

1. INTRODUCTION

Silicon Carbide (SiC) is an advanced material belonging to the family of **Ultra Wide Bandgap** semiconductors known for its excellent characteristics and a promising material for high power applications due to its potential to improve the energy efficiency and lowering energy consumptions. For these reasons, there is a rising interest from researchers and industry to synthesize good quality wafers with high carrier mobility and low defect density. For electronic applications a thin oxide gate is needed, which is usually produced by Chemical Vapour Deposition (CVD) of SiO_2 . It is crucial that the defect density at the interface between SiC and SiO_2 is as low as possible to maintain good electrical characteristics. Therefore, several **Post Deposition Annealing (PDA)** methods have been developed to passivate and reduce these defects located on the interface between oxide and semiconductor. However, these methods usually involve high temperature (1100- 1300 °C) thermal annealing in gases as NO , N_2O , $POCl_3$ at atmospheric pressure for a short time interval. Here we present a different PDA conducted at low temperature (400 °C) in an high pressure (50 bar) pure O_2 atmosphere. This method could provide valuable insight on the fundamental oxidation properties of SiC and at the same time offer a cost-effective alternative to standard PDA.

2. METHODS

The presence of defects in SiC crystal lattice induces variations in the vibrational modes, that can be highlighted using **Raman spectroscopy**.

It is also possible to evaluate the quality from the **exciton lifetime τ** . In our experimental setup (**Fig. 1**) we measure the PL intensity decay as a function of time after the excitation pulse. The PL intensity follows:

$$I(t) = I(0) \exp(-t/\tau)$$

We conducted **thermal treatments** in high pressure O_2 atmosphere (300 – 800 PSI) in a reactor like in **Fig. 2**.

