are considered inert, previous studies have reported the adsorption of metal ions onto various biopolymeric material, including PLA and PBAT [\[31](#page-14-9), [57\]](#page-15-5). The highest removal efficiency for all the three type of matrices was observed for the -SLP an -SLP-NPK composites. In detail, the maximum removal efficiency was achieved by MB-SLP-NPK with 78% of Cu removed from the aqueous solution, followed by MB-SLP with 74%. MB-NPK showed a removal efficiency of 46%. Both PLA and MB/PLA composites showed the same trend. However, in both cases, a lower removal efficiency was achieved, especially for MB/PLA composites, which exhibited the lowest values. This behavior can be explained by considering the more compact structure of the MB/PLA samples, which remains almost unchanged after immersion in water for 30 days, preventing water from being conveyed inside the sample and thus reducing the surface area available for the absorption.

These results clearly demonstrated that Cu removal efficiency of the composites could be modulated by adding SPL particles. Indeed, the composites that do not contain SLP exhibited the lowest removal efficiency values of the series. This behavior could be reasonably explained by taking into account that, SLP contains natural compounds such as cellulose, lignin, and hemicellulose that have a high affinity for metal ions [[2,](#page-13-1) [58](#page-15-6)]. It is reasonable to assume that these compounds can absorb copper ions, effectively removing them from the aqueous solution. Moreover, the release of NPK particles generated channels along the polymeric matrices, facilitating the water to be conveyed inside the sample. This promoted the contact with more SLP particles, resulting in higher removal efficiency for -SLP-NPK samples. SEM images in Fig. [9](#page-11-0)b–j reveal that, for the -SLP and -SLP-NPK samples, the majority of the copper was absorbed by the SLP particles, identifiable as white spots in the micrographs (confirmed by punctual EDAX analysis; sample test identifying copper on white spots; see Fig. S5). On the other hand, for the -NPK samples (as shown in SEM and optical images in Fig. [9](#page-11-0)c–c', f–f', and i–i') Cu(II) clusters formed on the polymeric matrices (confirmed by punctual EDAX analysis; sample test identifying copper on white spots; see Fig. S5) which could also be easily identified through visual inspection thanks to the pale color of the composites.

## **4 Conclusions**

Biodegradable green composites for contextual controlled soil fertilization and Cu removal were produced by 3D-printing. NPK fertilizer flour and tomato plant waste particles (SLP) were added to three different biodegradable polymeric matrices: polylactic acid (PLA), a commercial blend of biodegradable co-polyesters (Mater-Bi®, MB) and their blend (MB/PLA, 50:50). All the obtained filaments were successfully printed by FDM, in agreement with morphological analysis and rheological characterization. Morphological analysis performed on the cross section of 3D printed composite samples confirmed a relatively uniform dispersion of both natural filler and fertilizer in all composite samples. Moreover, MB and MB/PLA samples displayed good adhesion between the fillers and the matrix, unlike PLA ones where imperfect adhesion was observed. These behaviors affect the mechanical performance of the composites. For MB and MB/PLA the filler and the fertilize effectively act as reinforcement. SLP and NPK elastic moduli were evaluated by powder nanoindentation and, for the majority of the composites, the theoretical Halpin-Tsai model satisfactorily fitted the actual tensile moduli. NPK release tests showed the ability of all the obtained composites to slow the release rate of the fertilization up to 30 days. Moreover, it was possible to tune the release kinetics by changing the polymeric matrix and adding the natural filler (SLP). Cu(II) removal tests confirmed that all the green composites exhibited a significant increment in the removal efficiency over the neat matrices. More in detail, the Cu removal efficiency could be increased by adding SPL particles that, as expected, have been very successful in absorbing copper. MB-SLP-NPK showed 100% release of NPK after 30 days and the highest copper removal ability with 78% of Cu removed from the aqueous solution. Moreover, the addition of SLP particles leads to a decrease in the amount of polymer needed, resulting in a lower cost of the final device. In addition, the use of FDM as the selected production technique allows to reduce production time and costs. Considering the release kinetics results and Cu absorption test, it is possible to conclude that by adding tomato plant waste particles to polymeric matrices, it was possible to produce biodegradable, cost-effective, and easy-toproduce devices for contextual and efficient fertilizer release and Cu(II) ions capture.

**Supplementary information** The online version contains supplementary material available at<https://doi.org/10.1007/s42114-024-00908-4>.

**Acknowledgements** Nanoindentation experimental data were provided by ATeN Center of University of Palermo.

**Author contribution** Maria Clara Citarrella: writings—original draft; Roberto Scaffaro and Maria Clara Citarrella: writing—review and editing; Emmanuel Fortunato Gulino and Maria Clara Citarrella: software, formal

analysis, investigation, visualization; Roberto Scaffaro, Emmanuel Fortunato Gulino and Maria Clara Citarrella: conceptualization, data curation, methodology, validation; Roberto Scaffaro: supervision, resources, project administration, funding acquisition.

**Funding** Open access funding provided by Università degli Studi di Palermo within the CRUI-CARE Agreement. SAMOTHRACE (Sicilian micro and nano technology research and innovation center) extended partnership and received funding from the European Union Next-GenerationEU (PIANO NAZIONALE DI RIPRESA E RESILIENZA (PNRR) – MISSIONE 4 COMPONENTE 2, INVESTIMENTO 1.5); MICS (Made in Italy – Circular and Sustainable) extended partnership and received funding from the European Union Next-GenerationEU (PIANO NAZIONALE DI RIPRESA E RESILIENZA (PNRR) – MISSIONE 4 COMPONENTE 2, INVESTIMENTO 1.3 – D.D. 1551.11-10-2022, PE00000004); PRIN: Green composites based on biodegradable polymers and vegetal biomasses of Mediterranean area: processing, characterization and degradability - Bando 2022 - Prot. 20228WNZ2Z.

**Data availability** No datasets were generated or analysed during the current study.

## **Declarations**

**Competing interests** The authors declare no competing interests.

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