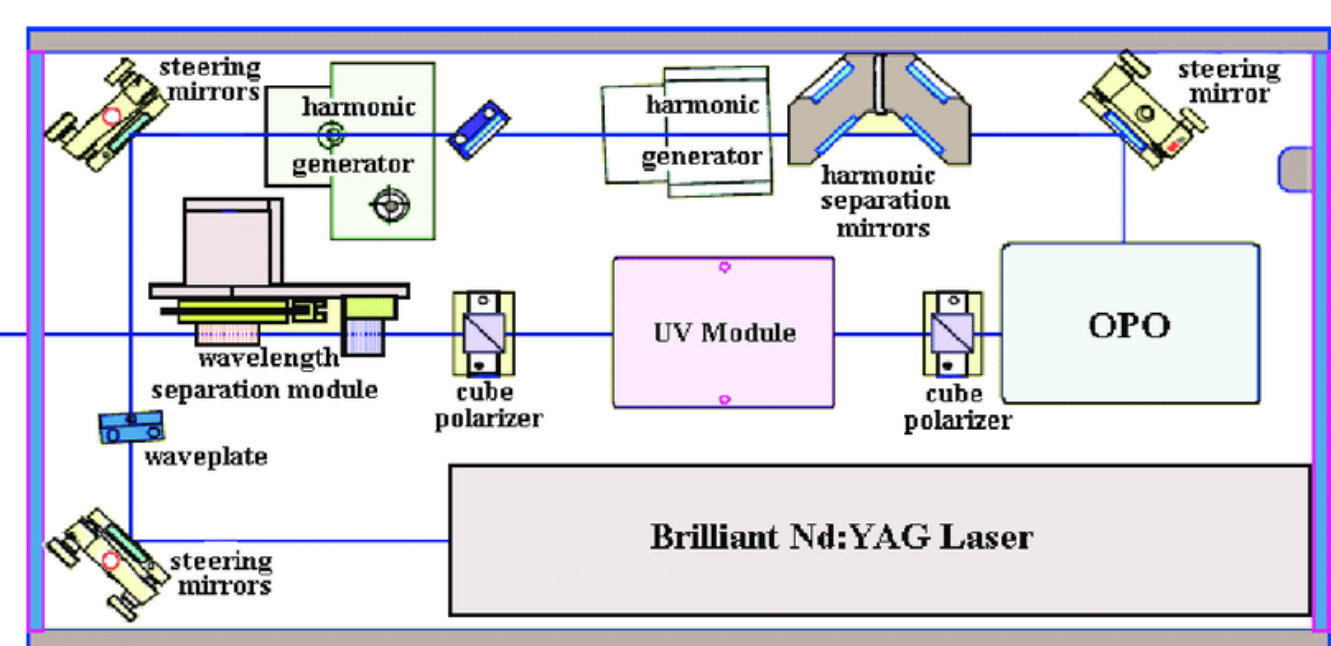


Figure 1: Schematic of tunable laser system used for TRPL

Figure 2: Parr reactor for thermal treatments

4. CHARACTERIZATION

Comparing samples' Raman spectra, we can see (**Fig. 3**) that in the Longitudinal Optical Phonon Coupled (LOPC) mode ($\sim 980 \text{ cm}^{-1}$) a "shoulder" appears on the right in samples with oxide layer (CIG6, 11EA). Given that the LOPC mode is highly sensitive to doping,^[1] this feature could be attributed to an increased density of defects at the SiC/SiO_2 interface and can give us information about the different doping levels between epitaxial layer and substrate.

When excited at $\lambda = 266 \text{ nm}$, samples show an emission peak centered at 390 nm (**Fig. 4**) due to excitonic recombination. We want to follow the time evolution of this excitonic recombination process under the assumption that the **mean lifetime τ** is a quality indicator of the material:^[2,3] high τ means low defect density and vice versa.

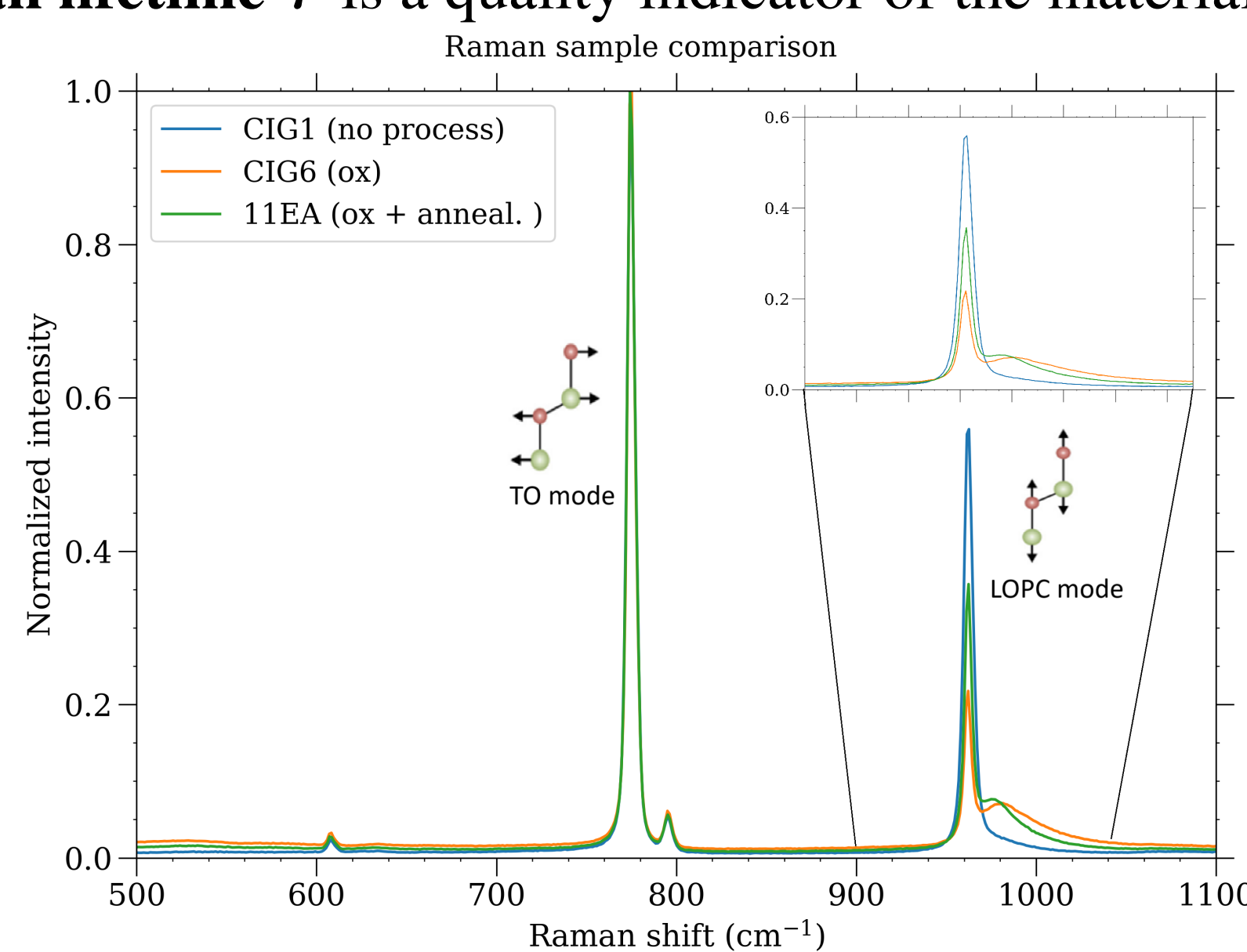


Figure 3: Raman spectra, comparison between samples.

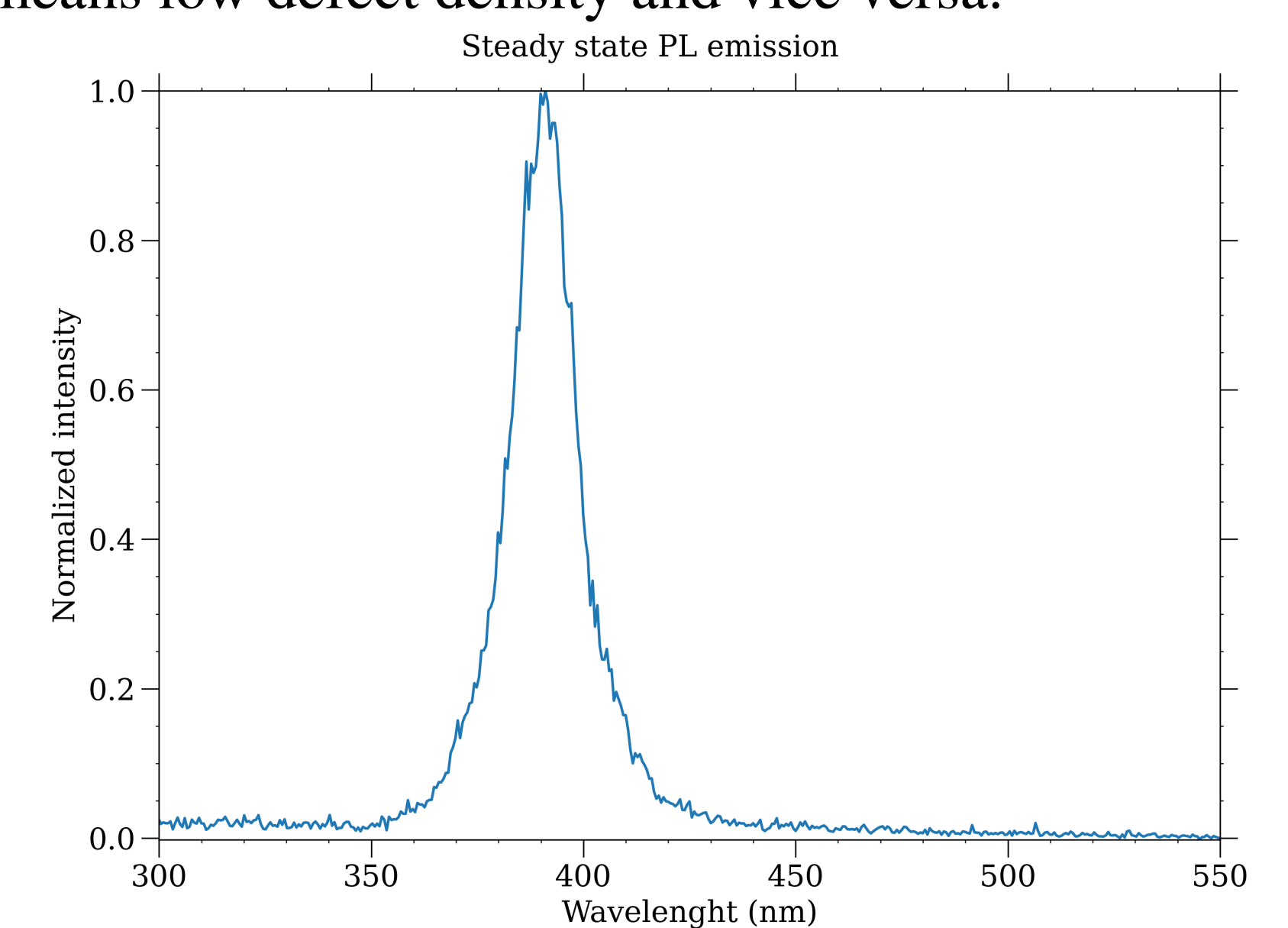
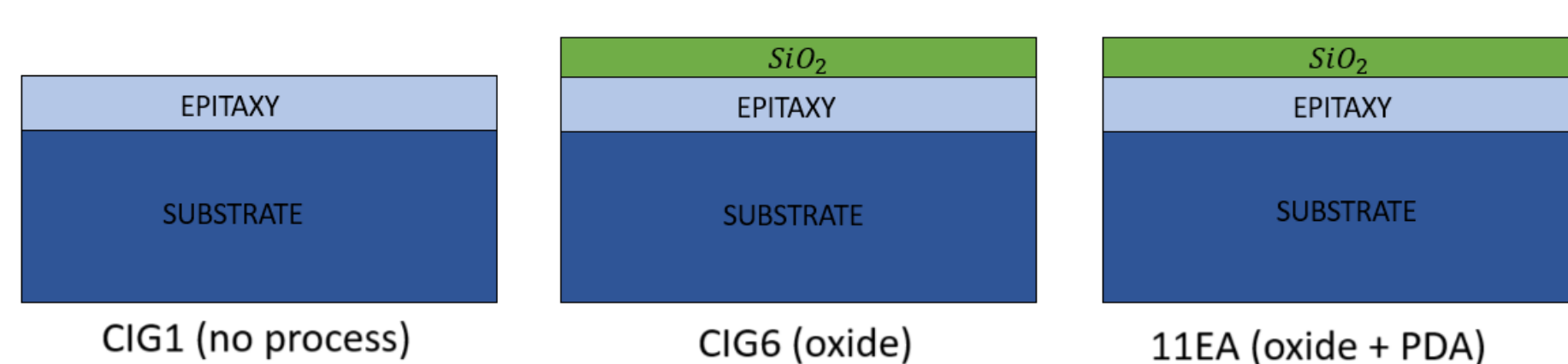


Figure 4: Steady state PL emission showing a single peak centered at 390 nm .

3. SAMPLES

Samples studied in this work are representative of three steps in the production line of SiC wafers.



- **CIG1:** Epitaxial layer of SiC on a highly-doped substrate
- **CIG6:** a 50 nm SiO_2 layer is deposited on the epitaxial SiC layer
- **11EA:** subjected to thermal annealing (PDA) in NO in order to improve electrical characteristics

6. CONCLUSION

- Samples show two different initial exciton lifetime before thermal treatments
- Different results were obtained for pressure above or below $P_{crit}(O_2) = 730 \text{ PSI}$

SAMPLE	RESULTS
CIG1	τ decreases for $P > P_{crit}$
CIG6	τ increases for $P > P_{crit}$
11EA	τ not affected by thermal treatments

- Thermal treatment at $P = 300 \text{ PSI}$ shows no change in lifetime
- CIG6 undergoes a recovery process similar to that obtained with NO PDA

Further studies will involve a more in-depth analysis on the time-dependence of this thermal annealing process, XPS characterization and electrical parameters measurements.

5. RESULTS

A first characterization on as received samples shows that **CIG1** and **11EA** have higher lifetime $\tau = 20 \text{ ns}$ compared to **CIG6** which has $\tau = 10 \text{ ns}$ (**Fig. 5**)

For each sample, thermal annealing process is performed in O_2 at $T=400 \text{ °C}$ for different pressures and durations.

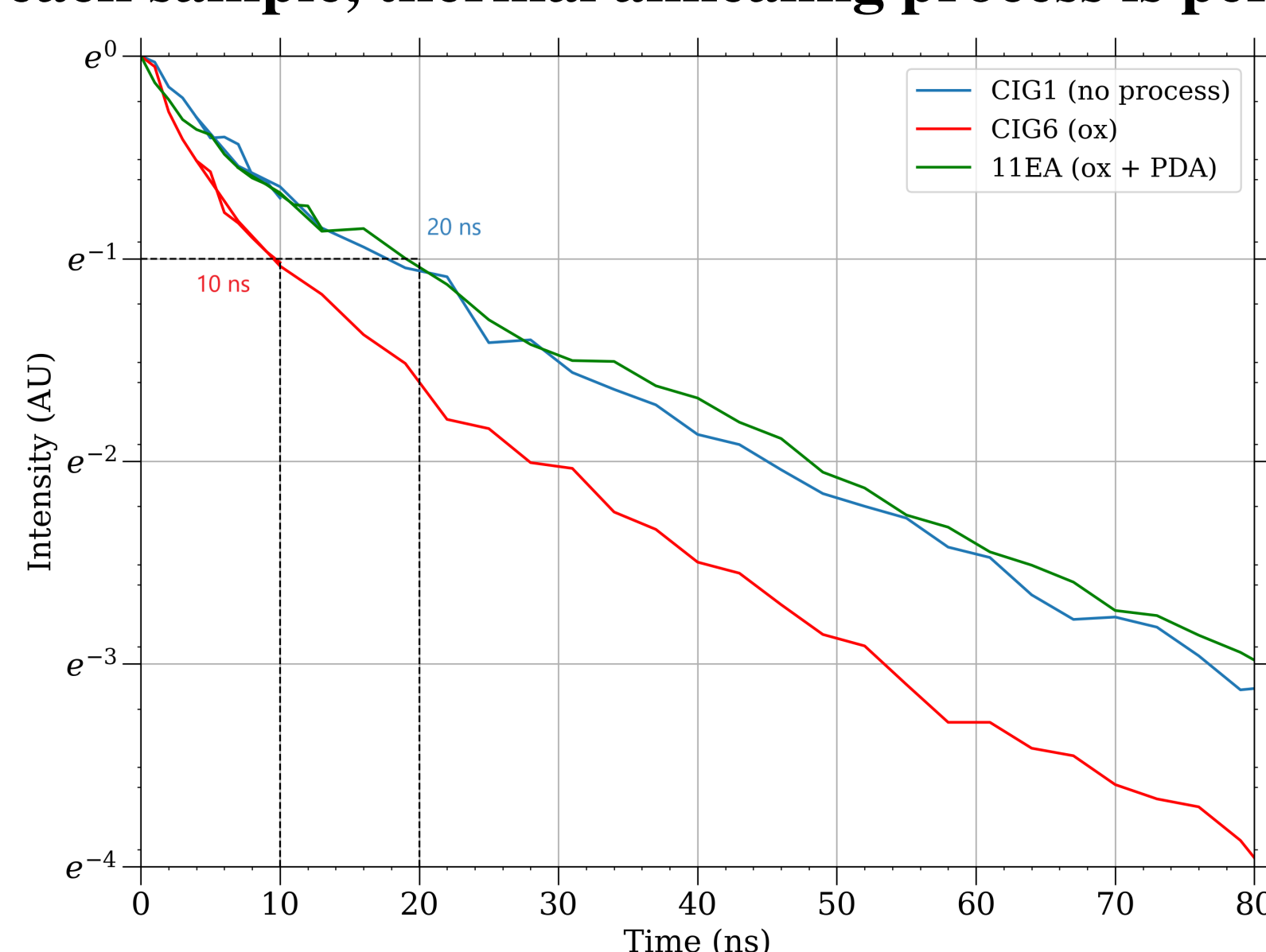


Figure 5: As received sample

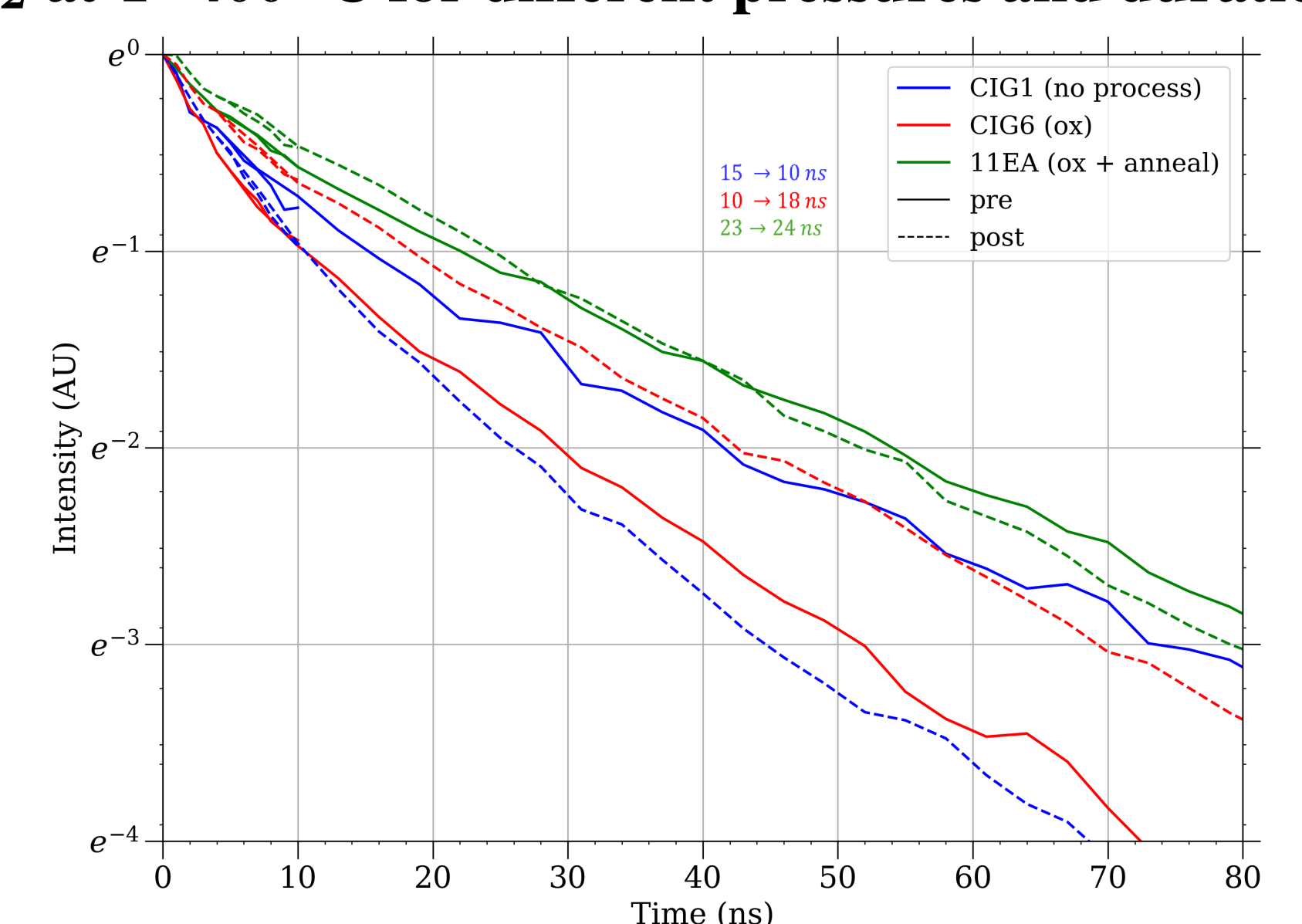


Figure 6: $T=400 \text{ °C}$, $P=800 \text{ PSI}$, $t=48\text{h}$

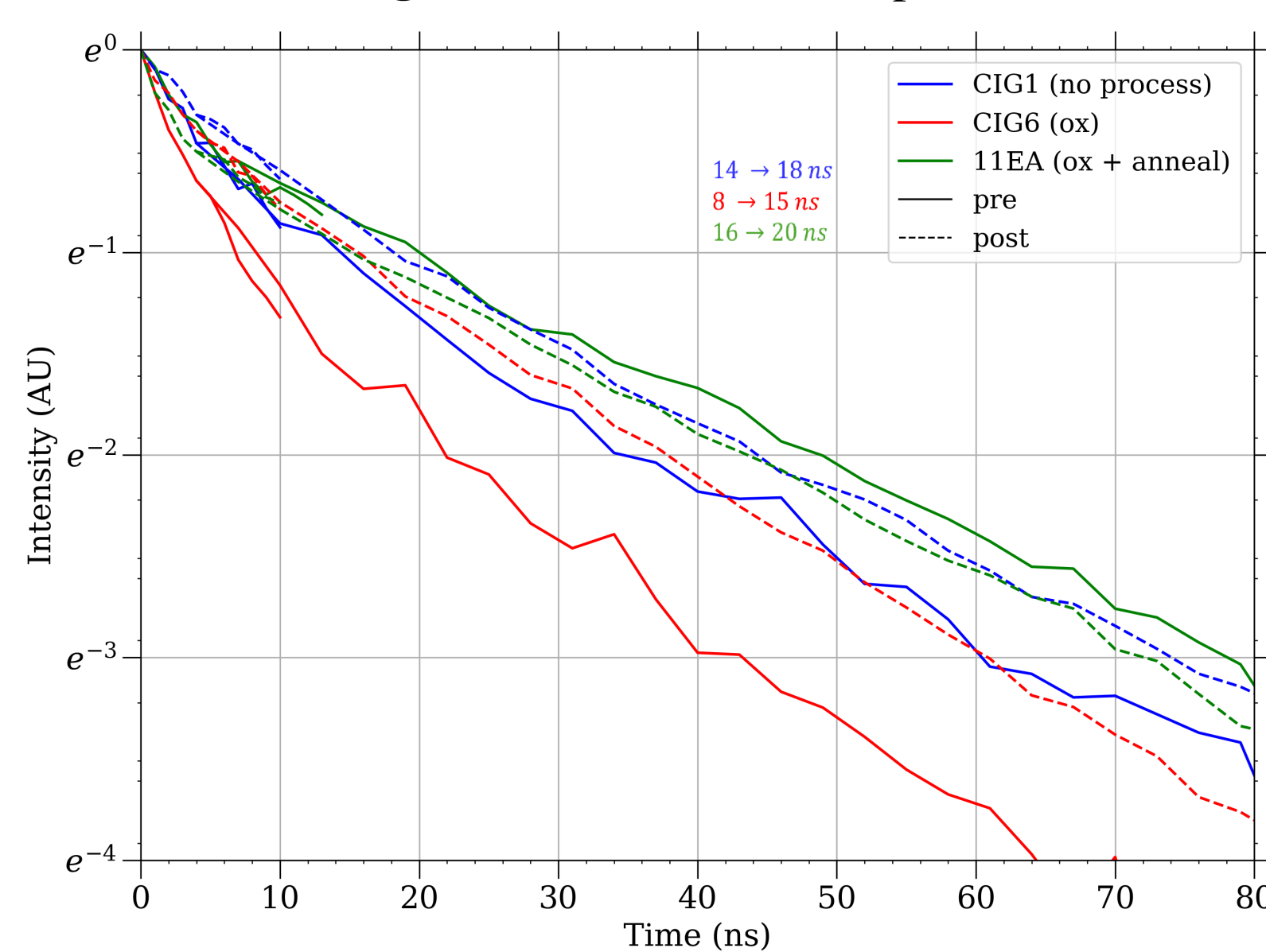


Figure 7: $T=400 \text{ °C}$, $P=800 \text{ PSI}$, $t=2\text{h}$

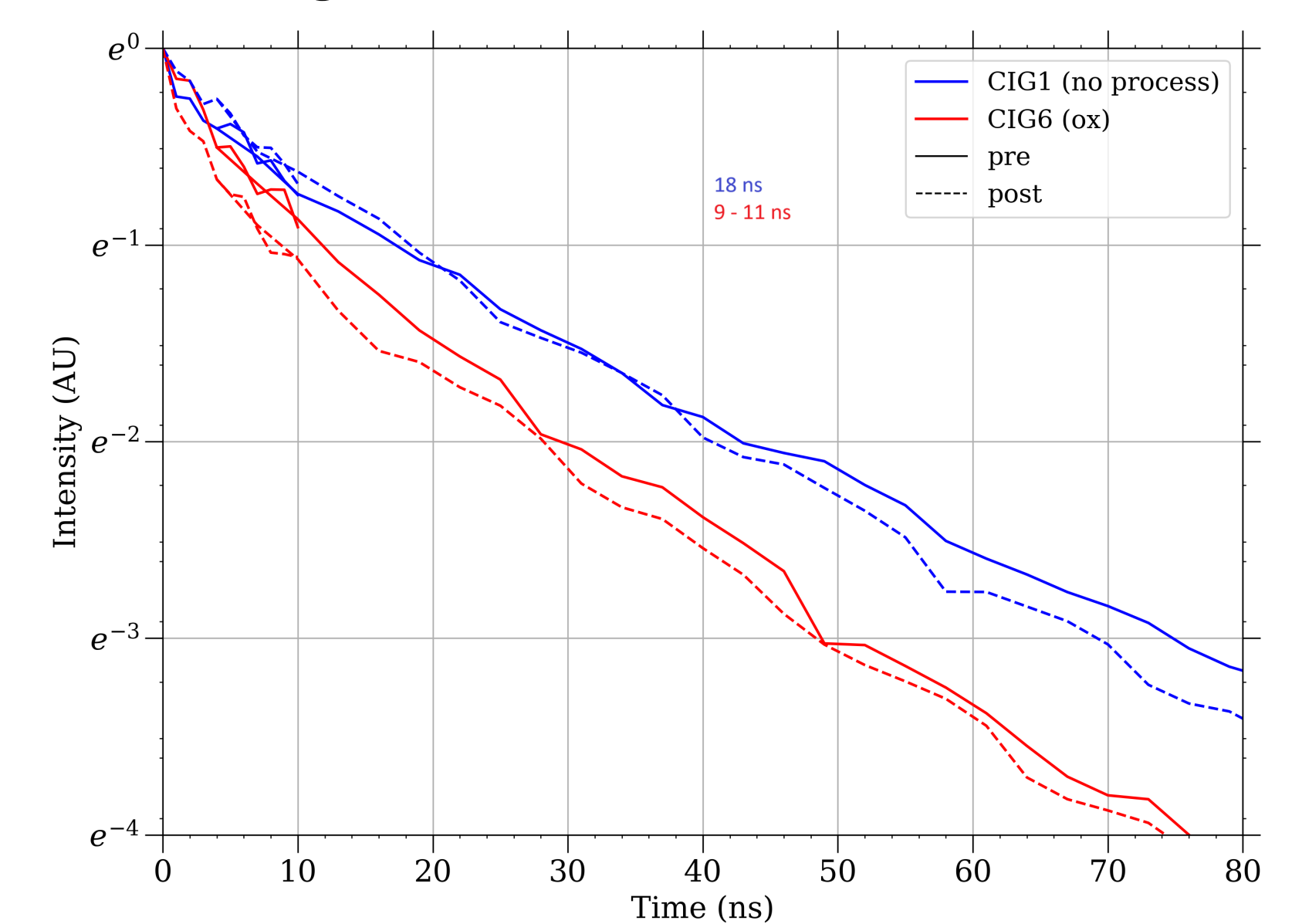


Figure 8: $T=400 \text{ °C}$, $P=300 \text{ PSI}$, $t=2\text{h}$

5. REFERENCES

- [1] Sun, G.S., et al., 2011. Determination of the transport properties in 4H-SiC wafers by Raman scattering measurement. Chinese Physics B
- [2] Migliore, F. et al, 2024, β -rays induced displacement damage on epitaxial 4H-SiC revealed by exciton recombination, Applied Physics Letters
- [3] Tawara, T. et al, 2004. Evaluation of Free Carrier Lifetime and Deep Levels of the Thick 4H-SiC Epilayers. MSF 457-460, 565-568.